

**Increasing use of composites in Aircraft, Automotive and Civil Engineering requires a better understanding of its behaviour under static, cyclic and dynamic loading.**



***... therefore, reliable material models and properties are required !***

## Ralf Cuntze: *Retired engineer and hobby material modeller*

*Formerly:* MAN-Technologie AG Augsburg, Head of Main Department 'Structural and Thermal-Analysis'

*Now:* linked to Carbon Composites e.V. (CCeV) Augsburg and IHK-Schwaben, Bayern Innovativ

- 1959 - 1964 Study of Civil Engineering at Hannover ('stress man')
- 1968 Dr.-Ing. in Structural Dynamics
- 1978 Dr.-Ing. habil. in Mechanics of *Lightweight Structures*
- 1980 - 2003 Lecturer at 'Universität der Bundeswehr' München on fracture mechanics and *lightweight structures* from Fiber Reinforced Plastics (FRP)

### Professional career:

1968 - 1970 DLR (German aerospace center: finite element analysis programming)

1970 - 2004 MAN-Technologie, Munich/Augsburg: involved in the *Development of:*

- ARIANE 1-5 Launcher family: Components of central stage + Boosters, high pressure vessels, etc.
- Windenergy rotors (Growian Ø103m, WKA 60, Aeroman). *Since 1970 in Carbon Comp. business*
- Satellite components, Space Antennas; FRP light weight structures, IRAM antenna, SOFIA telescope (in Boeing 747); Automated Transfer Vehicles (ATV1 Jules Verne) + Crew Rescue Vehicle CRV (for ISS) + NASA X38 Demonstrator space plane
- CMC body flap for Crew Rescue Vehicle (2002), Spacelab mission D1 (1985): material experiments,
- Apogee solid propellant motor cases MAGE and IRIS, water-tanks for AIRBUS
- Heat exchanger for gas cooled Solartower GAST(20 MW) and Solarfield constructions (Almeria)
- Gasultra-Centrifuge, Fly wheels (for ship "Trans Swartow", MAN-Buses), Diesel engine parts in metal and in monolithic ceramics for trucks, Structural calculations for MAN buses
- Fusion reactor WENDELSTEIN VII: toroidal ring chamber.

*Material applications in the range: 20 K through 2000 K (FRP, CMC, metals, concrete).*

### Co-author and Convenor of various Handbooks and Working Groups

- IASB - German Aircraft Structural Handbook HSB: since 1972 author and co-author of a large number of design sheets. Co-author of the HSB-Handbook's 'transfer' team into English
- VDI 2014 Guideline "Development of Fibre Reinforced Plastic Compon." (co-author, convenor), VDI-WG 4.3 "Reliability of Structural Components" (1980ies, WG member)
- ESA/ESTEC, since 1978: Structural Requirements Standards, co-author in the 3 working groups: Structural Analysis, High Pressure Vessels (metal and composites), Safety Factors; Contributor to Handbooks: PSS and follower Structural Materials Handbook SMH; Buckling Handbook (first convenor, contributor to several chapters); ECSS (European Cooperation for Space Standardization)
- was involved in EU projects (BRITE, BRITE-EURAM), and BMFT and BMBF research projects.

### Miscellaneous

- Surveyor/advisor (since 1980) for German Ministry BMFT + BMBF on R&D Material Programmes (MaTech, MatFo, LuFo, German Material Modelling competence centres). Advisor to EU-project MAAIXIMUS (Airbus Toulouse: On improving Aircraft sizing)
- Advisor for German Research Foundation DFG (SFBs, SPPs on modelling structural textiles)
- Originator of the (never funded) successful Failure Mode Concept (FMC), a general invariant-based foundation for the derivation of failure mode-linked *strength failure conditions* for isotropic, transversely-isotropic (UD), and orthotropic (fabrics) materials applicable for FRP, CMC, metals, concrete, etc. Winner of the World-Wide-Failure-Exercise-I on "UD composites strength failure theories" (WWFE-I, bi-axial stress states). Now, top-ranked in WWFE-II on "UD, tri-axial stress states" considering hydrostatic pressures > 700 MPa
- Numerous 'single-author' publications in different structural fields
- Founder (2010) and co-worker of the working group 'Fatigue of Composites' (only group, world-wide) of all respective German universities. Founder and Leader of the Carbon Composites working groups 'Engineering' (2009) and 'Fiber-Reinforcement in Civil Engineering' (2011). Member of the world-wide NAFEMS Composite Working Group

**1964:**        **Diplom**        *Statiker*

**1968:**        **Dr.-Ing.**        *Strukturdynamik*

**1978:**        **Dr.-Ing. habil.** *Mechanik des Leichtbaus*

**1968- 1970:** frühere DLR    *Finite Element Analyse*

**1970-2004:** MAN-Technologie (GUZ, Raumfahrt, Wind- und Sonnenenergie, ...)

**1980-2002:** Dozent an der Universität der Bundeswehr

**jetzt:**        **Ingenieur, Unruheständler + Simulant**

*Convenor of CCeV working groups : '(mechanical) Engineering and 'Modelling Fiber Reinforcement in Civil Engineering'*

### Theoretical works in the areas:

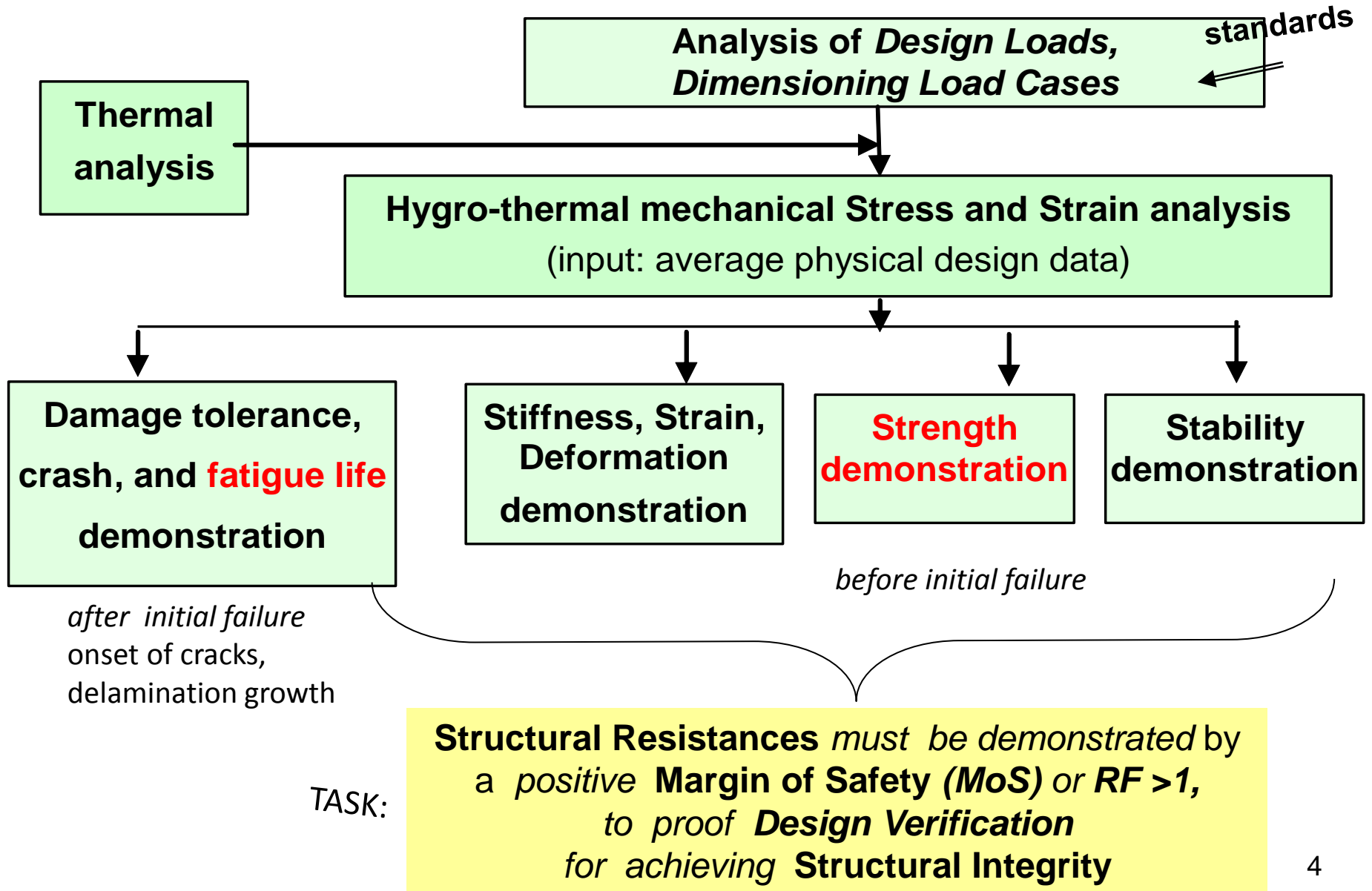
**Finite Element Analysis, Structural and Rotor dynamics,**

**Structural reliability and Development policy,**

**Strength failure modes and hypotheses (isotropic + composites),**

**Composites fatigue, Damaging mechanics and Fracture mechanics.**

# Which Design Verifications are mandatory in Structural Design ?



# Verification Levels of the Structural Part with

- Local Stress at a critical material 'point': **continuum mechanics, strength criteria**  
verification by a basic strength or a multi-axial failure stress state  
*Applied stresses are local stresses*
- Stress concentration at a notch (stress peak at a joint): **notch mechanics**  
verification by a *notch strength* (usually Neuber-like, Nuismer, etc..)  
*'Far'-field stresses are acting and are not directly used in the notch strength analysis*
- Stress intensity (delamination = crack): **fracture mechanics**  
verification by a *fracture toughness* (energy-related)  
*Applied stresses are 'far'-field stresses. (far from the crack-tip)*

... gilt statisch wie zyklisch

# CONSTRAINTS in Design Development Process : *Cost and Time Reduction*

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Industry looks for robust & reliable analysis procedures in order to replace the expensive ‘Make and Test Method‘ as far as reasonable.

***Virtual tests shall reduce the amount of physical tests.***

In this context:

**Structural Design Development**

**can be only effective and offer high fidelity**

**if**

**qualified analysis tools and necessary test data input are available**

**for Design Dimensioning and for Manufacturing as well.**

*a Strength Failure Condition (SFC) is for instance such an Analysis Tool*

The presentation plus further literature may be downloaded from <http://www.carbon-composites.eu/leistungsspektrum/fachinformationen/fachinformation-2>

## **Material Properties and Model Parameters, necessary for the Analysis of Static, Cyclic, and Dynamic Stress States**

- embedded in Structural Design Development

**Short Presentation of CCeV + personal activities**

- 1. Structural Development, Design Requirements, and Design Verifications**
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- 4. Material Strength Failure Conditions (SFC)**
- 5. Application of SFCs to Some Materials**
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- 8. Model Parameters**
- 9. Standardized Material Test Methods**
- 10. Structural Testing, NDI, Damage Tolerance**
- 11. Structural Verification, Reserve Factor**

(matrix, fiber, interphase, composite)

*Prof. Dr.-Ing. habil. Ralf Cuntze VDI*

*retired from MAN-Technologie, now linked to Carbon Composites e.V. (CCeV), Augsburg*

*presents results of a time-consuming 'Hobby'*

**Carbon Composites eV (CCeV) =**  
**Association of companies and research institutions,**  
**covering the entire value chain of**  
**high-performance fiber reinforced composites**  
**in Germany, Austria and Switzerland (DACH).**

**Focus : Promotion of Carbon Fiber Technology**

**Serving as competence network :**

- Support and linking collaboration between science, small and large companies
- Transfer of available know-how and existing competences
- Organized as an association
- Founded in 2007, based in Augsburg
- Financed by membership fees
- The leading Carbon Composites Network in the German-speaking world !





### Regional Departments

CC OST	2012
CC SÜDWEST	2012
CC SCHWEIZ	2012
CC AUSTRIA	2012

### Cluster Department

M-A-I CARBON	2012
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### DACH Area: Specialist Departments

CERAMIC COMPOSITES	2008
CC TUDALIT	2012

Together with TUDALIT e.V.

● = Research Institution    ● = Industry

**Distribution of the - at present - 275 members**

# Sectors

## System companies

- Aerospace
- Automotive engineering
- Civil engineering
- Medical technology
- Energy technology
- etc.

## Supplier companies

- Fibres, semi-finished products, ancillary materials, coatings
- Assemblies, components
- Tooling machines, processing systems, equipment, plants
- Software and services  
(e.g. engineering, factory planning)



Bildnachweis: Airbus, ALIEN-Projektteam, KUKA

# CCeV's activities

## Technical working groups - fiber-reinforced plastics

### *... Material*

AG Materialien

AG Garne und Textilien

AG Thermoplaste

AG Biocomposites

AG Faserbewehrte Kunststoffe im Bauwesen

### *... Design & Characterisation*

AG Engineering

UAG Composite Fatigue

AG Multi-Material-Design

AG Klebtechnik

AG Smart Structures

AG Werkstoff- und Bauteilprüfung

AG Werkstoffmod./Berechn. im Bauwesen

### *... Process*

AG Herstellverfahren

AG Automatisierung

UAG Herstellprozess-Simulation

AG RTM Next Steps

AG Werkzeug- und Formenbau

### *... Finishing*

AG Bearbeitung

UAG Absaugtechniken & Schutzmaßnahmen

AG Oberflächenbeh., Beschichtung, Lackierung

UAG Roadmap OBL

### *... Cross Section Issues*

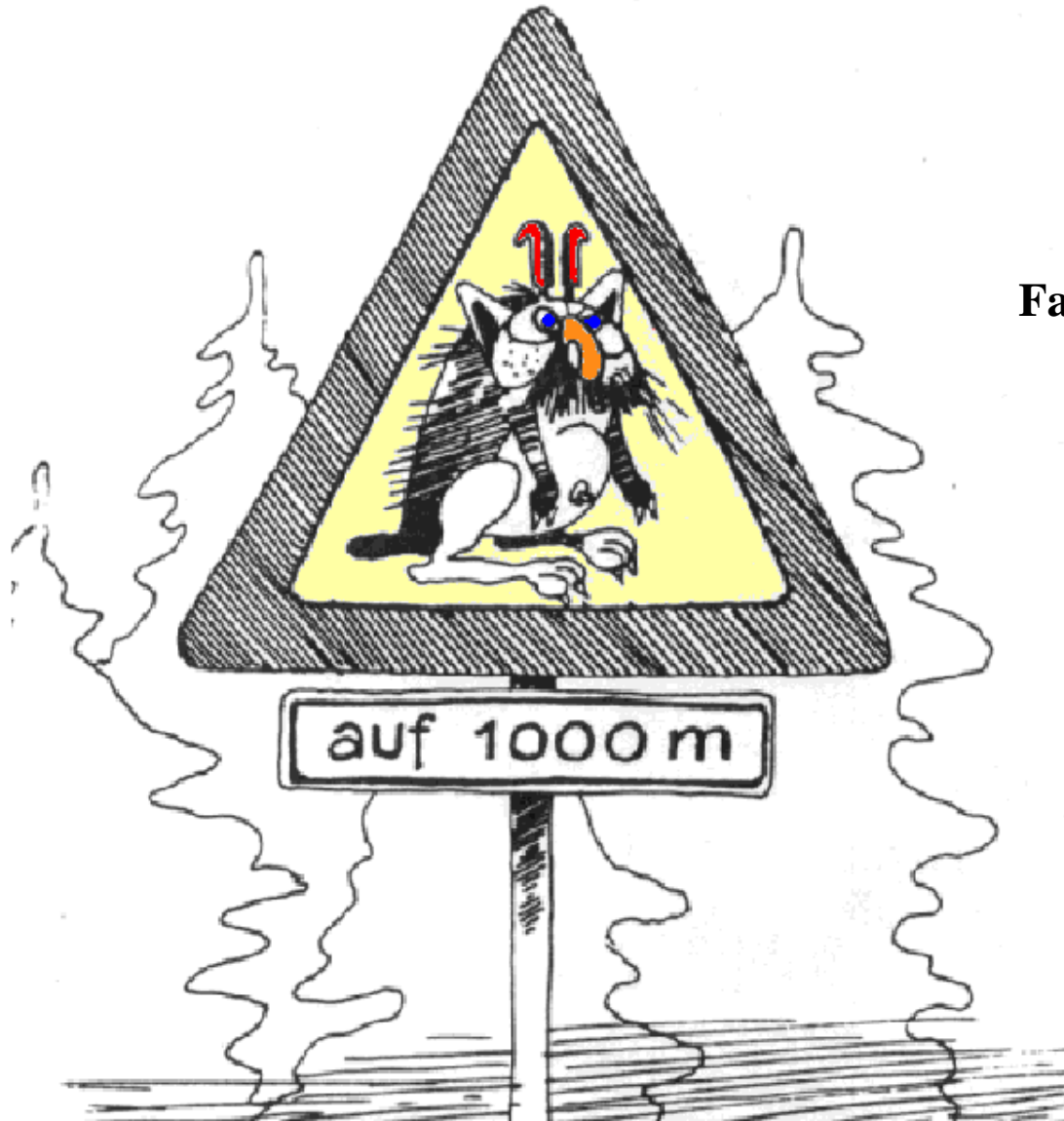
AG Kostenschätzung

AG Normung und Standardisierung

AG Roadmap CFK

AG Umweltaspekte

Why did we perform this in Bavaria, first ? Experience ? ..



Yes,  
**the Wolperdinger.**  
Famous Composite-Construction  
of the High-Tec Country  
*Freistaat Bayern*

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8. Model Parameters
9. Standardized Material Test Methods **Kollege**
10. Structural Testing, NDI, Damage Tolerance
11. Structural Verification, Margin of Safety, Reserve Factor

# Development Phases and Associated Topics

Phase		DESIGN	Design Analysis	Test
1	concept	conceptual	sizing	
2	design development	preliminary	dimensioning	design development tests
3		critical (final)	analytical design verification	
4	qualification	accepted		experimental design verification
5	production			


  
 certification of product

**Development:** Process phases from defining requirements until product delivery

**Designing:** Iterative process in the development of the structural component whereby various concepts are evolved and evaluated against a set of specified design requirements and constraints from manufacturing etc.

**Design Verification:** Process, whereby a structural design is comprehensively examined and qualification-tested to ensure that it will perform in the required way, before and during operational use.

# Some Definitions

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## **Safety Concept**

Concept that implements structural reliability (safety is a wrong term) in design

## **(design) Factor of Safety (FoS)**

Factor by which design limit loads (DLL) are multiplied in order to account for uncertainties of the verification methods, uncertainties in manufacturing process and material properties

## **Failure Modes (material, structural and others)**

Yield initiation, fracture, degradation, excessive wear, fibre fracture, inter fibre fracture, delamination, instability, or any other phenomenon resulting in an inability to sustain environmental 'loadings' (not only loads)

## **Service life of a Structural Component**

Starts with the manufacture of the structure and continues through all acceptance testing, handling, storage, transportation, operation, repair, re-testing, re-use

## **What is a Material ?**

= homogenized (smeared) model of the envisaged complex material which might be a material combination

## **What is Failure?**

If the structural part does not fulfil its functional requirements  
(FF = fiber failure, IFF = inter-fiber-failure (matrix failure, leakage, deformation limit, delamination size limit, ...))

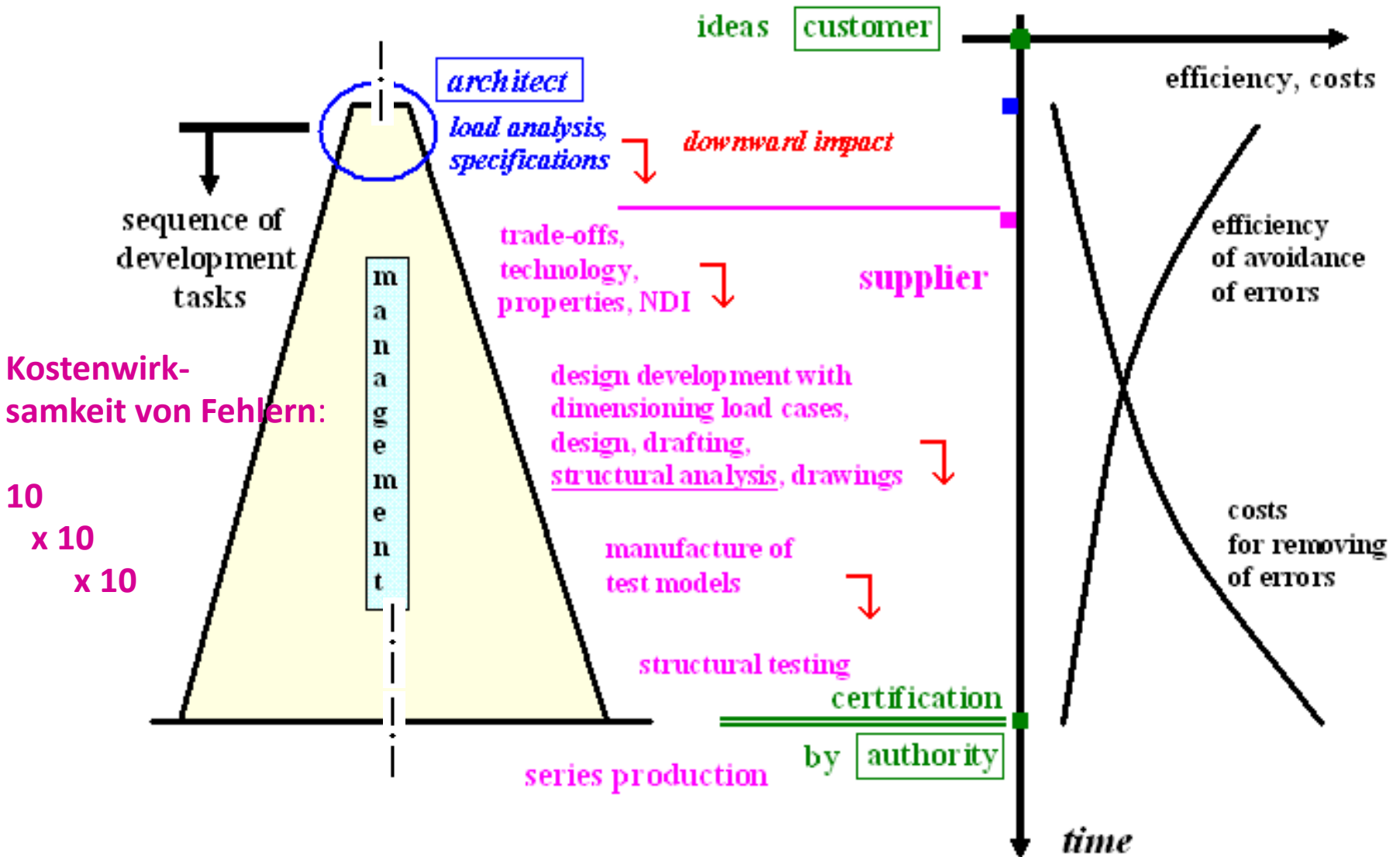
## **What is Fatigue ?**

= process, that degrades material properties



# Cost Penalty by Mistakes during Design Development Process Phases

Compromise: Cost → Minimum, Quality → Maximum



*Robust design helps to smooth out not-foreseen errors, to save cost & reduce troubles !*

# What does the stress engineer speak about ?

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Material: homogenized macromechanical model of the envisaged solid consisting of different constituents

Failure: structural part does not fulfil its functional requirements such as onset of yielding, onset of brittle fracture, Fiber-Failure FF, Inter-Fiber-Failure IFF, leakage, deformation limit, delamination size limit, frequency bound

= project-fixed Limit State with  $F$  = Limit State Function

Failure Criterion:  $F \geq 1$  , Failure Condition :  $F = 1 = 100\%$

$F$  = mathematical formulation of the failure surface (body)

Failure Theory: general tool to predict failure of a structural part,

*captures (1) Failure Conditions, (2) Non-linear Stress-strain Curves of a material as input, (3) Non-linear Coding for structural analysis*

Strength Failure Condition (SFC) = subset of a strength failure theory

tool for the assessment of a

'multi-axial failure stress state ' in a critical location of the material.



# Industrial Requirements for Improved Designing of Composite Parts

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## Static loading:

- Validated 3D strength failure conditions for isotropic (foam), transversely-isotropic UD materials, and orthotropic materials (e.g. textiles) to determine 'Onset of fracture' and 'Final fracture'
- Standardisation of material test procedures, test specimens, test rigs, and test data evaluation for the structural analysis input
- Consideration of manufacturing imperfections (tolerance width of uncertain design variables) in order to achieve a production cost minimum by „Design to Imperfections“ includes defects

## Cyclic (dynamic) loading : fatigue

- Development of practical, physically-based lifetime-prediction methods
- Generation of S-N curve test data for the verification of prediction models
- Delamination growth models: for duroplastic and thermoplastic matrices
- Consideration of media, temperature, creeping, aging
- Provision of more damping because parts become more monolithic.

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*for visualization some examples*

## Consideration of Load Assumptions:

z.B.



- Prüfvorschriften,
- Betriebs- und Mißbrauchslasten, Crash
- Fahrbetriebsmessung, Streckenmischung
- 1%-Fahrer, Lastkollektiv,
- Sicherheitsklasse des Bauteils,
- Unterschiede für : Pkw, LKW, Anhänger mit Kupplung, Dachlast, Motorrad

# Load Analysis

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## Main task is:

**Establishment of load events the structure is likely to experience (= load history)**

**Includes the estimation of all external + internal loadings of the structural component :**

- thermal,
- mechanical (static, cyclic, and dynamic) and
- acoustical environment as well as of the
- corresponding lifetime requirements (duration, number of cycles)

**Loadings are specified by**

**a Technical Specification from the customer, or an authority or a common standard (EN, DIN, Betonkalender, ...)**

*GROWIAN 1979,  
under-estimation  
of load peaks*

## Result:

**Set of Combinations of Loadings termed *Load Cases*, including the design driving *Dimensioning Load Cases***

*Involves a Worst case scenario wrt. combinations of loadings, temperature and moisture, and undetected damage.*

# Dimensioning Load Cases

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From the numerous *Load Cases*

the design driving *Dimensioning Load Cases (DimLC)* are to be sorted out:

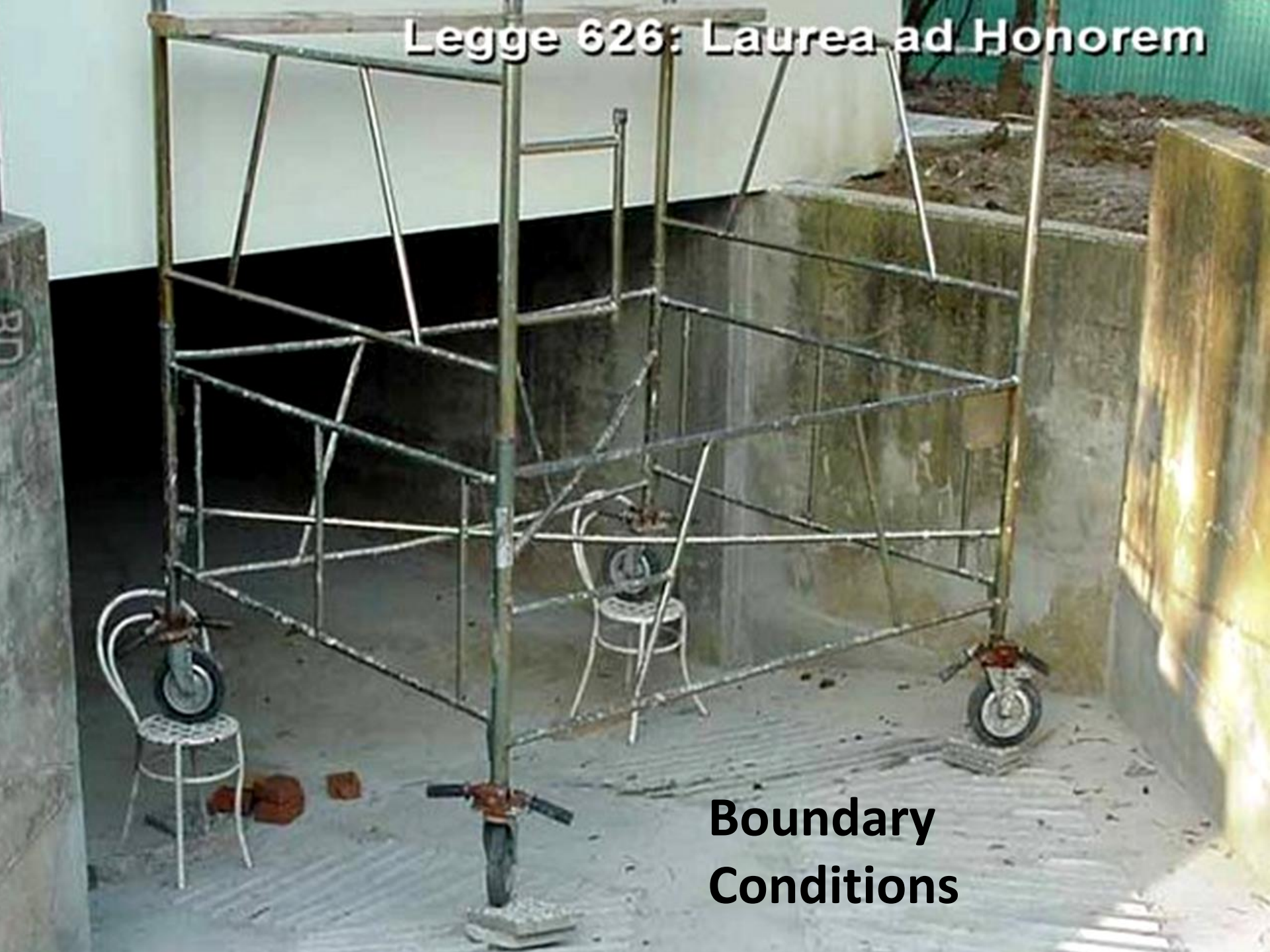
- for ductile behaviour the : Yielding-related Load Cases,
- for brittle behaviour the : Ultimate-related Load Cases (for CFRP).

**A minimum set of DimLCs is searched in order to:**

- support fast engineering decisions in cases of ‘input’ changes
- avoid analysis and analysis data evaluation overkill and
- better understand structural behaviour (as hidden aspect).

Which LC is a *DimLC* can be often firstly recognized after the analysis of the conceptual design !

**Legge 626: Laurea ad Honorem**



**Boundary  
Conditions**



# Example for a Factors of Safety (FOS) Table

*shows up higher risk than usual*

New Standard: prepared 10 years ago.

Structure type / sizing case	FOSY $j_{p0.2}$	FOSU $j_{ult}$	FOSY for verification 'by analysis only'	FOSU for verification 'by analysis only'	Design Factor	FOSY $j_{p0.2}$	FOSU $j_{ult}$	$j_{proof}$	$j_{burst}$
	external loadings incl. extern press.					internal pressure			
Metallic structures	1.1	1.25	1.25	1.5		1.0	1.0	1.2	1.5
FRP structures (uniform material)	?	1.25	-	1.5		1.0	1.0	1.?	1.5
FRP structures (discontinuities)	-	1.25	-	1.5	1.2				
Sandwich struct.:		1.25		1.5					
- Face wrinkling	-	1.25	-	1.5					
- Intracell buckl.		1.25		1.5					
- Honeycomb shear		1.25		1.5					
Glass/Ceramic structures	-	2.5	-	5.0					
Buckling	-	1.5	-	?					

**(ECSS-E-30-10, spacecraft)**

Term  $j_{p0.2}$  does not so much fit to actual (relatively brittle) composites!

## Note the Difference: *Test Data Mapping and Design Verification*

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- Validation of SFCs with Failure Test Data by mapping their course by an average Failure Curve (surface)

For each distinct Load Case with its single Failure Modes a RF must be computed:

- Delivery of a reliable Design Verification by calculation of a Margin of Safety or a (load) Reserve Factor

$$MoS > 0 \quad \text{oder} \quad RF = MoS + 1 > 1$$

on basis of a statistically reduced failure curve (surface) .

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# Features of Modeling laminated, high-performance Composites

- \* *Lamina-based, sub-laminate-based (e.g. for non-crimp fabrics) or laminate-based !*
- \* *Is performed, if applicable, according to the distinct symmetry of envisaged material*
- \* *For the chosen material model, if material symmetry-based, the number of the measured inherent Strengths and Elasticity Properties is the same as the observed number of Failure Modes !! Test costs reduction*
- \* **Achievement of equivalent stresses for each failure mode to obtain information where the lamina design screw must be turned !**

***Lesson-Learned: As far as the failure mode or failure mechanism remains,***

***Static Strength Criteria can be used for Cyclic Loading, too !***

**Very essential !**



# Modeling: 'Simple' UD material = Lamina (ply)

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1 Lamina = Layer of a Laminate, e.g. UD-laminas = "Bricks"

- *Homogenisation of a solid to a material brings benefits.*

- Then knowledge from Material Symmetry applicable :  
*number of required material properties is minimal, test-costs too*

UD-lamina, modeled a homogenised ('smeared') material requires in

The Material Characterisation  $f(Temp, Moisture, time, etc.)$

# Assumptions for UD Modelling and Mapping of Failure Stress data

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- The UD-lamina is macroscopically homogeneous.

It can be treated as a homogenized ('smeared') material

*Homogenisation of a solid to a material brings benefits.*

*Then Knowledge of Material Symmetry applicable : number of required material properties are minimal, test-costs too*

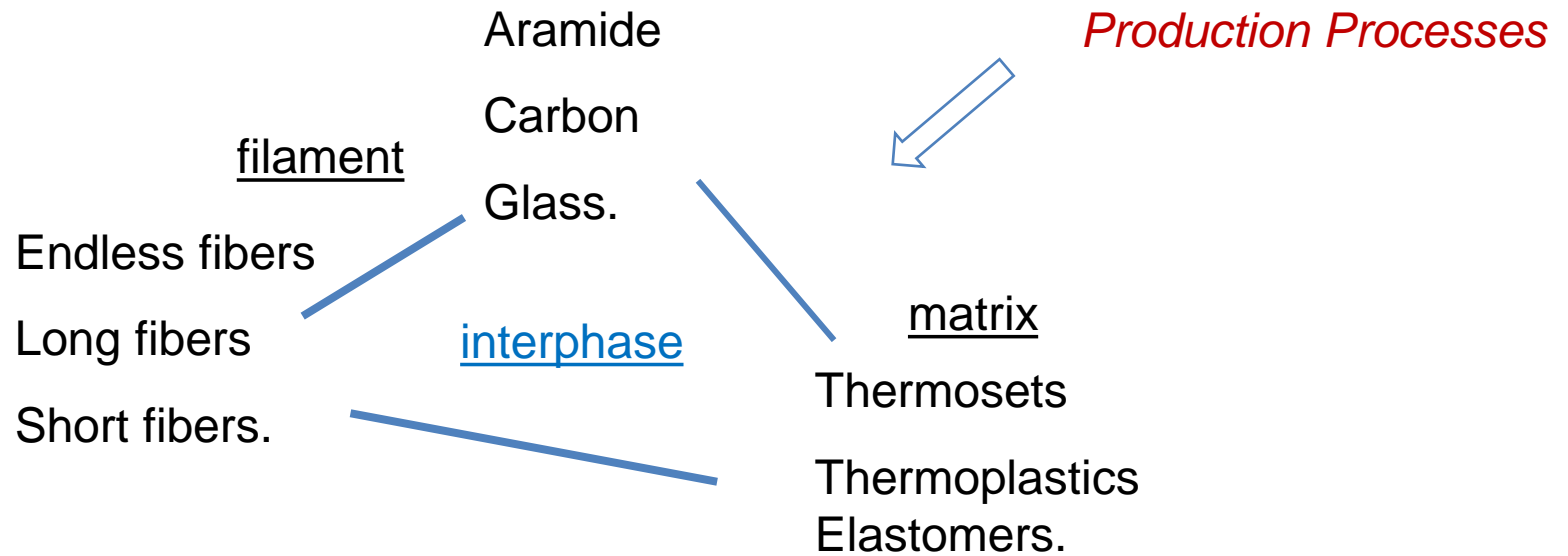
1 Lamina (ply) = Layer of a Laminate, e.g. UD-laminas = "Bricks"

- The UD-lamina is transversely-isotropic:

On planes, parallel to the fiber direction it behaves orthotropic and on planes transverse to fiber direction isotropic (quasi-isotropic plane)

- Mapping: Uniform stress states are about the critical stress location !

# Plenty combinations of different Constituents of polymeric Composites



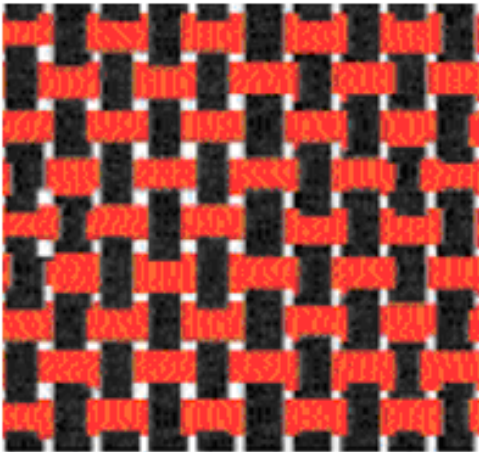
All these combinations

- need a different treatment and
- afford an associated understanding of its internal material behaviour.

... and - coming up more and more – an increasing variety of 2D- and 3D-fabrics

# Coming up: The Textile Challenge to achieve Certification

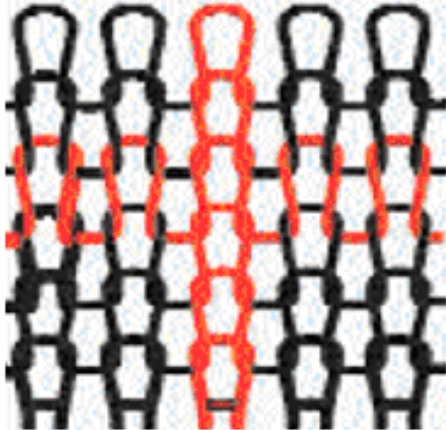
*UD is much simpler !*



**plain weave**



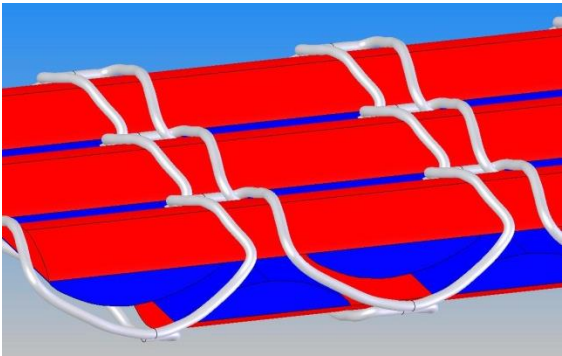
**braid**



**weft knit**



**warp knit**



non-crimp fabrics from UD-laminas for high-performance applications





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**Consequence for the poor Designer:** *To ask*

*Is there any Strength Failure Condition (“criterion“)  
he can apply with high fidelity?*



*Not at all !*

# Some well-known Developers which formulated isotropic **3D** Strength Failure Conditions (SFCs)

Willam-Warnke,  
Ottosen etc.

Hencky-  
Mises-  
Huber



Richard von Mises  
1883-1953  
*Mathematician*



Eugenio Beltrami  
1835-1900  
*Mathematician*



Otto Mohr  
1835-1918  
*Civil Engineer*



Charles de Coulomb  
1736-1806  
*Physician*

**'Onset of Yielding'**

**'Onset of Cracking'**

Hence again, a *civil engineer* may proceed



## State of the Art: Static Strength Analysis of UD laminas

Is represented best by the results of the *World-Wide-Failure-Exercises*

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Organizer : **QinetiQ, UK** (*Hinton, Kaddour, Soden, Smith, Shuguang Li*)

Aim: *'Testing Predictive Failure Theories for*

*Fiber-Reinforced Polymer Composites to the full !'*

*(was for the transversely-isotropic UD materials , only)*

**Method of the World-Wide-Failure-Exercises** (since 1991):

Part A of a WWFE: ***Blind Predictions on basic strength data***

Part B of a WWFE: ***Comparison Theory-Test with (reliable )  
Uni-axial 'Failure Stress Test Data' (= basic strength) and  
Multi-axial 'Failure Stress Test Data'***

(plain test specimens, no notch)

*Drucker-Prager, Tsai-Wu*

1 Global strength failure condition :  $F(\{\sigma\}, \{R\}) = 1$  (usual formulation)

Set of Modal strength failure conditions:  $F(\{\sigma\}, R^{mode}) = 1$  (addressed in FMC)

*Mises, Puck, Cuntze*

Example: UD vector of 6 stresses (general)

$$\{\sigma\} = (\sigma_1, \sigma_2, \sigma_3, \tau_{23}, \tau_{31}, \tau_{21})^T$$

vector of 5 strengths

$$\{R\} = (R_{\parallel}^t, R_{\parallel}^c, R_{\perp}^t, R_{\perp}^c, R_{\perp\parallel})^T$$

needs an Interaction of Failure Modes: performed by a

*probabilistic-based 'rounding-off' approach (series failure system model)  
directly delivering the (material) reserve factor in linear analysis*

Note: In the quasi-isotropic plane of the UD material just 5 stresses are active:

$$\{\sigma\}_{principal}^{quasi-isotropic\ plane} = (\sigma_1, \sigma_2^p, \sigma_3^p, 0, \tau_{31}^p, \tau_{21}^p)^T$$

By-the-way: Experience with Failure Prediction prove

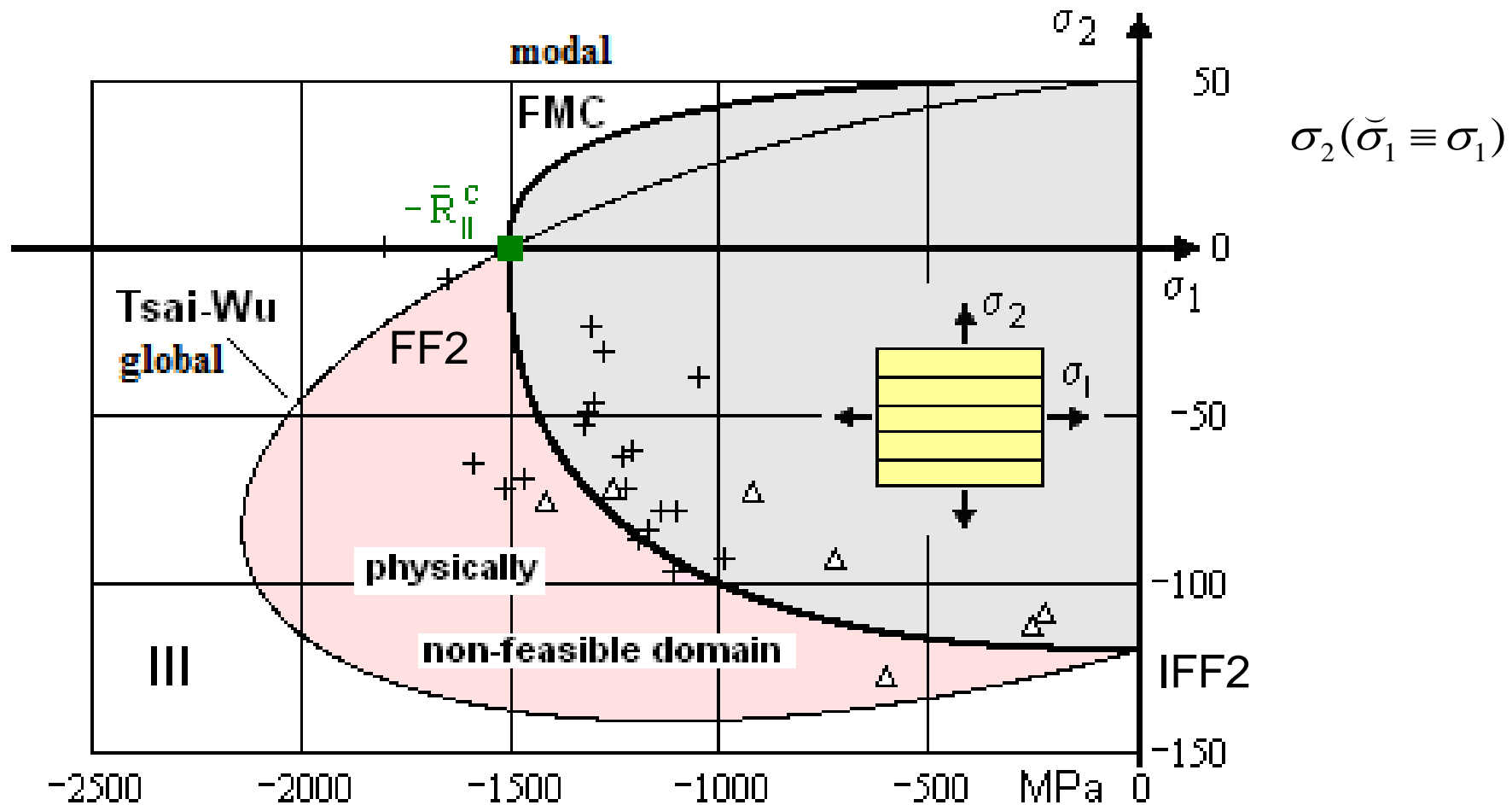
A Strength Failure Condition (SFC) is a necessary but not a sufficient condition to predict Strength Failure (example: thin-layer problem).

On top, an energy condition may be to fulfill.

## Modal SFCs (multi-surface domains)

- Describe one single failure mode in one single mathematical formulation (= one part of the failure surface)
  - \* determine all mode model parameters in the respective failure mode domain
  - \* capture a twofold acting failure mode separately, such as  $\sigma_I = \sigma_{III}$  (isotropic) or  $\sigma_2 = \sigma_3$  (transversely-isotropic UD material), mode-wise by the well-known Ansatz  $f(J_2, J_3)$
- Re-calculation of the model parameters and of RF just in that failure mode domain where test data must be replaced.

# Mapping in the 'Tsai-Wu non-feasible domain' (quadrant III)



Data: courtesy IKV Aachen, Knops

Lesson Learnt: The modal FMC maps correctly, the *global* Tsai-Wu formulation predicts in quadrant III a non-feasible domain !

# Test-observed Strength Failure Modes of Brittle behaving Isotropic Materials

## Normal Fracture (NF)

- no material element change before fracture

## Shear Fracture Mode (SF)

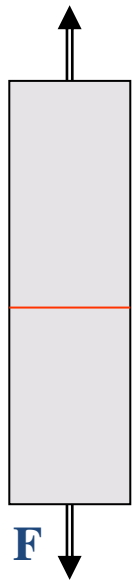
- shape change of material element

## Crushing Fracture (CrF):

- volumetric element change before fracture  
*helpful knowledge for the later choice of invariants*

### Tension

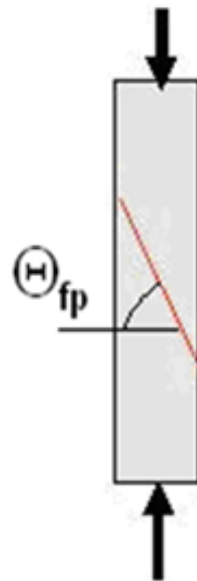
*if brittle: failure = fracture failure*



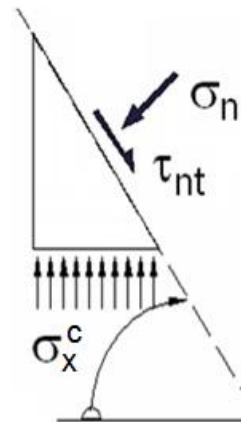
$$R^t = f_t$$

### Compression

dense consistency



$$R^c \equiv f_c$$



z. B.  $\theta_{fp} = 54^\circ$

fracture plane angle = measure for friction value

porous consistency



= decomposition of texture



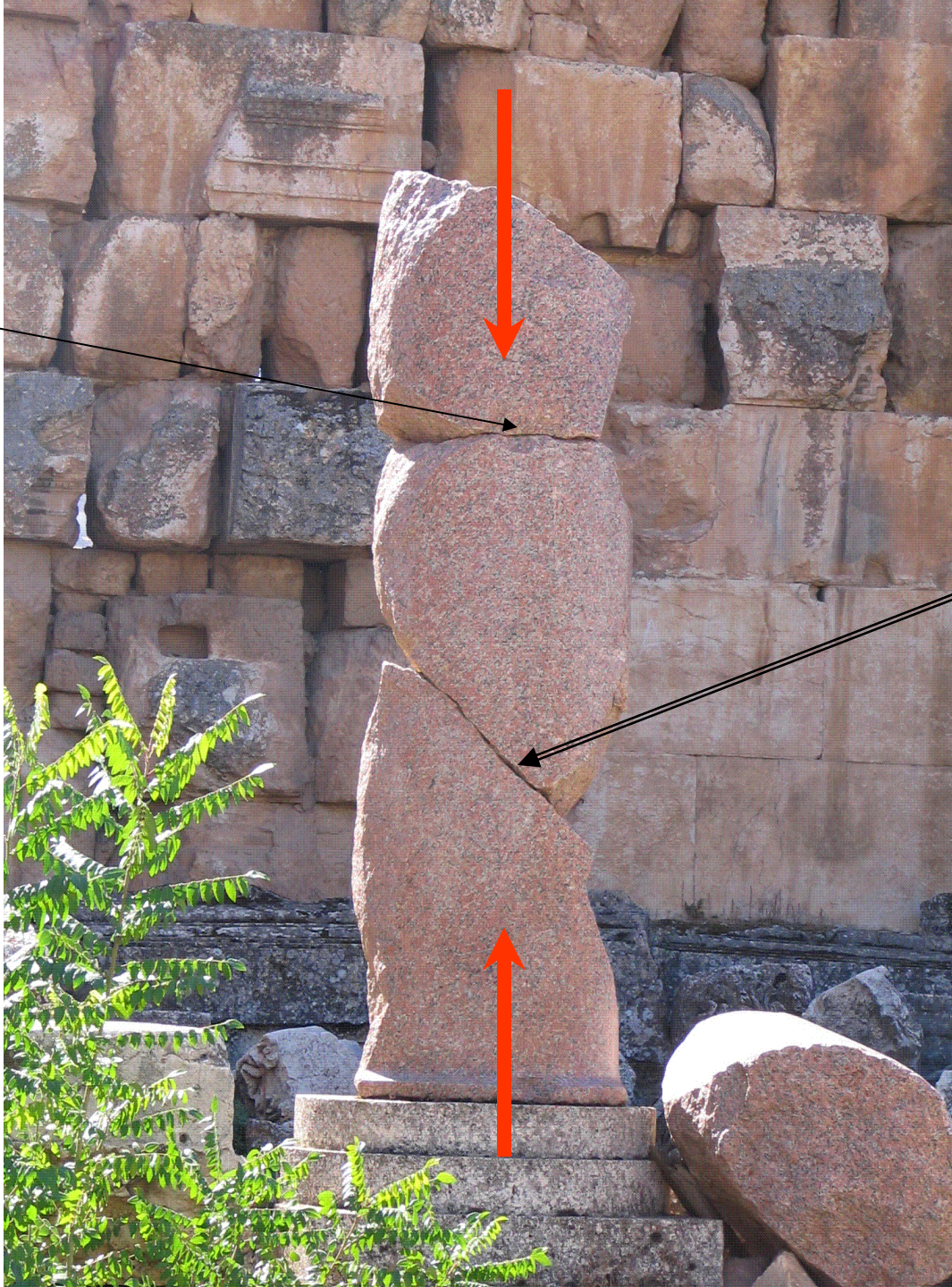
= hill of fragments (crumbs) as result of compression tests

t = tension  
 c = compression  
 R = strength, resistance  
 F = Fracture

**Observed: ▶ Each single Failure Mode is governed by one single strength, only !!**



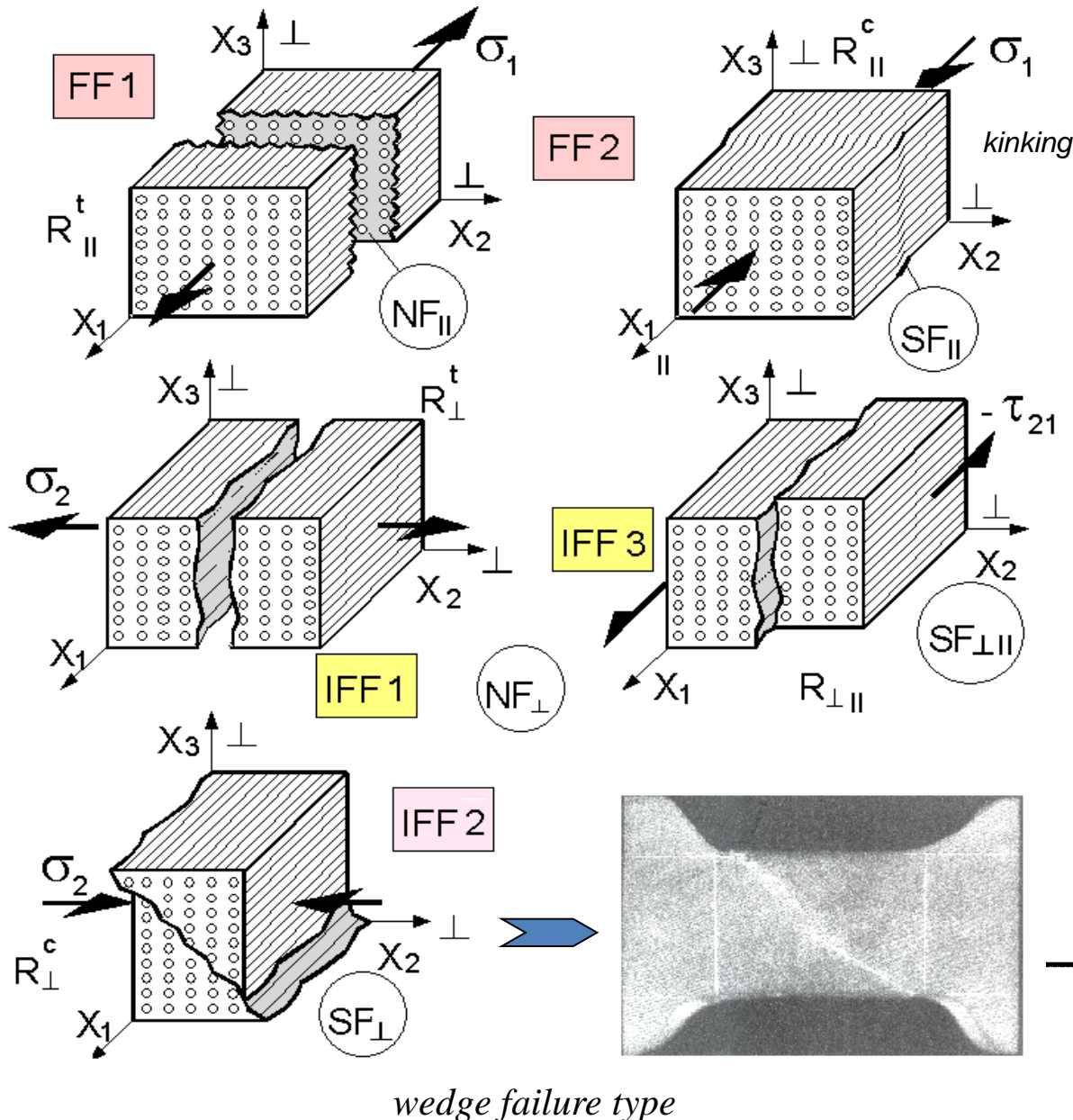
just a  
joint



Example SF :  $R_m^c$   
Shear Fracture plane  
under compression

(Mohr-Coulomb, acting *at a  
rock material column,  
at Baalbek, Libanon*)

# Test-observed Strength Failure Modes of Brittle behaving UD-Materials



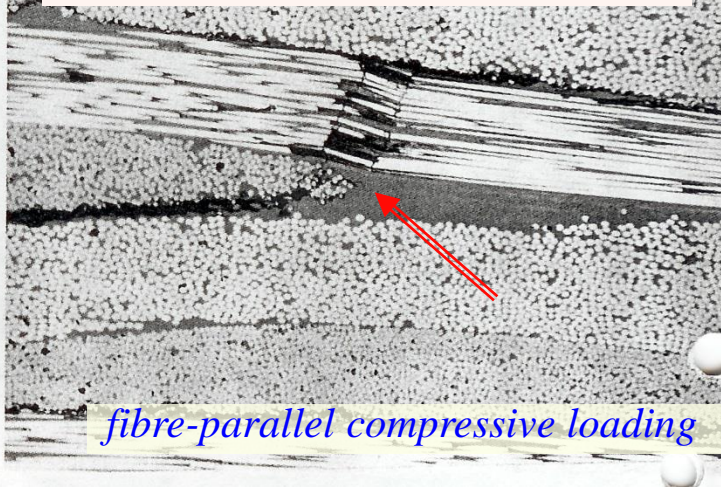
t = tension  
c = compression

- 5 Fracture modes exist
- = 2 FF (Fibre Failure)
- + 3 IFF (Inter Fibre Failure)

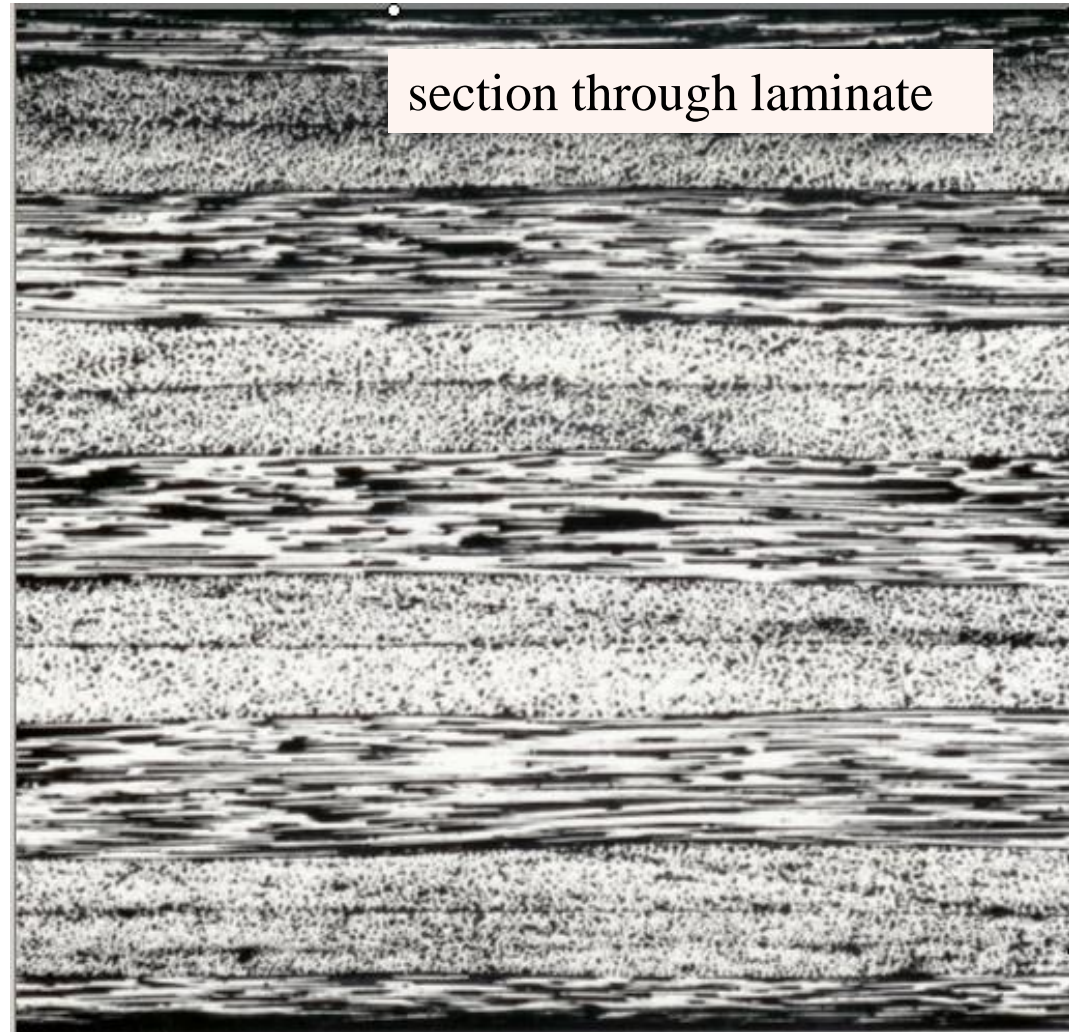
Fracture Types  
(macroscale-associated):  
NF := Normal Fracture SF  
:= Shear Fracture

Friction occurs in  
IFF2 and IFF3 !

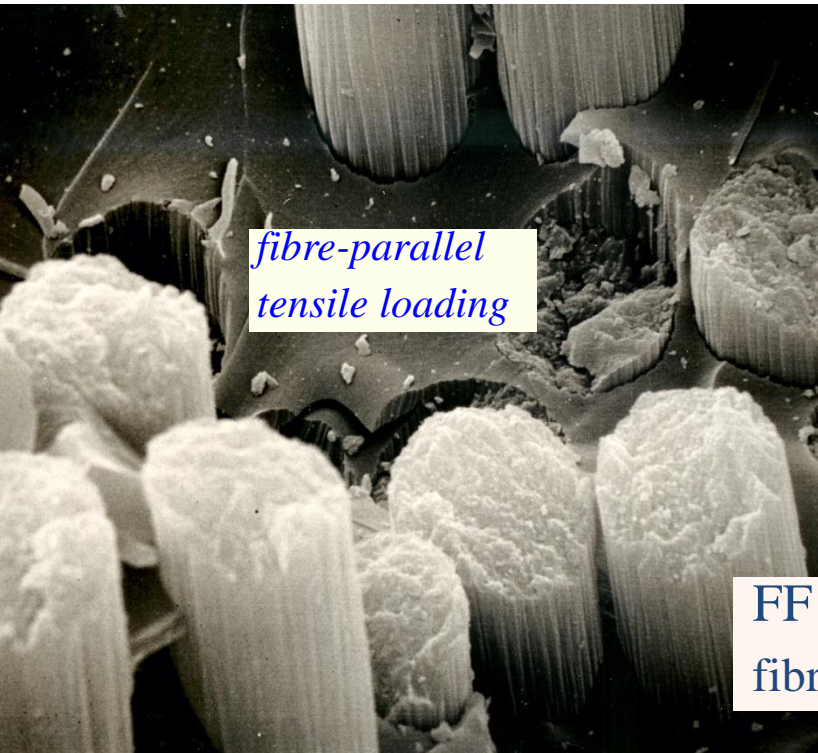
FF2 compressive fibre fracture = kinking causes onset of delamination



*fibre-parallel compressive loading*



section through laminate



*fibre-parallel tensile loading*

FF1 tensile fibre fracture

Fractography pictures as proofs

# Cuntzes 3D Modal Strength Failure Cond. (criteria) for Isotropic Foams

**Approaches:**  $\frac{\sqrt{4J_2 - I_1^2/3} + I_1}{2 \cdot \bar{R}_t} = 1$   $\frac{\sqrt{4J_2 - I_1^2/3} - I_1}{2 \cdot \bar{R}_c} = 1$

**Considering bi-axial strength (failure mode occurs twice):** in Effs now

$$Eff^{NF} = c_{NF} \cdot \frac{\sqrt{4J_2 - I_1^2 \cdot (\Theta_{NF})/3} + I_1}{2 \cdot \bar{R}_t} = \underline{\underline{\sigma_{eq}^{NF}}} / \bar{R}_t, \quad Eff^{CrF} = c_{CrF} \cdot \frac{\sqrt{4J_2 - I_1^2 \cdot (\Theta_{CrF})/3} - I_1}{2 \cdot \bar{R}_c} = \sigma_{eq}^{CrF} / \bar{R}_c$$

The two-fold failure danger can be excellently modelled by using the often used invariant **J3** in :

$$\Theta_{NF} = \sqrt[3]{1 + D_{NF} \cdot \sin(3\theta)} = \sqrt[3]{1 + D_{NF} \cdot 1.5 \cdot \sqrt{3} \cdot J_3 \cdot J_2^{-1.5}} \quad \Theta_{CrF} = \sqrt[3]{1 + D_{CrF} \cdot \sin(3\theta)} = \sqrt[3]{1 + D_{CrF} \cdot 1.5 \cdot \sqrt{3} \cdot J_3 \cdot J_2^{-1.5}}$$

**Mode interaction:**  $Eff^{NF} = [(Eff^{NF})^m + (Eff^{CrF})^m]^{m^{-1}}$

**The failure surface is closed at both the ends:** A simple cone serves as closing cap and bottom

$$\frac{I_1}{\sqrt{3} \cdot R_t} = s_{NF} \cdot \left( \frac{\sqrt{2J_2 \cdot \Theta_{NF}}}{R_t} \right) + \frac{\max I_1}{\sqrt{3} \cdot R_t} \quad \text{Rt-normalized Lode-Coordinates} \quad \frac{I_1}{\sqrt{3} \cdot R_t} = s_{CrF} \cdot \left( \frac{\sqrt{2J_2 \cdot \Theta_{CrF}}}{R_t} \right) + \frac{\min I_1}{\sqrt{3} \cdot R_t}$$

The slope parameters  $s$  are determined connecting the respective hydrostatic strength point with the associated point on the shear meridian,  $\max I_1$  must be assessed whereas  $\min I_1$  could be measured.

# Cuntzes 3D Modal SFCs (criteria) for Transversely-Isotropic UD-materials

Invariants replaced by their stress formulations

FF1	$Eff^{\parallel\sigma} = \check{\sigma}_1 / \bar{R}_{\parallel}^t = \sigma_{eq}^{\parallel\sigma} / \bar{R}_{\parallel}^t,$	$\check{\sigma}_1 \cong \varepsilon_1^t \cdot E_{\parallel}^*$	strains from FEA	[Cun04, Cun11]
FF2	$Eff^{\parallel\tau} = -\check{\sigma}_1 / \bar{R}_{\parallel}^c = +\sigma_{eq}^{\parallel\tau} / \bar{R}_{\parallel}^c,$	$\check{\sigma}_1 \cong \varepsilon_1^c \cdot E_{\parallel}$		<b>2 filament modes</b>
IFF1	$Eff^{\perp\sigma} = [(\sigma_2 + \sigma_3) + \sqrt{(\sigma_2 - \sigma_3)^2 + 4\tau_{23}^2}] / 2\bar{R}_{\perp}^t = \sigma_{eq}^{\perp\sigma} / \bar{R}_{\perp}^t$			<b>3 matrix modes</b>
IFF2	$Eff^{\perp\tau} = [(\frac{\mu_{\perp\perp}}{1-\mu_{\perp\perp}}) \cdot (\sigma_2 + \sigma_3) + \frac{1}{1-\mu_{\perp\perp}} \sqrt{(\sigma_2 - \sigma_3)^2 + 4\tau_{23}^2}] / \bar{R}_{\perp}^c = +\sigma_{eq}^{\perp\tau} / \bar{R}_{\perp}^c$			<b>3 matrix modes</b>
IFF3	$Eff^{\perp\parallel} = \{ [\mu_{\perp\parallel} \cdot I_{23-5} + (\sqrt{\mu_{\perp\parallel}^2 \cdot I_{23-5}^2 + 4 \cdot \bar{R}_{\perp\parallel}^2 \cdot (\tau_{31}^2 + \tau_{21}^2)}) / (2 \cdot \bar{R}_{\perp\parallel}^3) \}^{0.5} = \sigma_{eq}^{\perp\parallel} / \bar{R}_{\perp\parallel}$			
	with $I_{23-5} = 2\sigma_2 \cdot \tau_{21}^2 + 2\sigma_3 \cdot \tau_{31}^2 + 4\tau_{23}\tau_{31}\tau_{21}$			

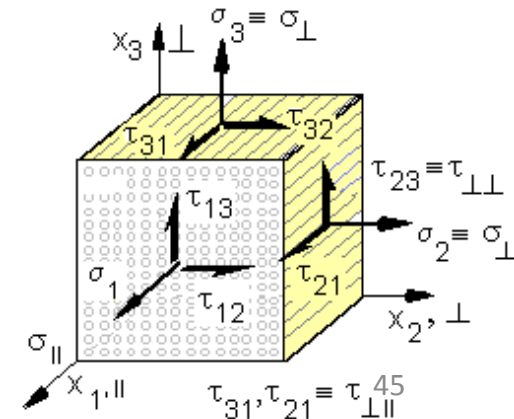
Modes-Interaction :

$$Eff^m = (Eff^{\parallel\tau})^m + (Eff^{\parallel\sigma})^m + (Eff^{\perp\sigma})^m + (Eff^{\perp\tau})^m + (Eff^{\perp\parallel})^m = 1$$

with mode-interaction exponent  $2.5 < m < 3$  from mapping tests data

Typical friction value data range:  $0.05 < \mu_{\perp\parallel} < 0.3, 0.05 < \mu_{\perp\perp} < 0.2$

Poisson effect \* : bi-axial compression strains the filament without any  $\sigma_1$   
 t:= tensile, c:= compression, || := parallel to fibre, ⊥ := transversal to fibre



# Interaction of Single Strength Failure Modes in the modal FMC

---

Interaction of adjacent Failure Modes by a *series failure system* model

= 'Accumulation' of interacting *failure danger portions*  $Eff^{mode}$

$$Eff^* = \sqrt[m]{(Eff^{mode\ 1})^m + (Eff^{mode\ 2})^m + \dots} = 1 = 100\%, \text{ if failure}$$

with mode-interaction exponent  $2.5 < m < 3$  from mapping experience

as *modal material stressing effort* \* (in German Werkstoffanstrengung)

and

$$Eff^{mode} = \sigma_{eq}^{mode} / \bar{R}^{mode}$$

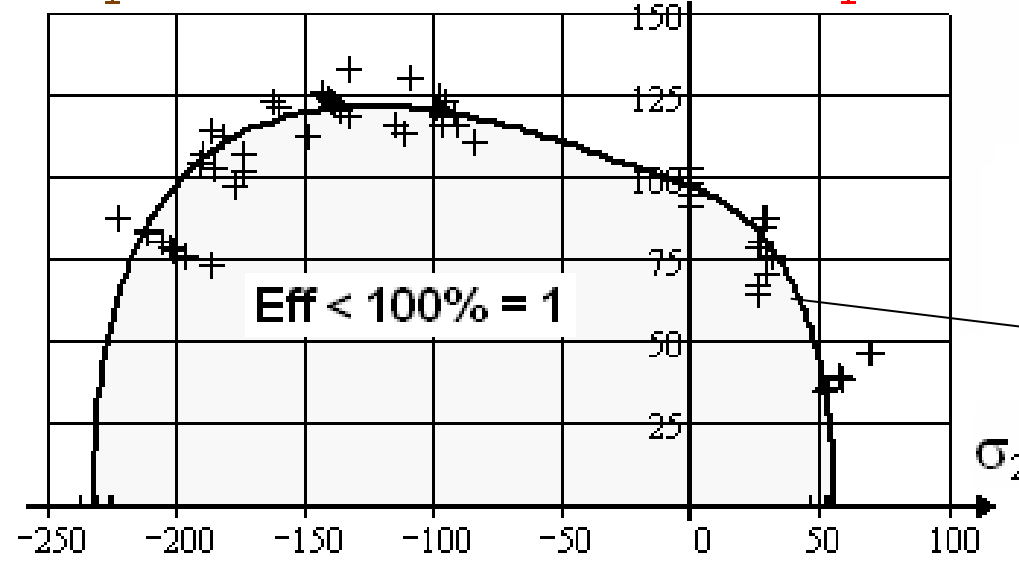
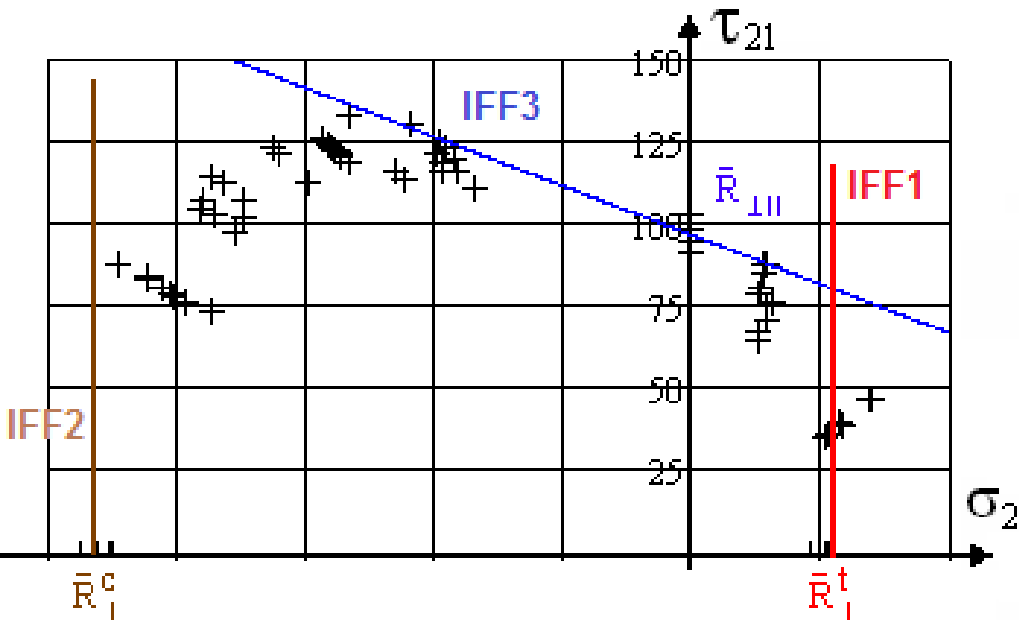
equivalent mode stress

mode associated average strength

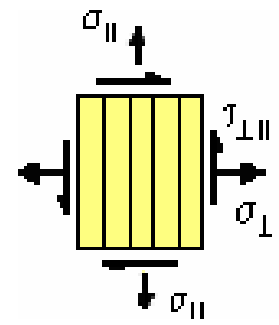
example

\* **artificial technical term** created together with QinetiQ in the World-Wide-Failure-Exercise

# 2D-Demonstration: Interaction of UD Failure Modes for $\tau_{21}(\sigma_2)$ , $\bar{\sigma}_1 = 0$



Mapping of course of IFF test data in a pure mode domain by the associated Mode Failure Condition.  
 3 IFF pure modes = straight lines !



$$IFF 1: \frac{\sigma_2}{\bar{R}_1^t} = 1$$

$$IFF 2: \frac{-\sigma_2}{\bar{R}_1^c} = 1$$

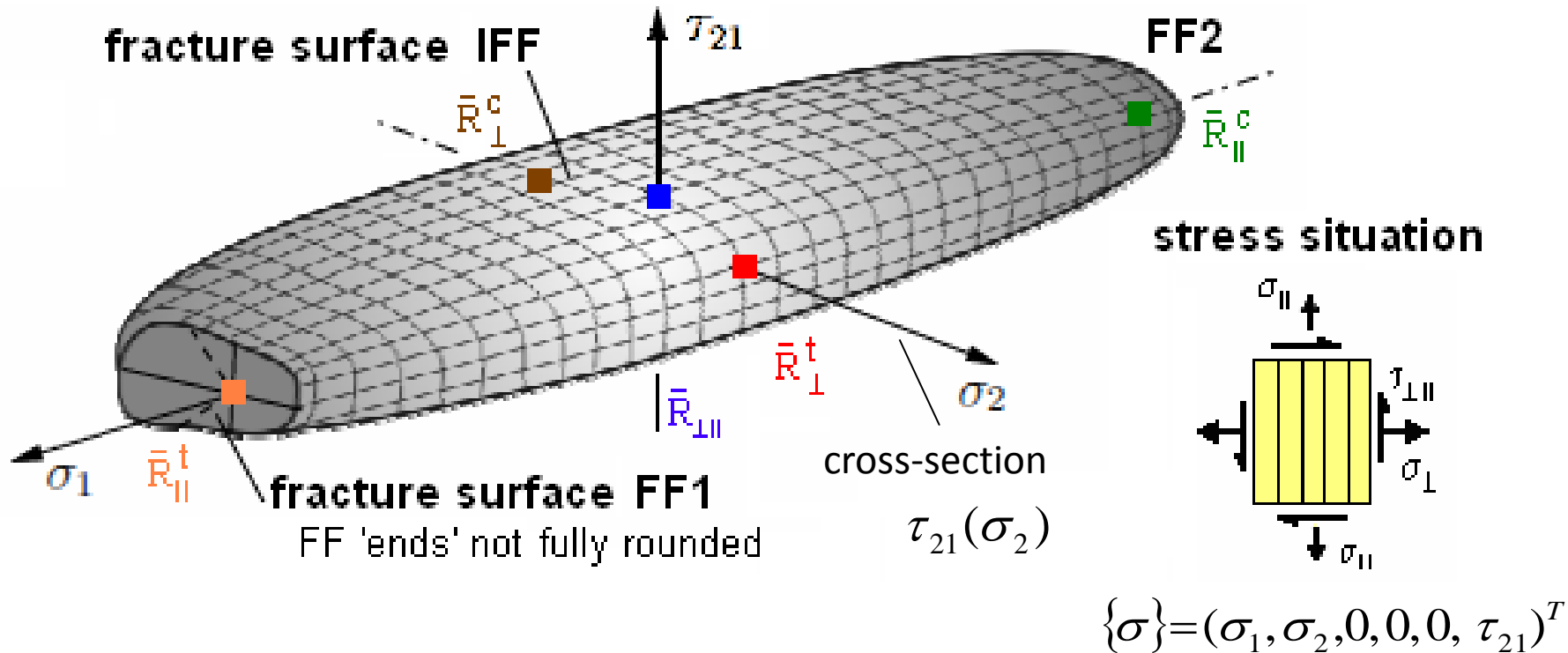
$$IFF 3 \text{ (2D-simplified): } \frac{|\tau_{21}|}{\bar{R}_{\perp\parallel} - \mu_{\perp\parallel} \cdot \sigma_2} = 1$$

Mapping of course of test data by Interaction Model

$$(Eff^{\perp\sigma})^m + (Eff^{\perp\tau})^m + (Eff^{\perp\parallel})^m = 1$$

$$m = 2.5, \mu_{\perp\parallel} = 0.3$$

# Visualization of 2D-UD-SFCs as Fracture Failure Surface (Body)



Mode interaction fracture failure surface of *FRP UD lamina*

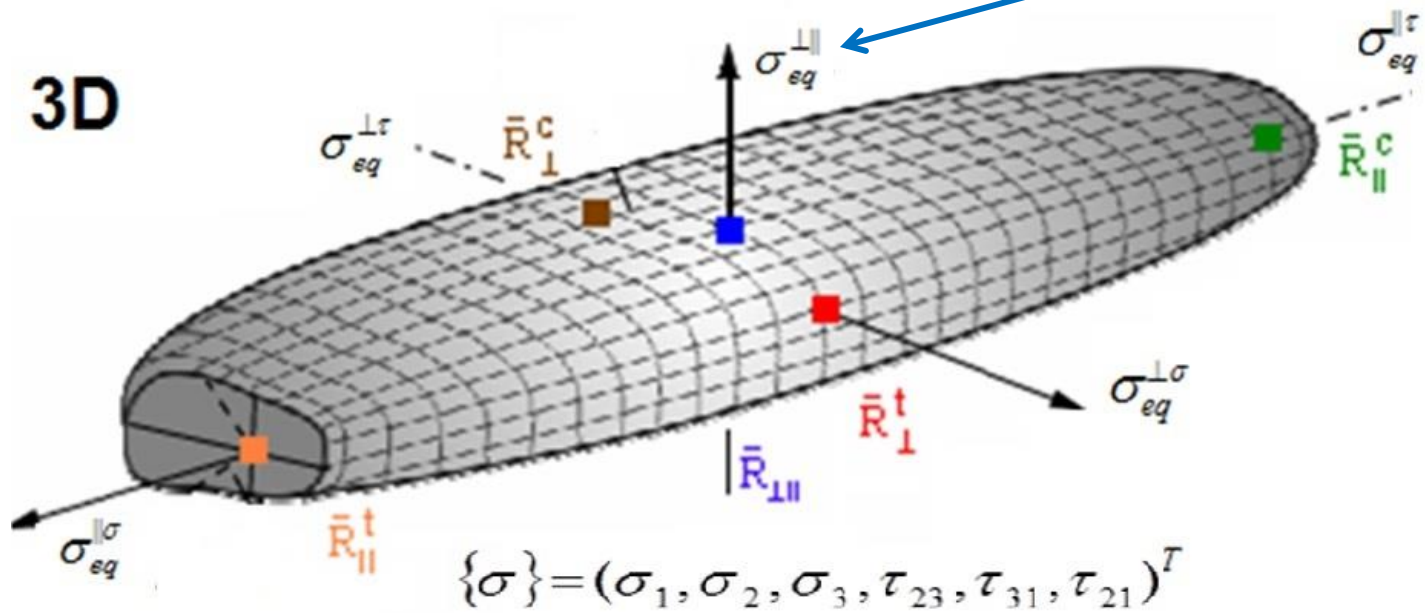
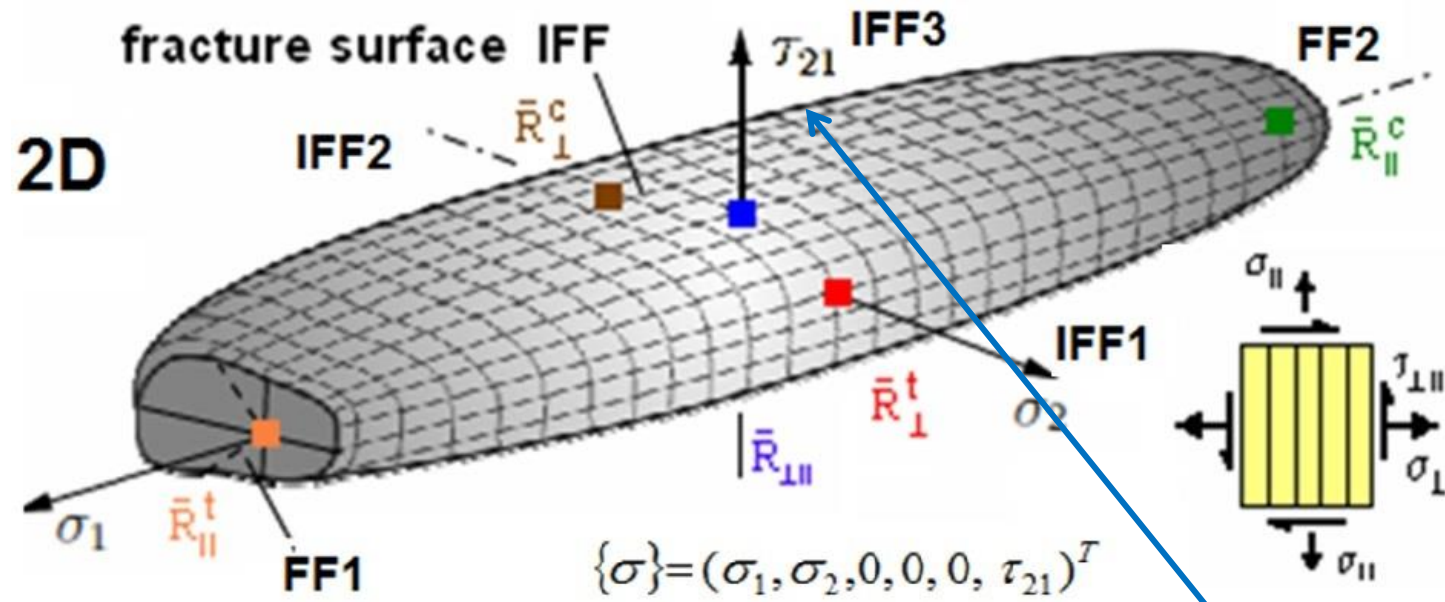
$$Eff^m = (Eff^{\parallel\tau})^m + (Eff^{\parallel\sigma})^m + (Eff^{\perp\sigma})^m + (Eff^{\perp\tau})^m + (Eff^{\perp\parallel})^m = 1$$

(courtesy W. Becker).

Mapping: Average strengths indicated



2D = 3D Fracture surface if replacing *stresses* by *equivalent stresses*



## **Short Presentation of CCEV + personal activities**

- 1. Structural Development, Design Requirements, and Design Verifications**
- 2. Dimensioning Load cases, Safety Concept and Design Factors of Safety**
- 3. Modelling of Composites (elasticity, strength)**
- 4. Material Strength Failure Conditions (SFC)**
- 5. Application of SFCs to Some Materials**
- 6. Lifetime Prediction**
- 7. Material Properties**
- 8. Model Parameters**
- 9. Standardized Material Test Methods**
- 10. Structural Testing, NDI, Damage Tolerance**
- 11. Structural Verification, Margin of Safety, Reserve Factor**

# Specifica for the UD-lamina-based High Performance Laminates

---

Specific Pre-requisites for the establishment of 3D-UD-SFCs:

- simply formulated from engineering point of view, numerically robust,
- physically-based, and therefore need only few information for pre-dimensioning
- shall allow for a simple determination of the design driving reserve factor
- shall capture failure of the constituents matrix (cohesive), interphase (adhesive), filament

- consider residual stresses

*Compliant with John Hart-Smith*

- consider micro-mechanical stress concentration of the matrix around the filaments under transversal stress (a means: using matrices showing > 6% fracture strain which helps to capture a stress concentration factor of about 6 up to 1% applied transversal strain)
- consider FF, if taking place under bi-axial compression with no external axial stress

$$\{\sigma\} = (\sigma_1 = 0, \sigma_2, \sigma_3, 0, 0, 0)^T$$

# Example: Assumptions for UD Modelling and Mapping

---

- The UD-lamina is macroscopically homogeneous. It can be treated as a homogenized ('smeared') material *Homogenisation of a solid to a material brings benefits.*

*Then Knowledge of Material Symmetry applicable : number of required material properties are minimal, test-costs too*

1 Lamina (ply) = Layer of a Laminate, e.g. UD-laminas = "Bricks"

- The UD-lamina is transversely-isotropic: On planes, parallel to the fiber direction it behaves orthotropic and on planes transverse to fiber direction isotropic (quasi-isotropic plane)
- Mapping creates fidelity, only, if: **uniform stress states are about the critical stress location in the material !** Is very seldom the case.

# Motivation for my non-funded Investigations

---

Existing Links in the Mechanical Behaviour show up: *Different structural materials*

- *can possess similar material behaviour* or
  - *can belong to the same class of material symmetry*
- > similarity aspect

Welcomed Consequence:

- The same strength failure function  $F$  can be used for different materials
- More information is available for pre-dimensioning + modelling  
from experimental results of a similarly behaving material.

# Basic Features of the author's Failure-Mode-Concept (FMC)

---

- Each failure mode represents 1 independent failure mechanism and thereby 1 piece of the complete *failure surface*
  - Each failure mechanism is governed by 1 basic strength (is observed !)
  - Each failure *mode* can be represented by 1 failure *condition*. *Therefore,*  
*equivalent stresses can be computed for each mode !*
- 

- In consequence, this separation requires :

*An interaction of the Modal Failure Modes !*

## Failure-Mode-Concept (FMC) Postulate (*example: UD material*)

---

Remember:

- Each single observed fracture failure modes is linked to one strength
- Symmetry of a material showed : *Number of strengths* =  $R_{//}^t, R_{//}^c, R_{\perp//}, R_{\perp}^t, R_{\perp}^c$   
*number of elasticity properties!*  $E_{//}, E_{\perp}, G_{\perp//}, \nu_{\perp//}, \nu_{\perp\perp}$

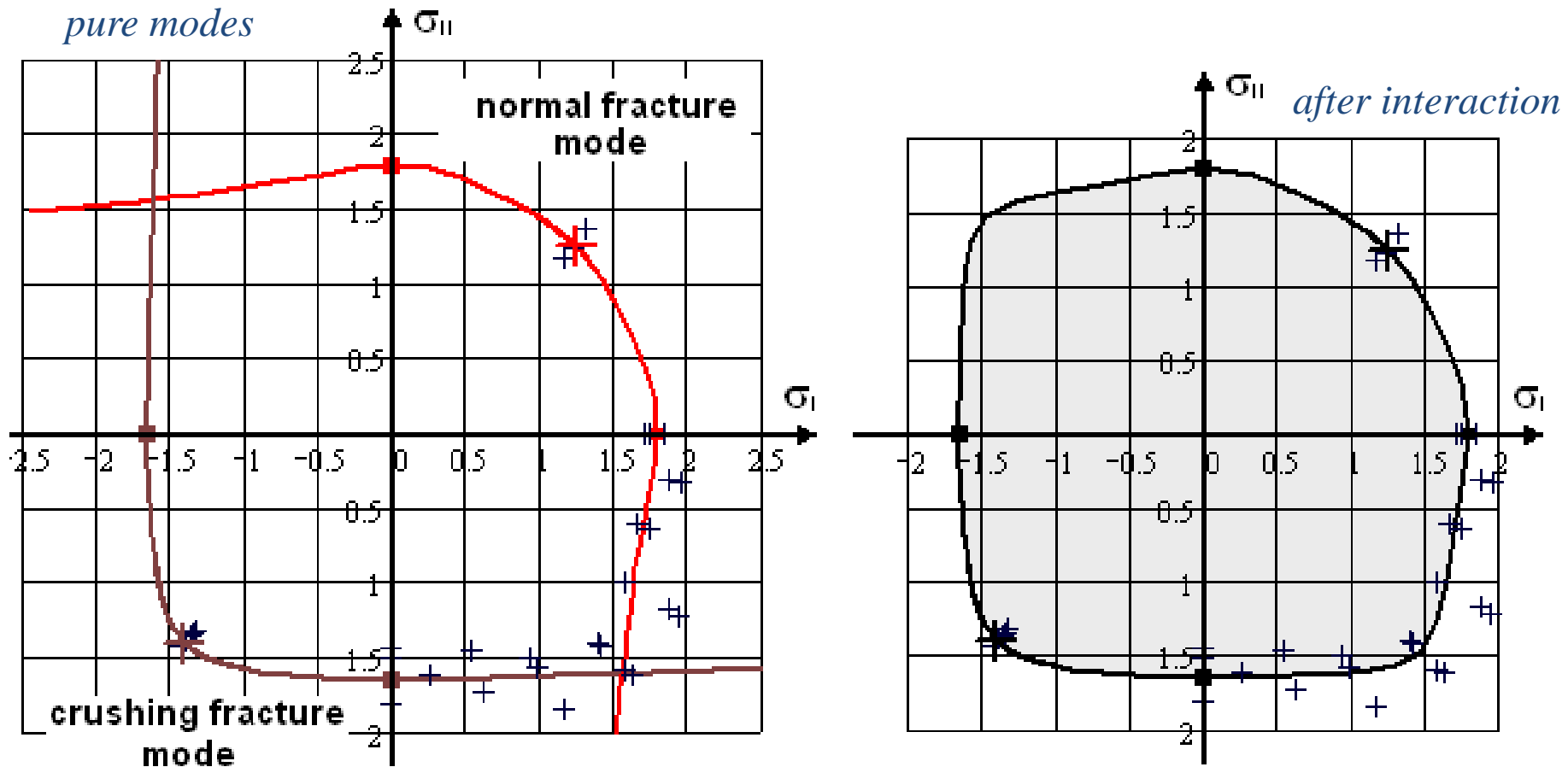
Due to the facts above Cuntze postulates in his FMC

▶ Number of failure modes = number of strengths, too !  
e.g.: isotropic = 2 or above transversely-isotropic (UD) = 5

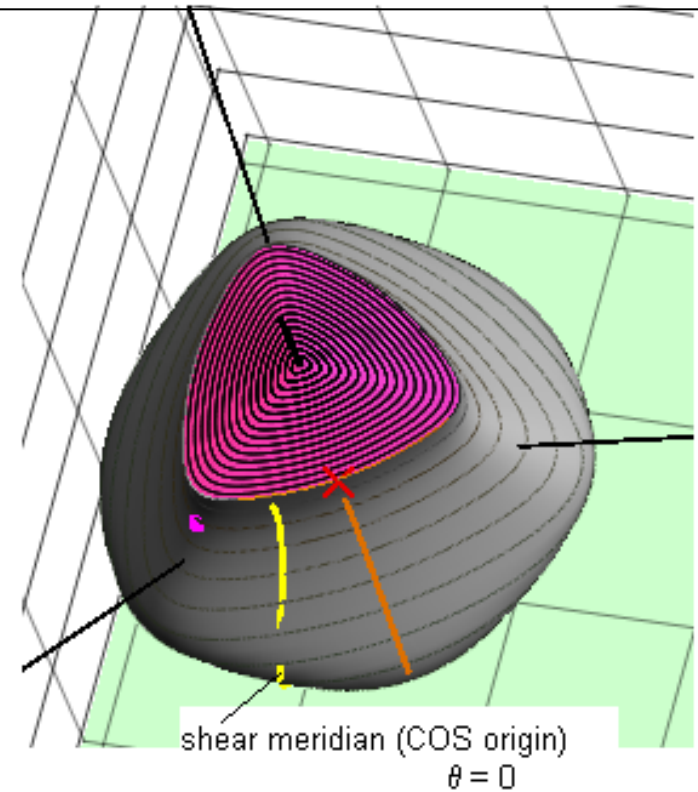
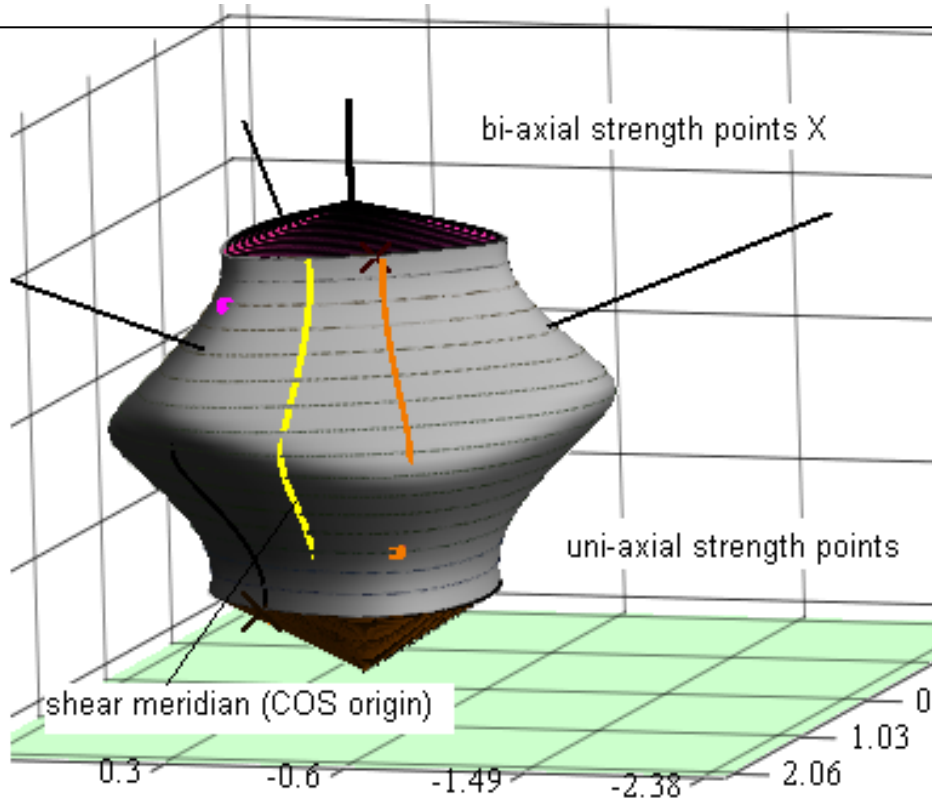
- 1 Introduction to Strength Failure Conditions (SFCs)
  - 2 Fundamentals when generating SFCs (criteria)
  - 3 Global SFCs versus Modal SFCs
  - 4 Requirements
  - 5 Short Derivation of the Failure-Mode-Concept (FMC)
  - 6 **FMC-model applied to an Isotropic Foam (Rohacell 71 G)**
  - 7 FMC-model applied to a transversely-isotropic UD-CFRP
- Conclusions



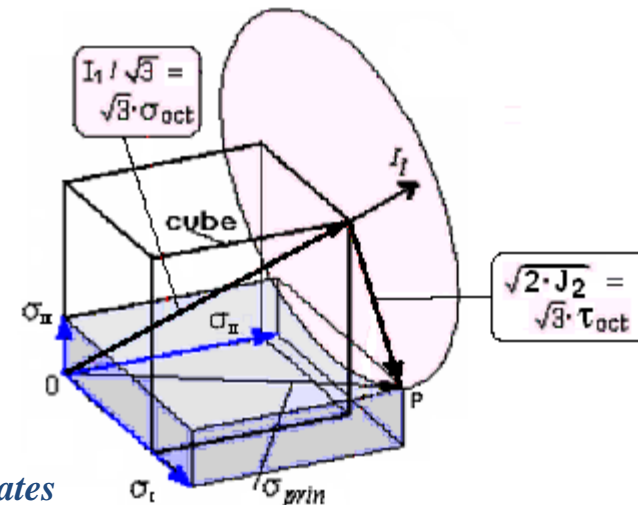
*Principal Stress Plane Cross-section of the Fracture Body (oblique cut)* as similarly behaving material



- Mapping must be performed in the 2D-plane because fracture data set is given there
- The 2D-mapping uses the 2D-subsolution of the 3D-strength failure conditions
- The 3D-fracture failure surface (body) is based on the 2D-derived model parameters.



The 3D-strength failure condition enables to predict the 120°-symmetric failure body and to judge a 3D- stress state

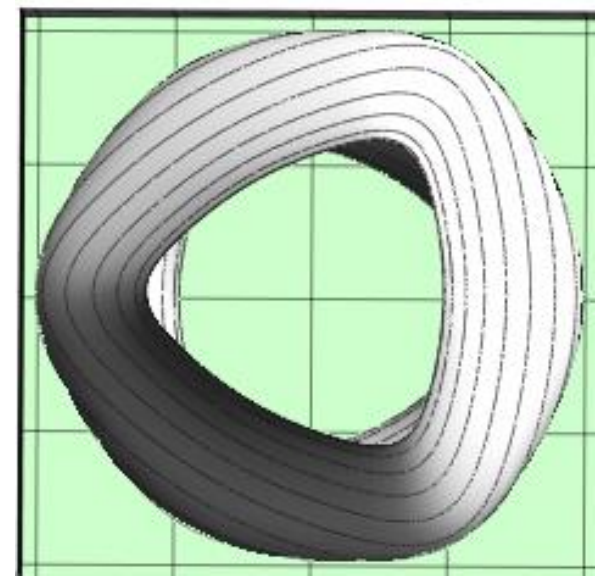
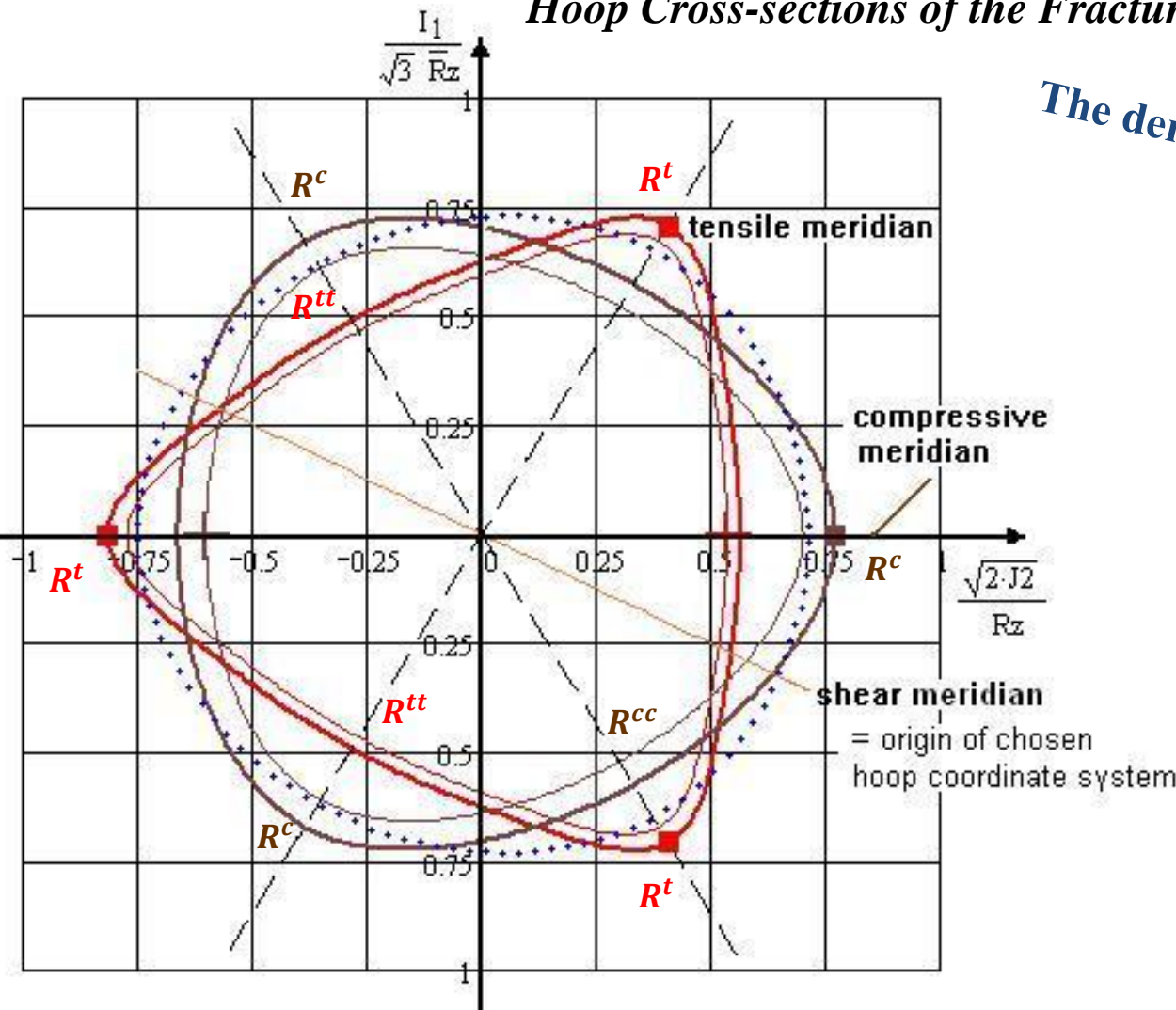


visualization of the Lode-Haigh-Westergaard coordinates

as similarly behaving material

## Hoop Cross-sections of the Fracture Body

The dent turns !

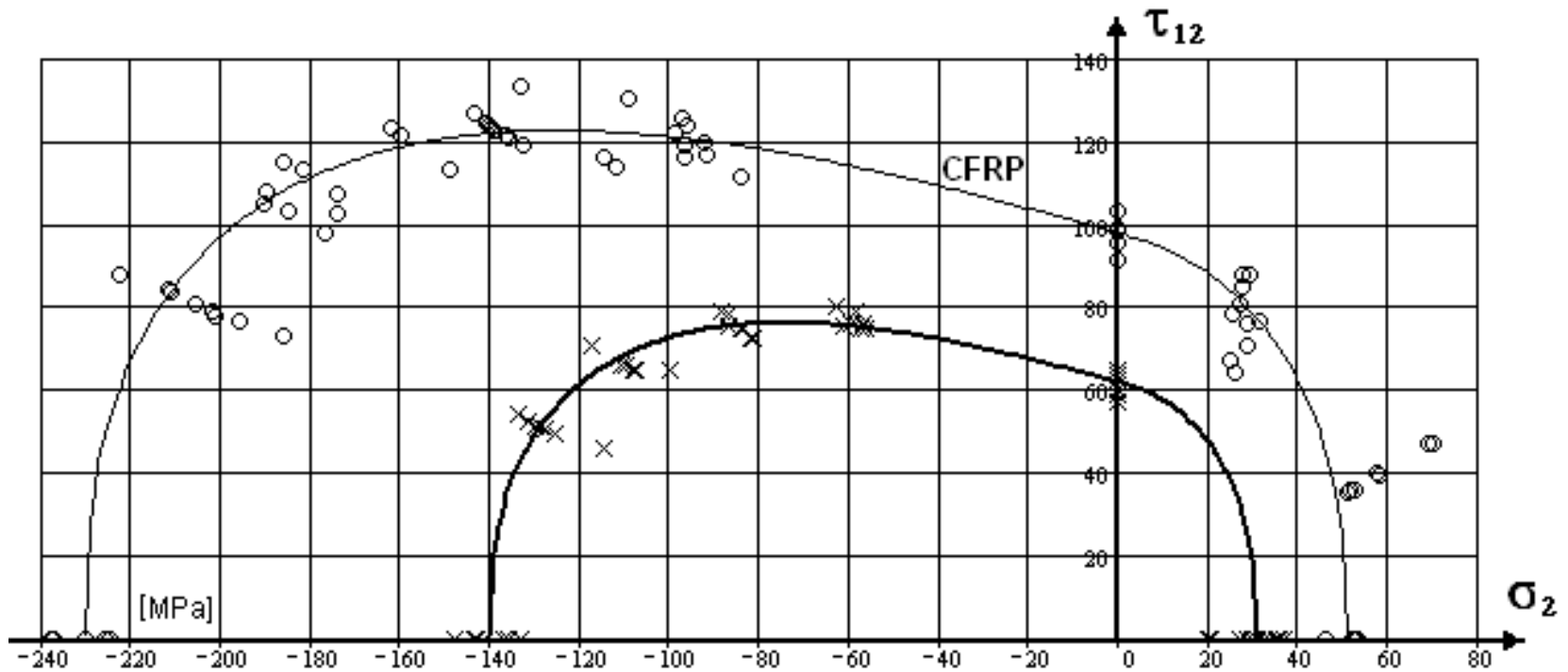
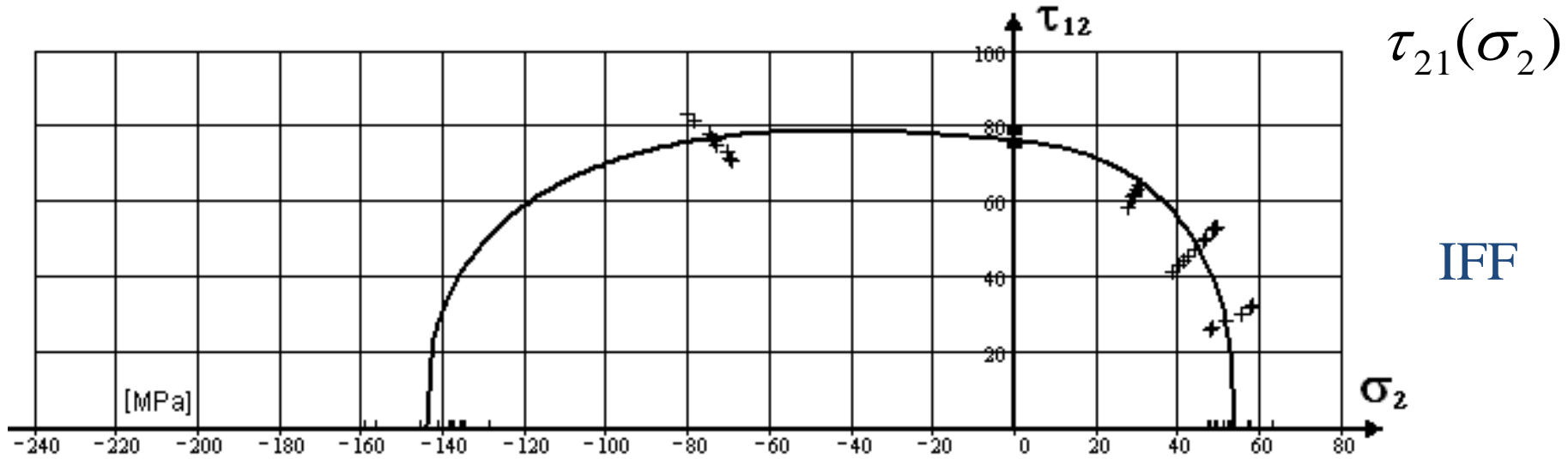


Caps: No test data, cone was chosen.

- Lode-angle, here set as  $\sin(3 \theta)$  :
- shear meridian angle =  $0^\circ$
- tensile meridian  $+30^\circ$  +
- compressive meridian  $-30^\circ$  +

$I_1 = 0$ , is interaction domain: Is about a circle.

# GFRP, CFRP examples, mapped by FMC-based UD SCF, 2D stress state



# Test Case 5, WWFE-II, UD test specimen, 3D stress state $\sigma_2 (\sigma_1 = \sigma_3)$

= hydrostatic pressure with additional loading

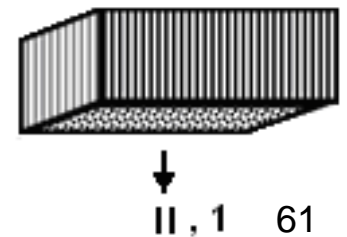
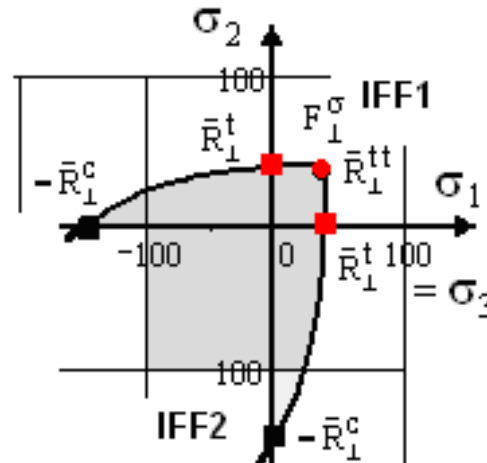
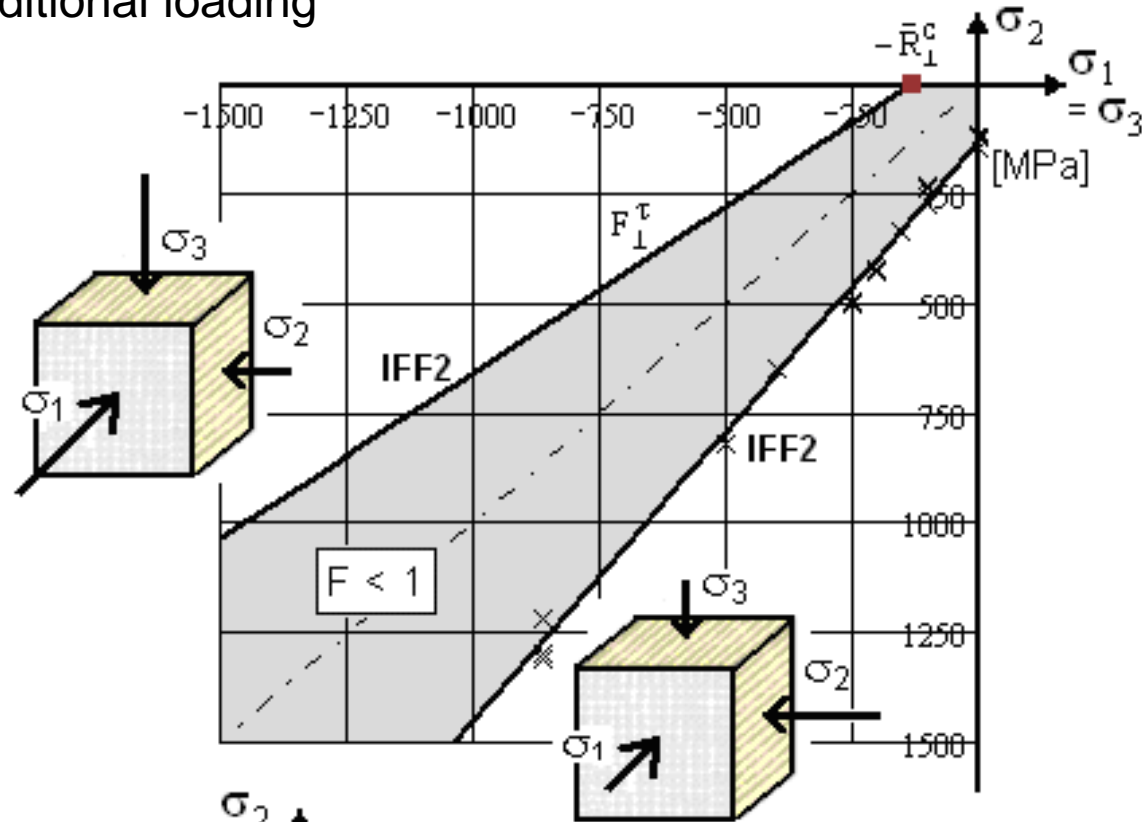
UD E-glass/MY750epoxy.

$$\nu_{\perp\parallel} = 0.28, \quad \mu_{\perp\perp} = 0.14, \quad m = 2.8,$$

$$\{\bar{R}\} = (1280, 800, 40, 132, 73)^T \text{ MPa}$$

Good Mapping, after QinetiQ re-evaluation of the lower branch test data. Then, the upper branch was fitting other test data, too!

Result: *Both branches were then reliable and could be used for model validation*



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8. Model Parameters
9. Standardized Material Test Methods **Kollege**
10. Structural Testing, NDI, Damage Tolerance
11. Structural Verification, Margin of Safety, Reserve Factor

What is Fatigue ? = process, that degrades material properties

What is Damaging? not damage, as used in English literature

Process wherein the results, the damaging portions, finally accumulate to a damage size such as a macro-scopic delamination.

The means is usually *Miner's Damaging Accumulation* model

What is Damage?

If above damage size is judged to be critical, then Damage Tolerance Analysis is used to predict its growth under further cyclic loading.

# State-of-the-Art in Cyclic Strength Analysis of UD Laminas (plies), Laminates

---

- Procedures base on specific laminates and therefore cannot be generally applied.  
Hence, no generally applicable Lifetime Prediction Method is available !
- Procedures base – as with metals – on stress amplitudes and mean stress correction. *Is this correct? Can one neglect that the damaging portions are linked to the various fracture failure modes in the case of brittle behaving materials?*
- Present: Engineering Approach: *Static Design Limit Strain of < 0.3%* , negligible matrix-microcracking.  
Design experience proved: No fatigue danger is given for multi-angle laminates
- Future : *Design Limit Strain shall be increased for better material exploitation*  
(EU-project: MAAXIMUS)  
Above  $\epsilon = 0.5\%$  level: *first filament breaks* , diffuse matrix-*microcracking* occurs in usually *fiber-dominated laminates*, used in high-stress applications.

*To tackle this, much effort must be put on this in future !*



## Was sind die benötigten zyklischen Eigenschaften?

---

- Wöhlerkurven  $R = const = \sigma_{unter} / \sigma_{ober}$
- Schädigungsakkumulationshypothese
- Quantifizierte Schädigungs'portionen' (-inkremente)

Dazu Anwendbarkeit der statischen Festigkeitshypothesen, wenn die

*Statischen Festigkeitswerte* durch  
*Restfestigkeitswerte für eine bestimmte Lebensdauer*  
ersetzt werden.

Statische Anstrengungssumme *Eff* (material stressing effort)

wird durch

Zyklische Schädigungssumme ***D***

ersetzt !

## State of the Art in Cyclic Strength Analysis of UD Laminas (plies)

---

- **No Lifetime Prediction Method** available, applicable to any Laminate
- **Procedures base** – as with metals – on stress amplitudes and mean stress correction
- **Procedures base** on specific laminates and therefore cannot be generally applied

- **Present: Engineering Approach:**

*Static Design Limit Strain* of  $< 0.3\%$  , negligible matrix-microcracking.

Design experience proved: **No** fatigue danger given

- **Future : *Design Limit Strain shall be increased*** (EU-project: MAAXIMUS)

**We must react!**

Above  $\varepsilon = 0.5\%$  *first filament breaks* , diffuse matrix-*microcracking* occurs  
in usually *fiber-dominated laminates* used in high-stress applications .



## Questions an engineer poses in the case of cyclic design

---

- 1. When does damaging start?**
- 2. How can one consider the single (micro-)damaging portion?**
- 3. How are the single damaging portions accumulated?**
- 4. When do the accumulated damageing portions form a damage?**
- 5. When becomes such a damage (delamination, impact) critical?**
- 6. How is the damage growth in the 3rd or final phase of fatigue life (fixation of part replacement time, inspection intervals)?**

## 1 Foundation of the German Academic Research Group (BeNa) “Betriebsfestigkeits-Nachweis“ for High-Performance Structures (2010)

- \* *physically-based (on failure modes),*
- \* *ply-oriented in order to obtain a generalisation for any UD lamina-composed laminate*

*Objective:  
Release of a VDI-Guideline*

## 2 Foundation of sub-group of my CCEV-working group ‘Engineering’ “*Composite Fatigue*“ together with the CCEV member company CADCON (2012).

- **Ductile material behaviour (e.g. many metals):**

- \* *Slip band shear yielding - as damaging driver - occurs under cyclic tensile stress, compressive stress, and under shear stress !*

- \* *Therefore, this single mechanism*

- shear stress–caused yielding can be principally described by*

- one yield failure condition to determine the needed damaging portions !*

- (Formulation is in normal stresses, but the shear stress is the damaging driver).*

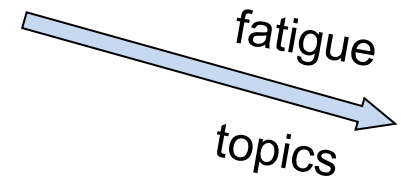
- \* *Increasing with brittleness, lifetime estimation is corrected by accounting for the ‘Mean stress effect’  $\sigma_{mean}$*

- (considers by Goodman Diagram that more mechanisms really act).*

- **Brittle material behaviour :**

- \* *Many mechanisms, causing damaging, must be considered*

- Question is whether a correction by the ‘Mean stress effect’ makes sense.*



# Applicability of Static Strength Failure Conditions?

---

## Proven Assumption:

If the damaging mechanisms (failure modes) are equal, then

- failure parameters that drive cyclic damaging are equal, too, and
- transferability from static failure to cyclic failure is permitted

However, static strength must be replaced by the  
fatigue strength = residual strength of the  
shrinking failure body.

Therefore,

as necessary static tool, my

FMC-based Static Failure Conditions (criteria) shall be briefly derived which  
were very successful in the World-Wide-Failure-Exercise (WWFE 1992-2014).

From all the contributors, my non-funded Failure Conditions  
well mapped the largest number of test data courses in WWFE-I and WWFE-II !

# Schritte bei der Lebensdauerabschätzung

---

## 1 Input

**Betriebsbelastungen: Last-Zeit-Kurven** (Modellierung mit rain flow, ..)

**Sicherheitskonzept: Design to Life**  $j_{\text{Life}} = 3 - 4$

## 2 Übertragung der Betriebsbelastungen in Beanspruchungen (Spannungen) mittels Strukturanalyse)

## 3 Bereiche der Ermüdungsanalyse

**LCF: high stressing,**

**HCF: intermediate stressing**

**VHCF: low stressing and strains (SPP1466)**

## 4 Erfassung der Betriebsbelastung


**Zeitbereich:** Zyklus-für-Zyklus oder Kollektiv-für-Kollektiv (weniger Rechenaufwand)

**Frequenzbereich:** Lastspektren (Verlust der Last-Reihenfolge) oder Blockbelastungen, etc.

Because semi-brittle, brittle behaving materials experience several failure modes or mechanisms.

**Consequence:**

**More than one strength failure condition (criterion) must be employed**



... and for the UD-composed brittle behaving laminates  
with 5 failure modes  
5 FMC strength failure conditions are considered !

Stress (not strain) criteria are applied to determine the subsequent damaging portions:

- capture the combined effect of lamina stresses and
- consider residual stresses from manufacturing cooling down (essential for HCF)

- **Determination of damaging portions** (from diffuse and later discrete damaging)
- **Accumulation of damaging portions** (cycle-wise, block-wise, or otherwise ? )



## Experience with to-date Composites from fiber-reinforced plastics

---

- *behave brittle*
- *experience early fatigue damage*
- *show benign fatigue failure behaviour in case of 'well-designed', fiber-dominated laminates until final 'Sudden Death'.*

( fiber-dominated:= 0° plies in all significant loading directions, > 3 angles )

**Annahmen:** Falls Versagensmechanismen(-modi) gleich?

- Dann auch die schädigungstreibenden Versagensparameter gleich.
- Übertragbarkeit statisches Versagen auf Ermüdung möglich,

*Dabei schädigen ebene (2D) und räumliche (3D) Spannungszustände*

**Meßbare Schädigungsgrößen:**

*Mikrorißdichte, Restfestigkeit, Reststeifigkeit*

- **Duktiles Werkstoffverhalten** (Beispiel: isotrope Metalle)

**1 Mechanismus = “Schubspannungsgleiten“**

passiert unter allen zyklischen Beanspruchungen:

*Zugspannungen, Druckspannungen, Schub- und Torsionsspannungen !*

*Deswegen kann dieser einzige Mechanismus ‘Schubspannungsbasiertes Gleiten‘ mit einer einzigen Fließbedingung beschrieben werden!*

- **Sprödes Werkstoffverhalten bei isotropen Werkstoffen**

**2** Schädigung erzeugende **Mechanismen** wirken

*(ingenieurmäßige Berücksichtigung durch sog. Mittelspannungskorrektur)*

- **Sprödes Werkstoffverhalten bei UD- Werkstoffen**

**5** Schädigung erzeugende **Mechanismen** wirken

*(Ansätze mit und ohne Mittelspannungskorrektur)*

# Example: Fatigue of endless fiber-reinforced UD Laminates

## Damaging drivers

---

- Ductile material behaviour (e.g. many metals):

- \* *Slip band shear yielding - as damaging driver - occurs under cyclic tensile stress, compressive stress, and under shear stress !*

- \* *Therefore, this single mechanism*

- shear stress–caused yielding can be principally described by*

- one yield failure condition to determine the needed damaging portions !*

- (Formulation is in normal stresses, but the shear stress is the damaging driver).*

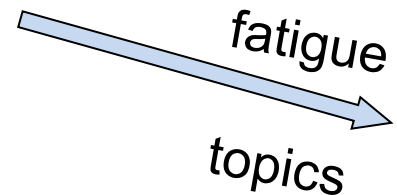
- \* *Increasing with brittleness, lifetime estimation is corrected by accounting for the ‘Mean stress effect’  $\sigma_{mean}$*

- (considers by Goodman Diagram that more mechanisms really act).*

### Brittle material behaviour :

- \* *Many mechanisms, causing damaging, must be considered*


- Question is whether a correction by the ‘Mean stress effect’ makes sense.*



Because semi-brittle, brittle behaving materials experience several failure modes or mechanisms.

**Consequence:**

**More than one strength failure condition (criterion) must be employed**



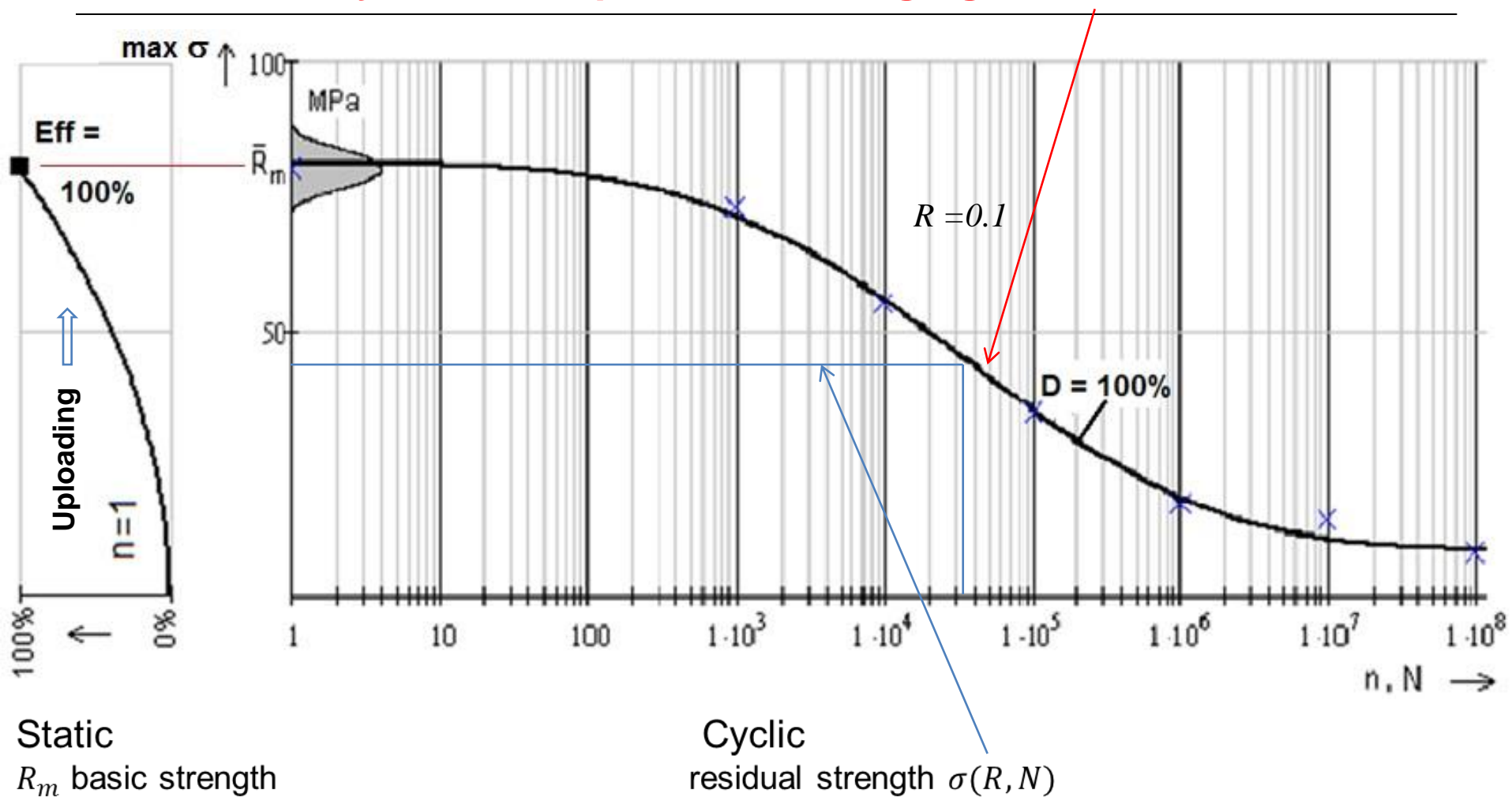
... and for the UD-composed brittle behaving laminates with 5 failure modes  
5 FMC strength failure conditions are considered !

Stress (not strain) criteria are applied to determine the subsequent damaging portions:

- capture the combined effect of lamina stresses and
- consider residual stresses from manufacturing cooling down (essential for HCF)

- *Determination of damaging portions* (from diffuse and later discrete damaging)
- *Accumulation of damaging portions* (cycle-wise, block-wise, or otherwise ? )

# Static and cyclic development of damaging, S-N-curve



Analogous limits of the material capacities :

- Static : material stressing effort  $Eff = 100\%$
- Cyclic : material damaging sum  $D = 100\%$

For brittle behaving materials it is advantageous to use  $max\sigma \equiv R_m$  instead of  $\Delta\sigma$

## Failure mode-linked Master S-N-curves

---

For lifetime estimation usually – even in a distinct failure mode – several S-N-curves are needed

*testing requires high effort!*

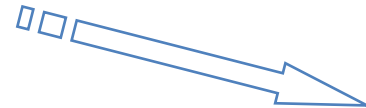
### Idea

Measurement of just one failure mode linked Master S-N-curve

- for a fixed stress ratio  $R$
- prediction of additionally necessary S-N-curves on basis of the master curve and on the 'principle of equivalent strain energy'!

Then, for the often used

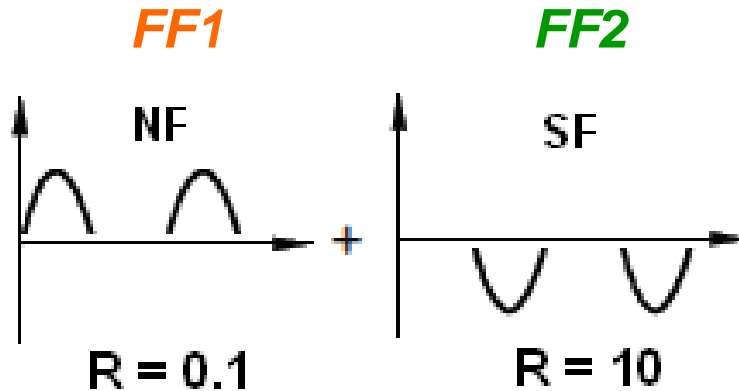
all possible load orientations capturing fiber-dominatedly designed, multidirectional laminates, composed of UD plies,  
an engineering-like model is derivable.



**S** := cyclic stress range  $\Delta\sigma$  , **N** := number of cycles to failure, stress ratio **R** :=  $\min\sigma/\max\sigma$

# Application of Miner-'Rule'

## Mode-wise Accumulation of Damaging Portions (novel)



Simple Example: again

$$R = -1$$

$$D (FF1, FF2) = NF : (n_1 / N_1 + n_2 / N_2 + n_3 / N_3) + SF : (n_4 / N_4)$$

$$+ D (IFF1, IFF2, IFF3) = D \leq D_{feasible}$$

from test experience

Calculation, see [Cun13b]



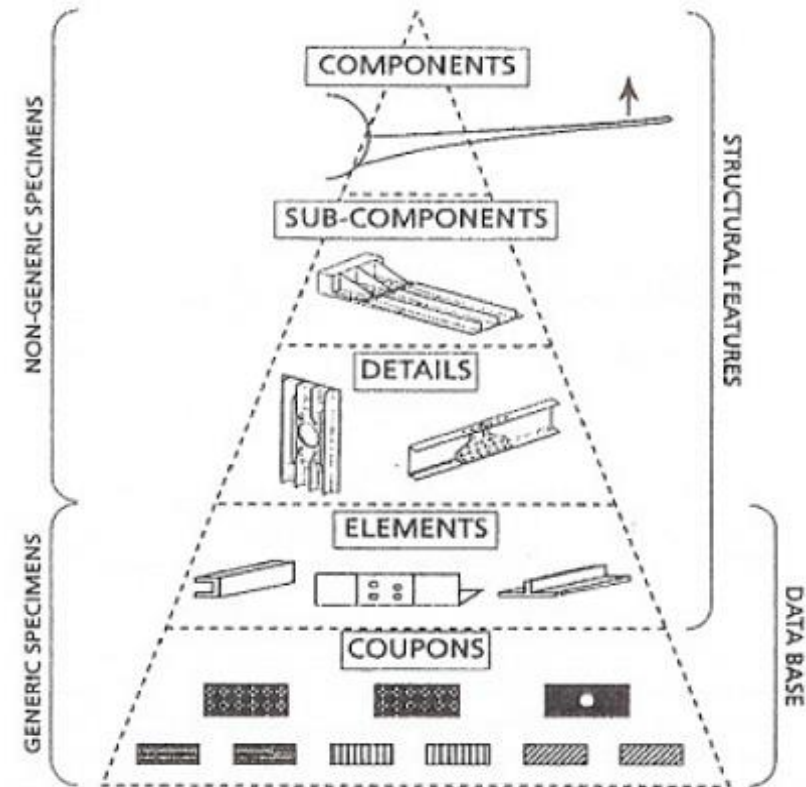
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7. **Material Properties**
8. Model Parameters
9. Standardized Material Test Methods **Kollege**
10. Structural Testing, NDI, Damage Tolerance
11. Structural Verification, Margin of Safety, Reserve Factor

# Characterisation of Composite Material and Components

MIL Hdbk 17: <i>Composites</i>	Material			Structure
Structural complexity level	<i>Screening</i>	<i>Qualification</i>	<i>Acceptance</i>	<i>structural substantiation</i>
<i>constituent</i>	X			
<i>lamina</i>	X	X	X	
<i>laminate</i>		X	X	X
<i>structural element</i>	X	X	X	X
<i>structural compon.</i>				X

The pyramid of tests MIL-HDBK-17-1E



*Modelling & Discretizing  
determines type  
of test specimen*

**composite test specimens**

**Gottfried Wilhelm Leibniz (about 1800)**

**A general system  
of signs and symbols is of  
high importance for  
a logically consistent universal language  
for scientific use !**

# Self-explaining Notations for Strength Properties (homogenised material) neu !!!!

		Fracture Strength Properties									<i>required by material symmetry</i>
loading		tension			compression			shear			
direction or plane		1	2	3	1	2	3	12	23	13	
9	general orthotropic	$R_1^t$	$R_2^t$	$R_3^t$	$R_1^c$	$R_2^c$	$R_3^c$	$R_{12}$	$R_{23}$	$R_{13}$	comments
5	UD, $\cong$ non-crimp fabrics	$R_{//}^t$ NF	$R_{\perp}^t$ NF	$R_{\perp}^t$ NF	$R_{//}^c$ SF	$R_{\perp}^c$ SF	$R_{\perp}^c$ SF	$R_{//\perp}$ SF	$R_{\perp\perp}$ NF	$R_{//\perp}$ SF	$R_{\perp\perp} = R_{\perp}^t / \sqrt{2}$ (compare Puck's modelling)
6	fabrics	$R_W^t$	$R_F^t$	$R_3^t$	$R_W^c$	$R_F^c$	$R_3^c$	$R_{WF}$	$R_{F3}$	$R_{W3}$	$Warp = Fill$
9	fabrics general	$R_W^t$	$R_F^t$	$R_3^t$	$R_W^c$	$R_F^c$	$R_3^c$	$R_{WF}$	$R_{F3}$	$R_{W3}$	$Warp \neq Fill$
5	mat	$R_{1M}^t$	$R_{1M}^t$	$R_{3M}^t$	$R_M^c$	$R_{1M}^c$	$R_{3M}^c$	$R_M^{\tau}$	$R_M^{\tau}$	$R_M^{\tau}$	$R_M^{\tau} ( R_M^t )$
2	isotropic	$R_m$ SF	$R_m$ SF	$R_m$ SF	<i>deformation-limited</i>			$R_M^{\tau}$	$R_M^{\tau}$	$R_M^{\tau}$	<i>ductile, dense</i> $R_M^{\tau} = R_m / \sqrt{2}$
		$R_m$ NF	$R_m$ NF	$R_m$ NF	$R_m^c$ SF	$R_m^c$ SF	$R_m^c$ SF	$R_m^{\sigma}$ NF	$R_m^{\sigma}$ NF	$R_m^{\sigma}$ NF	<i>brittle, dense</i> $R_M^{\sigma} = R_m^t / \sqrt{2}$

**NOTE:** \*As a consequence to isotropic materials (European standardisation) the letter R has to be used for strength. US notations for UD material with letters X (direction 1) and Y (direction 2) confuse with the structure axes' descriptions X and Y . \*Effect of curing-based residual stresses and environment dependent on hygro-thermal stresses. \*Effect of the difference of stress-strain curves of e.g. the usually isolated UD test specimen and the embedded (redundancy ) UD laminae.  $R_m :=$  'resistance maximale' (French) = tensile fracture strength (superscript t here usually skipped), R:= basic strength. Composites are most often brittle and dense, not porous! SF = shear fracture

# Elasticity Properties (*homogenised material*) (self-explaining denotations)

considers VDI  
2014,

proposed to  
ESA-Hdbk

Elasticity Properties										
direction or plane	1	2	3	12	23	13	12	23	13	
9 <i>general orthotropic</i>	$E_1$	$E_2$	$E_3$	$G_{12}$	$G_{23}$	$G_{13}$	$\nu_{12}$	$\nu_{23}$	$\nu_{13}$	<b>comments</b>
5 <i>UD, <math>\cong</math> non-crimp fabrics</i>	$E_{//}$	$E_{\perp}$	$E_{\perp}$	$G_{//\perp}$	$G_{\perp\perp}$	$G_{//\perp}$	$\nu_{//\perp}$	$\nu_{\perp\perp}$	$\nu_{//\perp}$	$G_{\perp\perp} = E_{\perp} / (2 + 2\nu_{\perp\perp})$ $\nu_{\perp//} = \nu_{//\perp} \cdot E_{\perp} / E_{//}$ <i>quasi-isotropic 2-3-plane</i>
6 <i>fabrics</i>	$E_W$	$E_F$	$E_3$	$G_{WF}$	$G_{W3}$	$G_{F3}$	$\nu_{WF}$	$\nu_{W3}$	$\nu_{W3}$	<i>Warp = Fill</i>
9 <i>fabrics general</i>	$E_W$	$E_F$	$E_3$	$G_{WF}$	$G_{W3}$	$G_{F3}$	$\nu_{WF}$	$\nu_{F3}$	$\nu_{W3}$	<i>Warp <math>\neq</math> Fill</i>
5 <i>mat</i>	$E_M$	$E_M$	$E_3$	$G_M$	$G_{M3}$	$G_{M3}$	$\nu_M$	$\nu_{M3}$	$\nu_{M3}$	$G_M = E_M / (2 + 2\nu_M)$ <i>1 is perpendicular to quasi-isotropic mat plane</i>
2 <i>isotropic for comparison</i>	$E$	$E$	$E$	$G$	$G$	$G$	$\nu$	$\nu$	$\nu$	$G = E / (2 + 2\nu)$

Lesson Learned: - *Unique, self-explaining denotations are mandatory*

- *Otherwise, expensively generated test data cannot be interpreted and go lost*

# Hygrothermal Properties (*homogenised material*)

		Hygro-thermal properties					
direction		1	2	3	1	2	3
9	<b>general orthotropic</b>	$\alpha_{T1}$	$\alpha_{T2}$	$\alpha_{T3}$	$\alpha_{M1}$	$\alpha_{M2}$	$\alpha_{M3}$
5	<b>UD, ≅ non-crimp fabrics</b>	$\alpha_{T//}$	$\alpha_{T\perp}$	$\alpha_{T\perp}$	$\alpha_{M//}$	$\alpha_{M\perp}$	$\alpha_{M\perp}$
6	<b>fabrics</b>	$\alpha_{TW}$	$\alpha_{TW}$	$\alpha_{T3}$	$\alpha_{MW}$	$\alpha_{MW}$	$\alpha_{M3}$
9	<b>fabrics general</b>	$\alpha_{TW}$	$\alpha_{TF}$	$\alpha_{T3}$	$\alpha_{MW}$	$\alpha_{MF}$	$\alpha_{M3}$
5	<b>mat</b>	$\alpha_{TM}$	$\alpha_{TM}$	$\alpha_{TM3}$	$\alpha_{MM}$	$\alpha_{MM}$	$\alpha_{MM3}$
2	<b>isotropic for comparison</b>	$\alpha_T$	$\alpha_T$	$\alpha_T$	$\alpha_M$	$\alpha_M$	$\alpha_M$

.. analogous for  $\lambda, c$   
material friction  $\mu$   
as strength property

NOTE: Despite of annoying some people, I propose to rethink the use of  $\alpha$  for the CTE and  $\beta$  for the CME.  
Utilizing  $\alpha_T$  and  $\alpha_M$  automatically indicates that the computation procedure will be similar.

Key Words: Material properties, CFRP, T300, Code69, UD-Prepreg

## References

- [1] Report QE-630 / 83, WEP, DORNIER, 1978/79
- [2] Report DOL73/74, DORNIER, 1973/74
- [3] Report SK50-266/85, DORNIER, 1985

## 1 Material

Material specification	CFRP T300 / Code69 UD-Prepreg
Specification for delivery	DOL 74, Edition January 1978

Characteristic	Unit	Value (remarks)
Fiber type		Toray T300/6K
Fiber density	g/cm <sup>3</sup>	1.75
Matrix type		Epoxy-Code69
Matrix density	g/cm <sup>3</sup>	1.27
Prepreg ply thickness	mm	0.231
Contents of prepreg resin	mass %	43±2.5
Fiber volume fraction	%	60
Prepreg density	g/cm <sup>3</sup>	1.56
Cure process specification		DOL 74, Edition January 1978
Cure temperature	°C	175 (hold time = 75 min.)
Cure vacuum	bar	0.07 (hold time = whole process)
Cure pressure	bar	7 (hold time = whole process)
Post cure temperature	°C	

## 2 Physical properties

Characteristic	Unit	Value	Statistics	Ref.
$c$	kJ/(K·kg)			
$\lambda_{  }$	W/(K·m)			
$\lambda_{\perp}$	W/(K·m)			
$\kappa_{  }$	1/( $\Omega \cdot m$ )			
$\kappa_{\perp}$	1/( $\Omega \cdot m$ )			

## 3 Mechanical properties

Test temperature		RT		
Moisture contents		SA <sup>a</sup>		
Property	Unit	Value	Statistics	Ref.
$R_{  t}$	MPa	1280	A	[1]
$R_{\perp t}$	MPa			
$R_{  e}$	MPa			
$R_{\perp e}$	MPa			
$R_{  l}$	MPa			
$ILSS$	MPa	65	A	[1]
$E_{  t}$	MPa	133000/116000	$\bar{x}/A$	[1]
$E_{\perp t}$	MPa			
$E_{  e}$	MPa			
$E_{\perp e}$	MPa			
$G_{  \perp}$	MPa			
$\nu_{  \perp}^e$	-	0.32	$\bar{x}$	[2]
$\nu_{\perp\perp}$	-	0.4 <sup>d</sup>		
$e_{  t}$	%	1.3	S	[2]
$e_{\perp t}$	%			
$e_{  e}$	%			
$e_{\perp e}$	%			
$e_{  \perp}$	%			
$\alpha_{M  }$	mm/(mm·%)			
$\alpha_{M\perp}$	mm/(mm·%)			
$\alpha_{T  }$	mm/(mm·K)	$-(0.8 \pm 0.2) \cdot 10^{-6}$	$\bar{x}$	[2]
$\alpha_{T\perp}$	mm/(mm·K)	$+(27.0 \pm 1.0) \cdot 10^{-6}$	$\bar{x}$	[2]
$T_g$	°C	200	<sup>b</sup>	[3]

$T_g$  = glass transition temperature

<sup>a</sup> = Standard Atmosphere according to ISO554/DIN50014; 23/50 = 23 ± 2°C/50 ± 5%RH

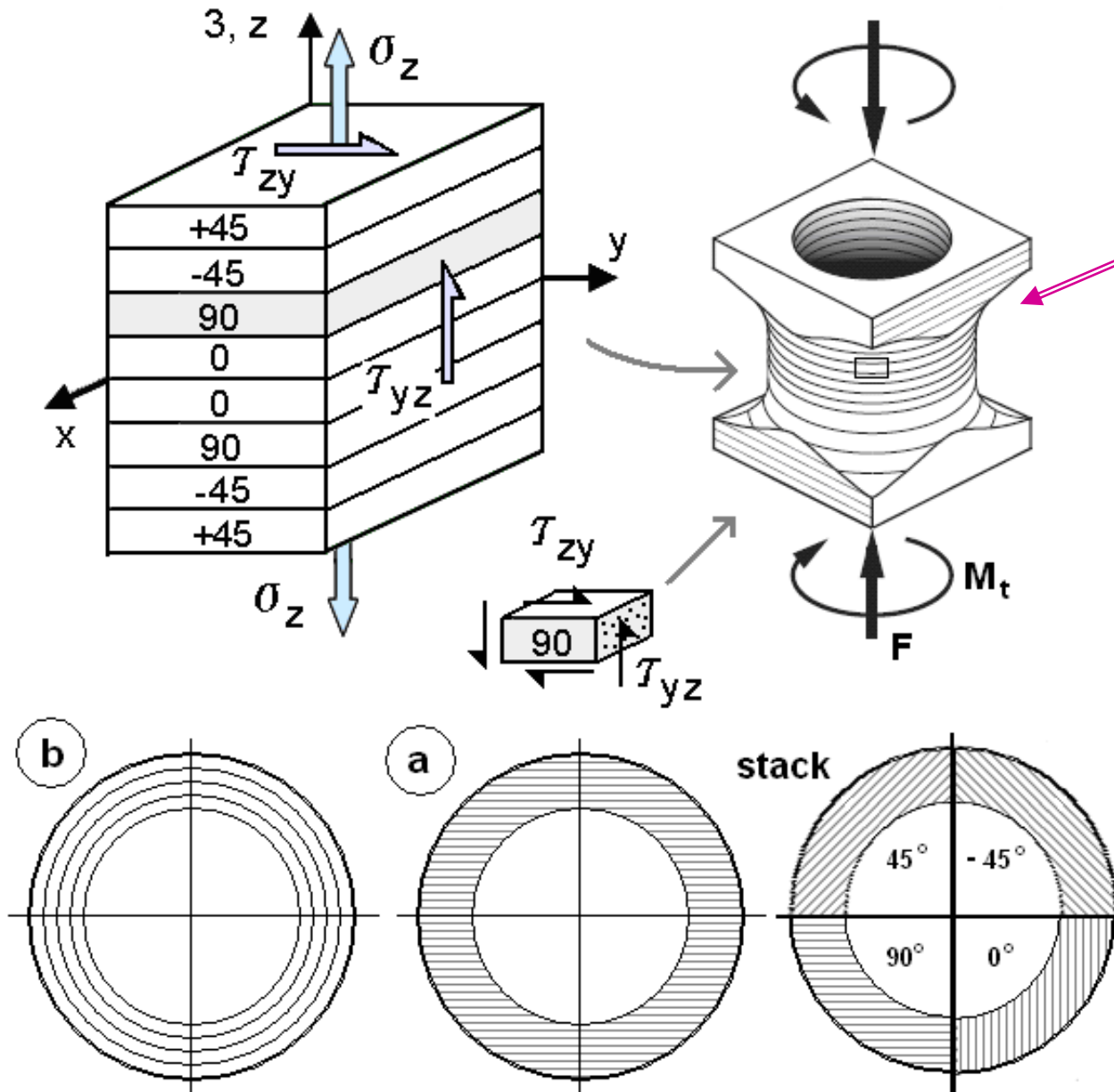
<sup>b</sup> = no statistical base

<sup>c</sup> = major value

<sup>d</sup> = assumed

Note: DOL 74 (Edition Jan.78) has been replaced by DOL 74 (Edition Nov.82), strength data have not changed. Determination of the elastic moduli according to LN 29971.

# Test Case 10, Test Specimen, WWFE-II, Test domain around the critical material location must be smooth!



tube milled from a laminate block

**Lesson Learnt: Not usable for the validation of a SFC!**

no smooth stress domain for validating failure conditions

edge effects etc.



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# UD lamina (ply) : Micro-mechanical Properties

---

**Some lamina analyses require a micro-mechanical input:**

**Problem:** Not all micro-mechanical properties can be measured.

**Solution:** Micro-mechanical equations are calibrated by macro-mechanical test results (lamina level) = an *inversal parameter identification*

**Condition:** micro-mechanical properties can be used only together with the equations they have been determined with.

# Mind the difference in UD-analysis : Isolated and embedded UD-behaviour

‘Isolated‘ lamina test specimens

= weakest link results (series failure system)

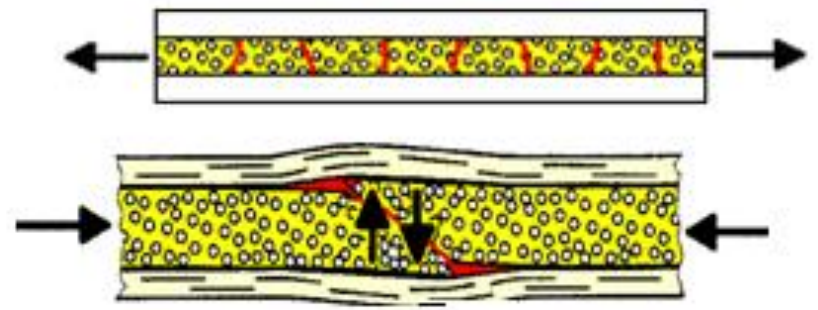
‘Embedded‘ laminas experience in-situ effects

Measurement/Determination of strain softening curve ?

= redundancy result (parallel failure system)



unconstrained lamina



mutually constrained laminas, in laminates

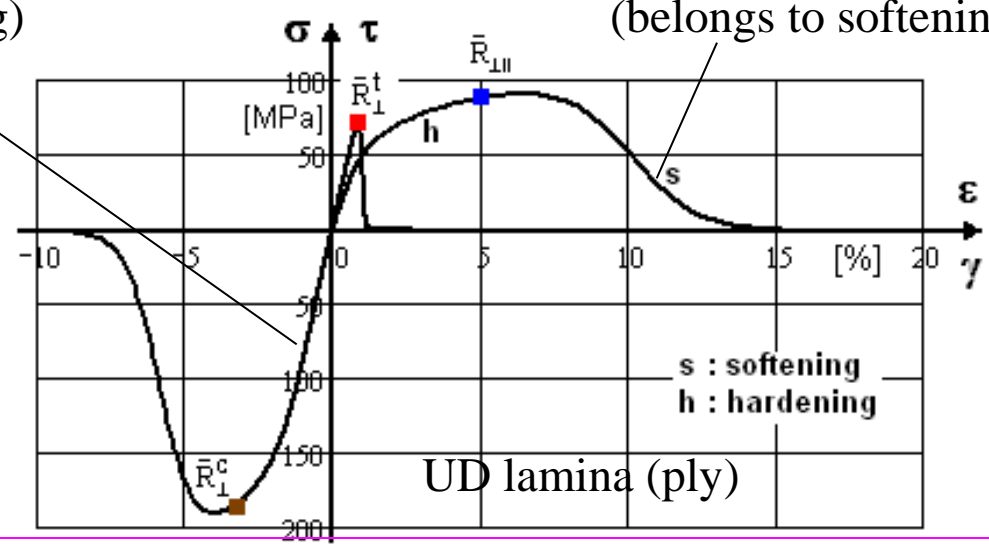
delivers strength property, stress-strain curve

in non-linear laminate analysis

(belongs to hardening)

(belongs to softening)

delivers basic strength as analysis input !



**Lesson Learned:** *In the Post-IFF regime the embedded lamina experiences no sudden death but still has residual strength and stiffness due to in-situ effect!*

# Determination of the 2 Friction Parameters (Mohr-Coulomb relationship)

(brittle behaviour)

IFF 3 :

$$\tau_{21} = R_{\perp\parallel} - b_{\perp\parallel} \cdot \sigma_2 \quad : \text{FMC corresponds}$$

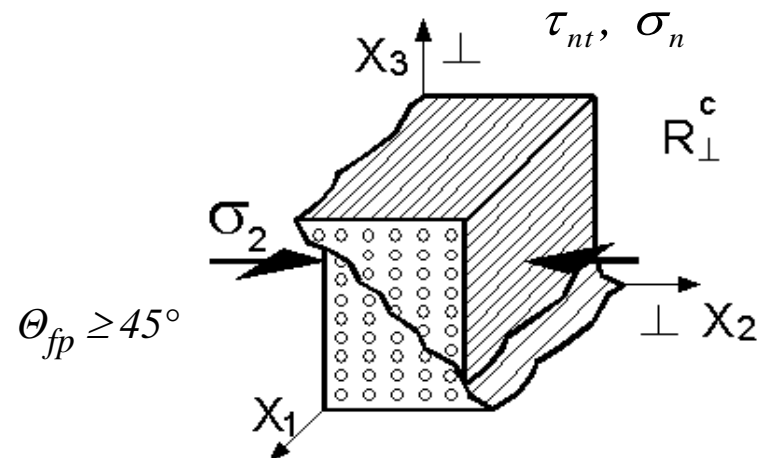
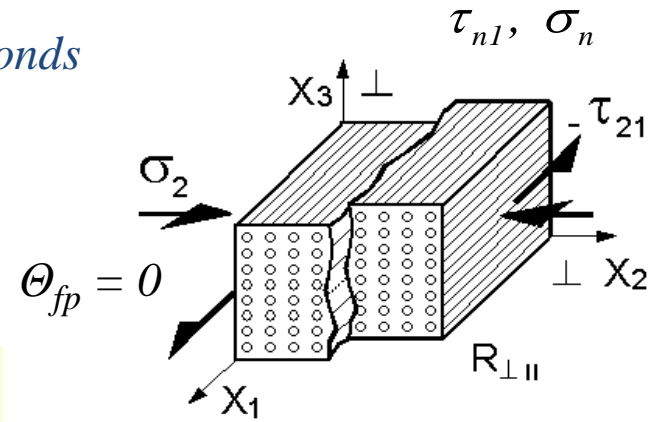
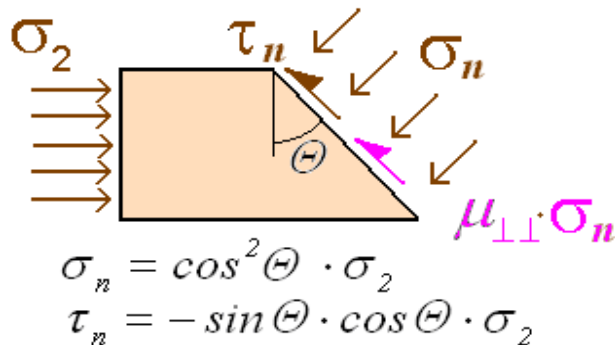
$$\tau_{n1} = R_{\tau}^{\perp\parallel} - \mu_{\perp\parallel} \cdot \sigma_n \quad : \text{Mohr}$$

cohesion strength      material internal friction coefficient

**Linear Mohr-Coulomb approach + denotation**

IFF 2 :

$$\tau_{nt} = R_{\tau}^{\perp\perp} - \mu_{\perp\perp} \cdot \sigma_n$$



UD material: 2 ; isotropic material: 1  
real material = crystal + friction

# Determination of the Friction Parameter $\mu_{\perp\perp}$ (linear Mohr-Coulomb relationship)

From evaluation of the test data:

$$\theta_{fp}^c = 55^\circ, R_{\perp}^c = 104 \text{ MPa}$$

FMC:

$$b_{\perp}^{\tau} \sqrt{I_4} = \bar{R}_{\perp}^c - (b_{\perp}^{\tau} - 1) I_2$$

$$b_{\perp}^{\tau} = \frac{1}{1 + (\cos 2\theta_{fp}^c)} = \frac{1}{1 - \mu_{\perp\perp}}$$

$$b_{\perp}^{\tau} = 1.52$$

Mohr-Coulomb:

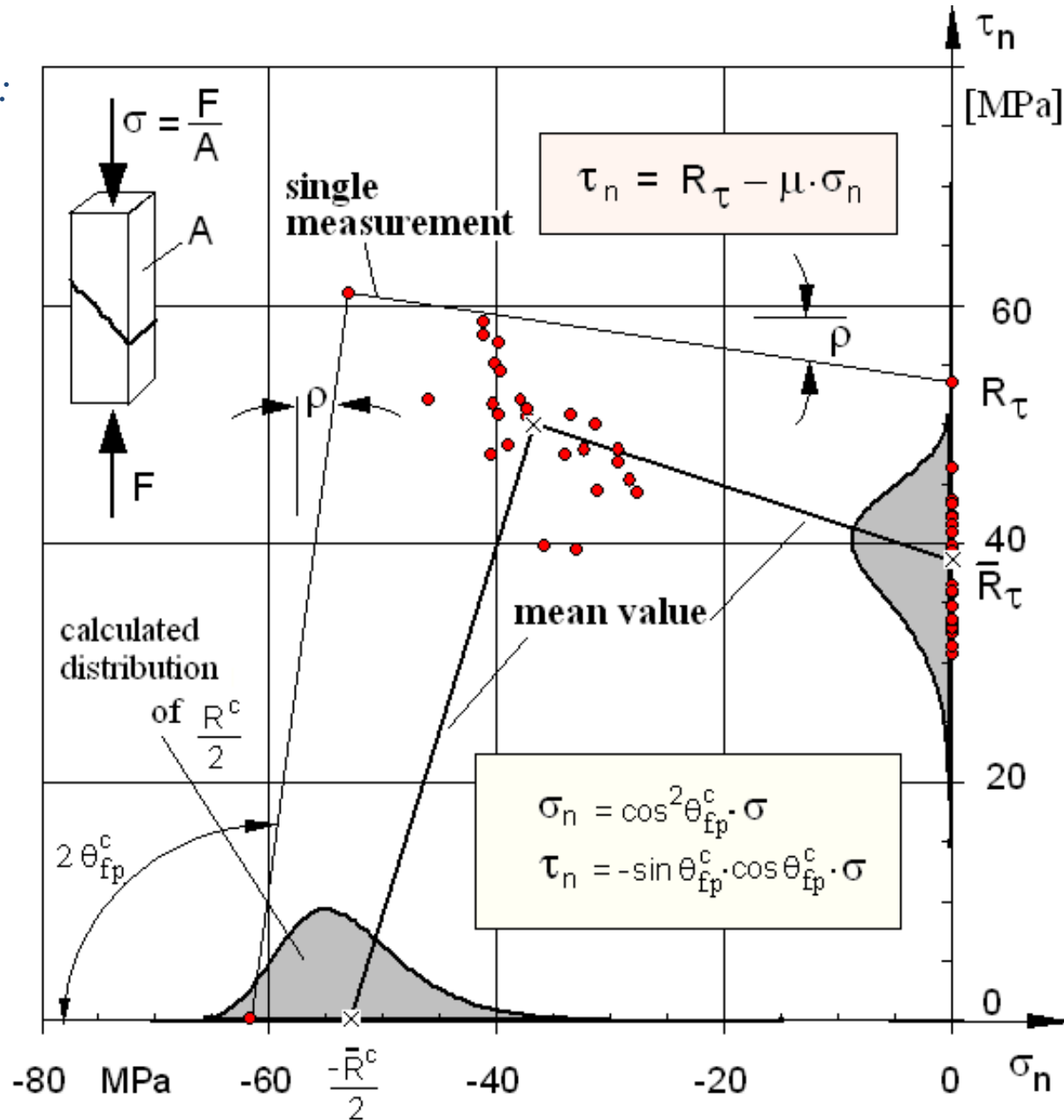
$$\tau_n = (R_{\tau}^{\perp\perp} - \mu_{\perp\perp} \cdot \sigma_n)$$

$$\mu_{\perp\perp} \geq -\cos 2\theta_{fp}^c,$$

$$\mu_{\perp\perp} = 0.34$$

$$R_{\tau}^{\perp\perp} = R_{\perp}^c \frac{1 + \cos 2\theta_{fp}^c}{2}$$

$$R_{\tau}^{\perp\perp} = 36.4 \text{ MPa}$$



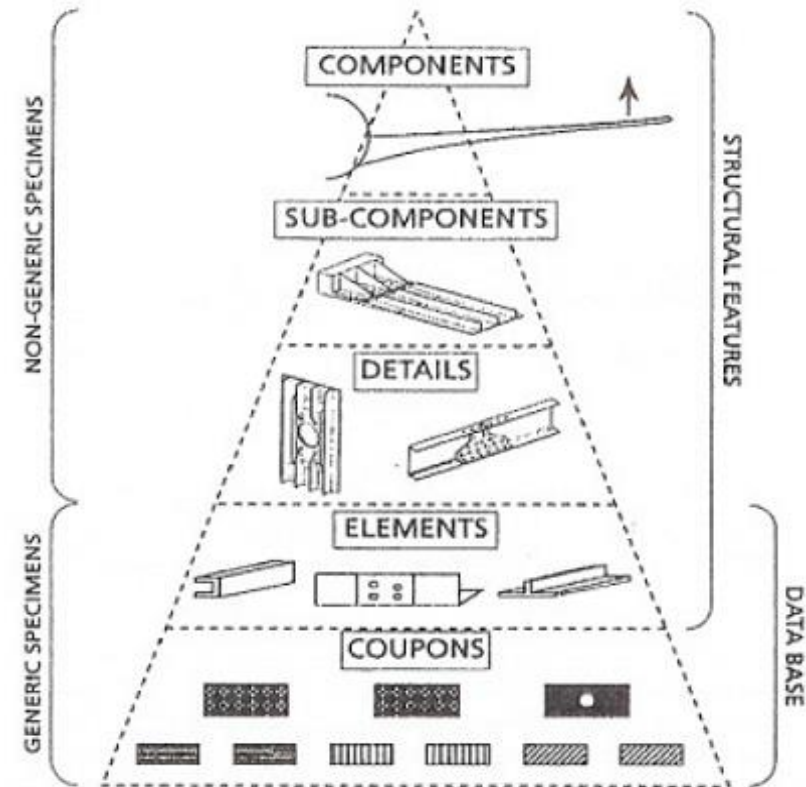
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constituent	X			
lamina	X	X	X	
laminate		X	X	X
structural element	X	X	X	X
structural compon.				X

The pyramid of tests MIL-HDBK-17-1E



*Modelling & Discretizing  
determines type  
of test specimen*

**composite test specimens**

Wyoming Test Fixtures, .  
information.

# Test Standards Used

## SECTION A: SHEAR LOADING

- A-1 [Iosipescu Shear \(ASTM D 5379\)](#)
- A-2 [V-Notched Rail Shear \(ASTM D 7078\)](#)
- A-3 [Short Beam Shear \(ASTM D 2344\)](#)
- A-4 [Two-Rail Shear \(ASTM D 4255\)](#)
- A-5 [Three-Rail Shear \(ASTM D 4255\)](#)
- A-6 [Shear Strength by Punch Tool \(ASTM D 732\)](#)
- A-7 [Sandwich Panel Flatwise Shear \(ASTM C 273\)](#)
- A-8 [Special Sandwich Panel Shear Fixture \(ASTM C 273\)](#)

## SECTION B: COMPRESSION LOADING

- B-1 [Wyoming Combined Loading Compression \(ASTM D 6641\)](#)
- B-2 [Modified ASTM D 695 \(Boeing BSS 7260\)](#)
- B-3 [IITRI Compression \(ASTM D 3410\)](#)
- B-4 [Wyoming Modified IITRI](#)
- B-5 [Wyoming Modified Celanese](#)
- B-6 [Celanese \(formerly ASTM D 3410\)](#)
- B-7 [German Modified Celanese \(DIN 65 380\)](#)
- B-8 [Edgewise Compressive Strength \(ASTM C 364\)](#)
- B-9 [NASA Short Block Compression](#)
- B-10 [Lockheed F-22 Test Fixtures](#)
- B-11 [Compression Subpress \(ASTM D 695\)](#)
- B-12 [Compression Platens - Fixed and Spherical Seat](#)

## SECTION C: SPECIALTY COMPRESSION TEST FIXTURES

- C-1 [Boeing Open-Hole Compression \(ASTM D 6484\)](#)
- C-2 [Northrop Open-Hole Compression \(NAI-1504C\)](#)
- C-3 [Boeing Compression After Impact \(ASTM D 7137\)](#)
- C-4 [NASA Compression After Impact \(NASA 1092\)](#)

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- D-1 [Three & Four Point Flexure \(ASTM D 790, D 6272 and D 7264\)](#)
- D-2 [Long Beam Flexure \(C 393\)](#)
- D-3 [Fixed-Span Long Beam Flexure \(C 393\)](#)
- D-4 [Ceramic Flexural Strength \(ASTM C 1161\)](#)
- D-5 [Ceramic Equibiaxial Flexural Strength \(ASTM C 1499\)](#)

## SECTION E: TENSILE LOADING [Back to Top](#)

- E-1 [Standard Tensile Wedge Grips](#)
- E-2 [Simple Tensile Wedge Grips](#)
- E-3 [Tensile Wedge Grip Inserts](#)
- E-4 [Specialized Tensile Testing Grips](#)
- E-5 [Line Grips for Thin Sheeting](#)
- E-6 [Split Capstan Grips](#)
- E-7 [Briquet Tensile Grips](#)
- E-8 [Split Collar Grips](#)
- E-9 [Adhesive Bond Tensile Grips](#)
- E-10 [Universal Joints](#)
- E-11 [Adapters, Lock Rings, Pins](#)

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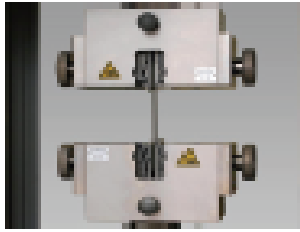

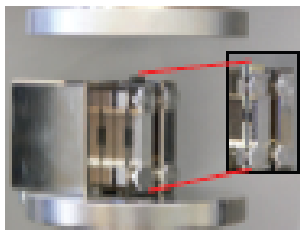
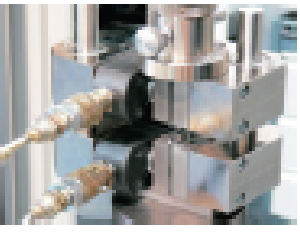
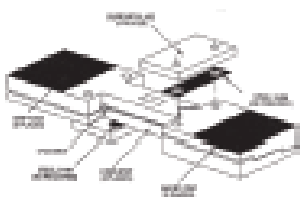

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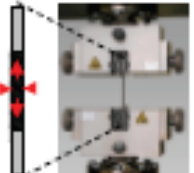

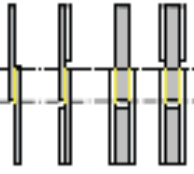

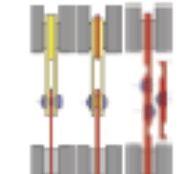
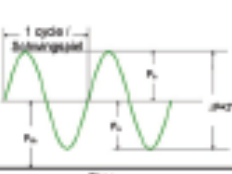

- H-1 [Fastener Bearing Specimen Support \(ASTM D 5961\)](#)
- H-2 [Laminate Bearing Strength – SACMA](#)
- H-3 [Laminate Bearing Strength – ASTM D 5961](#)
- H-4 [Plastic Bearing Strength – ASTM D 953](#)
- H-5 [Fastener Double Shear \(MIL-STD-1312-13\)](#)
- H-6 [Three-Plate Shear \(Fed. Test 406, Method 1041\)](#)
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- I-1 [Climbing Drum Peel \(ASTM D 1781\)](#)
- I-2 [Roller Drum Peel \(ASTM D 3167\)](#)
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- I-4 [Block-Shear of Adhesive Bonds \(ASTM D 4501\)](#)
- I-5 [Lapped Block Shear of Adhesive Bonds \(ASTM D 905\)](#)
- I-6 [Weld Shear \(ASTM A 497 & A 185\)](#)



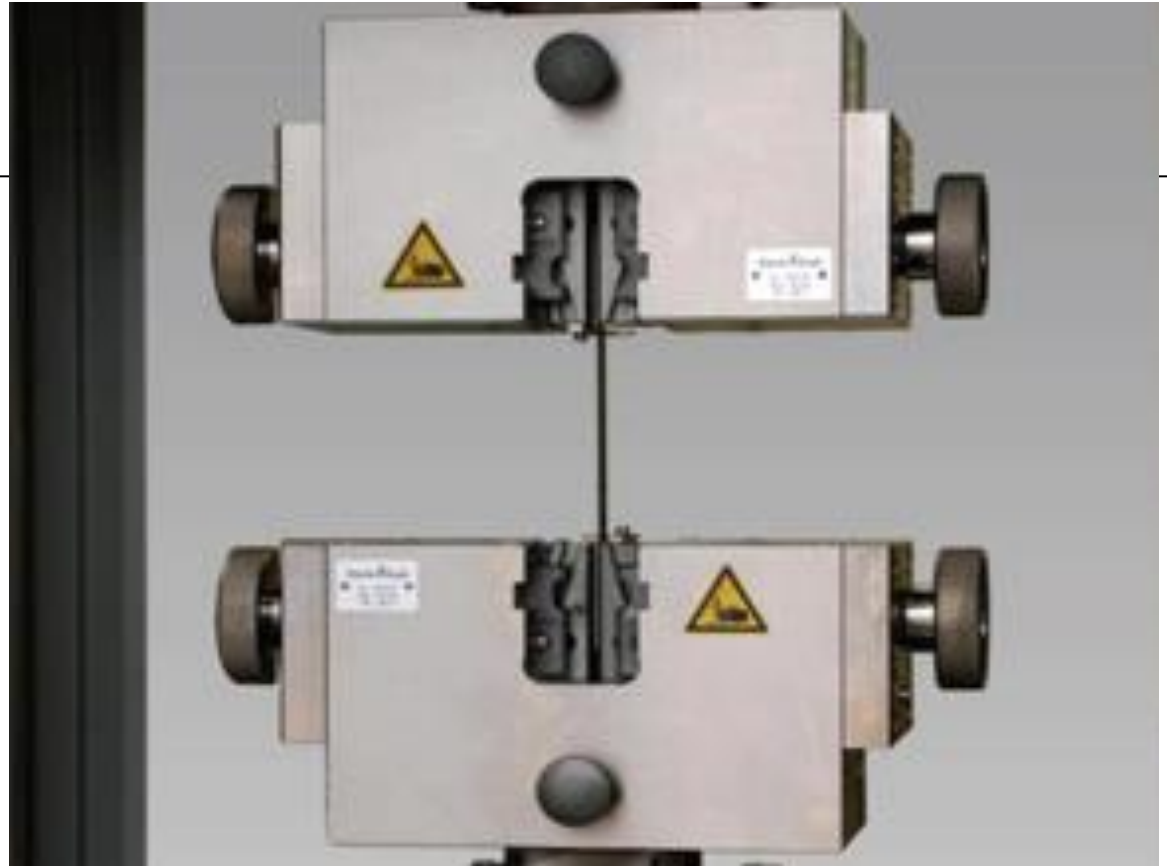
Prinzip	Prüfart	Normenbeispiele	Aussage
	Zugversuch	ASTM D 3039, EN 2581, EN 2597, ISO 527 Teil 4 und Teil 5, DIN 675378, Airbus AITM 1-0007, Boeing BSS 7320, SACMA SRM 4 und SRM 9  Für Filamentstränge: ASTM D 4018, ASTM D 3918, ISO 11588	Zugeigenschaften wie Zugmodul, Zugfestigkeit und Bruchdehnung, Poissonsche Zahl an flachen Probekörpern, Messungen an Filamentsträngen. Bei unidirektionalen Laminaten auch längs und quer zur Faserrichtung.
	Kerbzugversuch (open hole / bolted hole)	ASTM D 5786, ASTM D 6742, prEN 6035, Airbus AITM 1.0007	Beurteilung des Schädigungsmerkmals.
	Druckversuch mit stirnseitiger Krafteinleitung (end loading)	ASTM D 695 (modifiziert), prEN 2850, ISO 14126, AITM 1-0008, Boeing BSS 7260 - type III and IV	Druckmodul, Druckfestigkeit, Druckstauchung, Versagensart.
	Druckversuch mit flachseitiger Krafteinleitung (Shear loading / combined loading)	ASTM D 3410, ASTM D 6641, prEN 2850, ISO 14126, Airbus AITM 1-0008	Druckmodul, Druckfestigkeit, Druckstauchung, Versagensart  Bei dieser Prüfmethodik werden die Spannungskonzentrationen an den Probenenden vermieden und die Führung des Probekörpers ist besser als bei
	Kerbdrukversuch (open hole / bolted hole)	ASTM D 6484, ASTM D 6742, prEN 6036, Airbus AITM 1-0008, Boeing BSS 7260 - Type 1	Beurteilung des Schädigungsmerkmals.
	Interlaminarer Scherversuch, Kurzbiegemethode	ISO 14130, ASTM D 2344, EN 2377, EN 2563	Scheinbare interlaminare Scherfestigkeit.  Bei dieser Prüfmethode wirken starke Flächenpressungen an der

	<b>Schubversuch in Lagenebene (± 45° Schubversuch)</b>	ISO 14129, prEN 6031, ASTM D 3518, AITM 1-0002, DIN 65466	Schubmodul, Schubspannungen und -festigkeiten, Schubverformung
	<b>V-Kerb Schubversuch (Iosipescu)</b>	ASTM D 5379	Schubmodul, Schubspannungen und -festigkeiten, Schubverformung.
	<b>Scherung in Lagenebene</b>	EN 2243-1, EN 2243-6, prEN 6060, AITM 1-0019, DIN 65148, ASTM D 3914	Scherfestigkeit und Schubverformung zwischen den Lagen. Diese Prüfung wird auch bei Verklebungen angewandt.
	<b>Dreipunkt und Vierpunkt Biegeversuch</b>	EN 2562, EN 2746, ASTM D 7264, ASTM D 790, ISO 14125, ASTM D 6272	Biegeeigenschaften wie Biegemodul, Biegespannungen und Biegefestigkeit
	<b>Lochleibung</b>	ASTM D 5961, ASTM D 7248, Airbus AITM 1.0009, prEN 6037, DIN 65562	Charakterisierung einer Bolzen-, Niet-, oder Schraubverbindung hinsichtlich Tragfähigkeit und Lochleibung.
	<b>Dynamisch zyklische Beanspruchung</b>	ISO 13003, ASTM D 3479, ASTM D 6873	Charakterisierung der Ermüdung, Lebensdauer, Versagensart.
	<b>Compression After Impact (CAI)</b>	ISO 18352, prEN 6038, ASTM D 7137, DIN 65561, AITM 1-0010, Boeing BSS 7260 - type II	Beurteilung der Schädigung einer Prüfplatte anhand des Druckfestigkeitsverlustes. Die Schädigung wird durch eine definierte schlagförmige Beanspruchung üblicherweise mit einem instrumentierten Fallwerk aufgebracht.

# Test Standards Used

## Tensile Test

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Zugeigenschaften wie Zugmodul, Zugfestigkeit und Bruchdehnung, Poissonsche Zahl an flachen Probekörpern, Messungen an Filamentsträngen. Bei unidirektionalen Laminaten auch längs und quer zur Faserrichtung.



ASTM D 3039, EN 2561, EN 2597, ISO 527 Teil 4 und Teil 5, DIN 675378, Airbus AITM 1-0007, Boeing BSS 7320, SACMA SRM 4 und SRM 9

Für Filamentstränge: ASTM D 4018, ASTM D 3916, ISO 11566

## **Short Presentation of CCEV + personal activities**

- 1. Structural Development, Design Requirements, and Design Verifications**
- 2. Dimensioning Load cases, Safety Concept and Design Factors of Safety**
- 3. Modelling of Composites (elasticity, strength)**
- 4. Material Strength Failure Conditions (SFC)**
- 5. Application of SFCs to Some Materials**
- 6. Lifetime Prediction**
- 7. Material Properties**
- 8. Model Parameters**
- 9. Standardized Material Test Methods**
- 10. Structural Testing, NDI, Damage Tolerance**
- 11. Structural Verification, Reserve Factor**

# Material & Structural Testing and NDI

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## Materials Testing

**Structural Testing** (most often destructive testing)

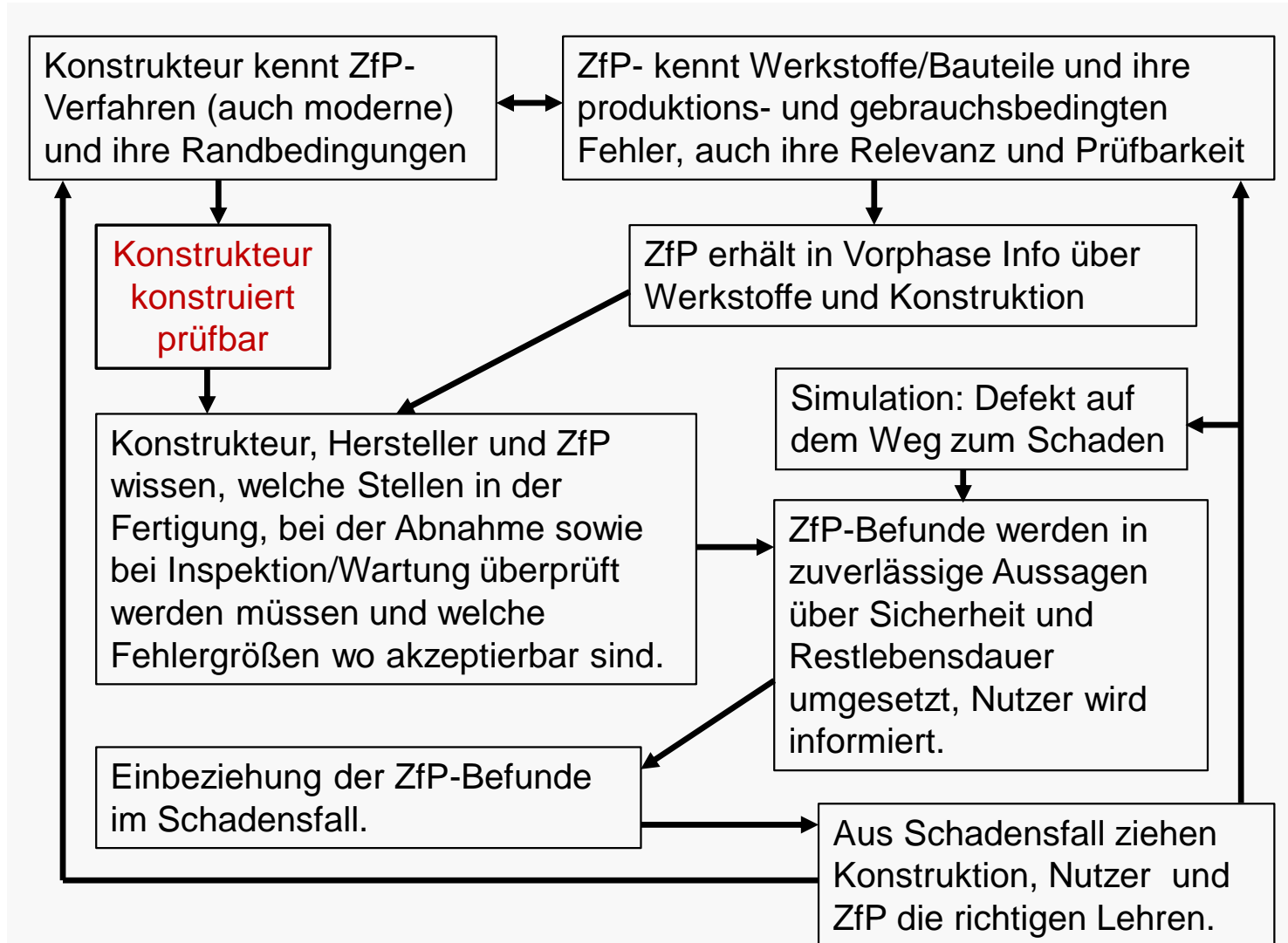
**Non-Destructive Testing (NDT, NDI, NDE),**

**NDI should be part of a  
“systems solution”**

- \* Failure: Detection, localization, sizing + shaping
- \* Failure: Assessment (*risk-based*)

# Non-Destructive Testing (Zerstörungsfreie Prüfung)

## Gerd Busse: Wunschtraum über Einbindung der ZfP



# Structural Testing (often destructive testing)

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# Structural Testing

(often destructive testing)



**ARIANE 5**  
**Front Skirt**

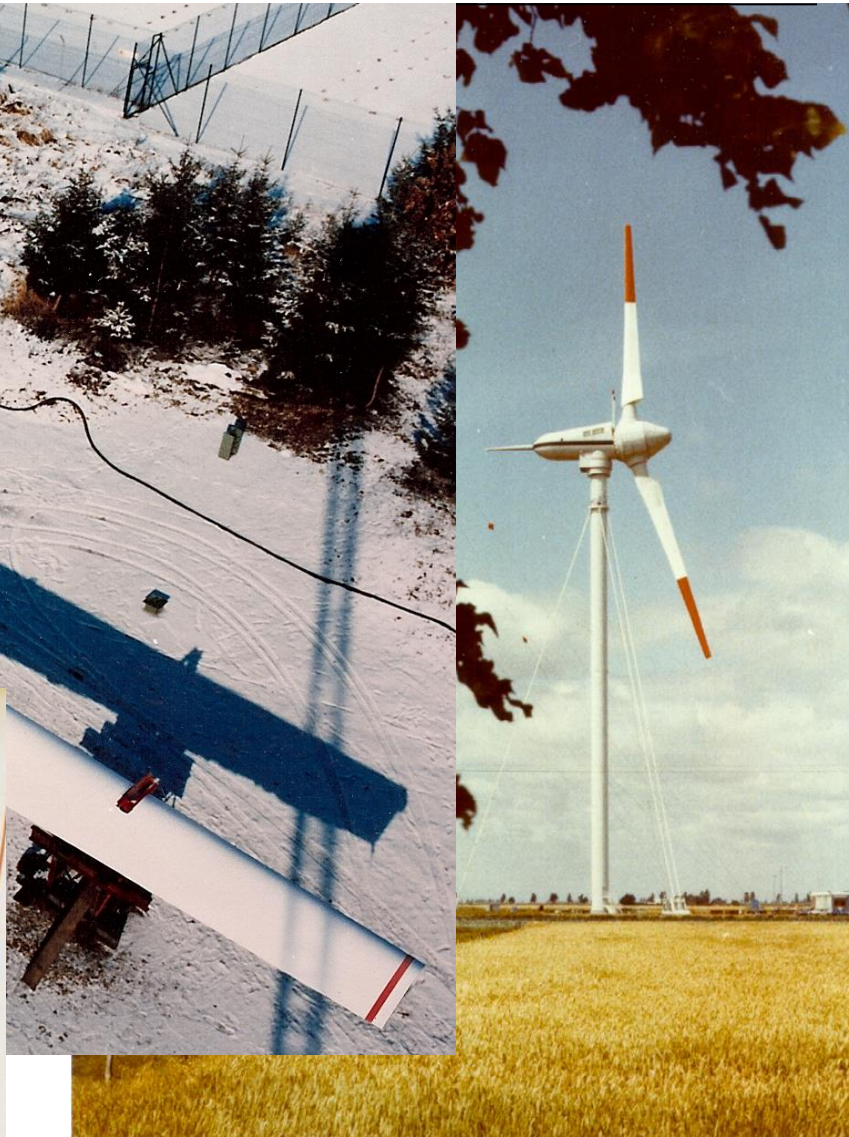
Lesson Learnt:

*Strain gages in the smooth strain regimes, only !*





# Structural Testing of GROWIAN



# Damage Threat Assessment for Composite Structure

**FAR 25.571 Damage Tolerance & Fatigue Evaluation of Structure** ... must show that catastrophic failure due to fatigue, corrosion, *manufacturing defects, or accidental damage* will be avoided through the operational life of the airplane.

## **Short Presentation of CCEV + personal activities**

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# Design Verification = Achievement of a Reserve against a Limit State

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For each distinct Load Case with its single Failure Modes must be computed:

**Reserve Factor** (is load-defined) :  $RF = \text{Failure Load} / \text{applied Design Load}$

**Material Reserve Factor** :  $f_{Res} = \text{Strength} / \text{Applied Stress}$

if linear analysis:  $f_{Res} = RF = 1 / Eff$

**Material Stressing Effort** :  $Eff = 100\%$  if  $RF = 1$  (Anstrengung)  
(Werkstoff-Anstrengung)

is applicable in linear and non-linear analysis.

# Conclusions w.r.t. Failure Mode Concept – derived Strength Failure Conditions

- The FMC is an efficient concept = may be viewed as **'Anisotropic Mises'**  
that improves prediction + simplifies design verification  
is applicable to brittle and ductile, dense and porous,  
isotropic, transversely-isotropic and orthotropic materials  
if clear failure modes can be identified and if the material element can be homogenized.  
**Formulation basis is whether the material element experiences  
a volume change, a shape change and friction .**  
*Builds not on the material but on material behaviour !*
- Delivers a **combined formulation** of *independent modal failure modes*,  
without the well-known drawbacks of **global** SFC formulations  
(which *mathematically combine in-dependent failure modes*) .
- The FMC-based Failure Conditions are simple but  
describe physics of each single failure mechanism pretty well.
- **Mapping of a brittle behaving isotropic porous foam and of a transversely-isotropic UD material was successful, thereby validating the SFC models.**

## Some Literature

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- [VDI2014] VDI 2014: German Guideline, Sheet 3 “*Development of Fiber-Reinforced Plastic Components, Analysis*”. Beuth Verlag, 2006 (*in German and English, author was convenor*).