

## Static Failure and Fatigue Failure of UD ply-composed Structural Parts (*novel treatment*)

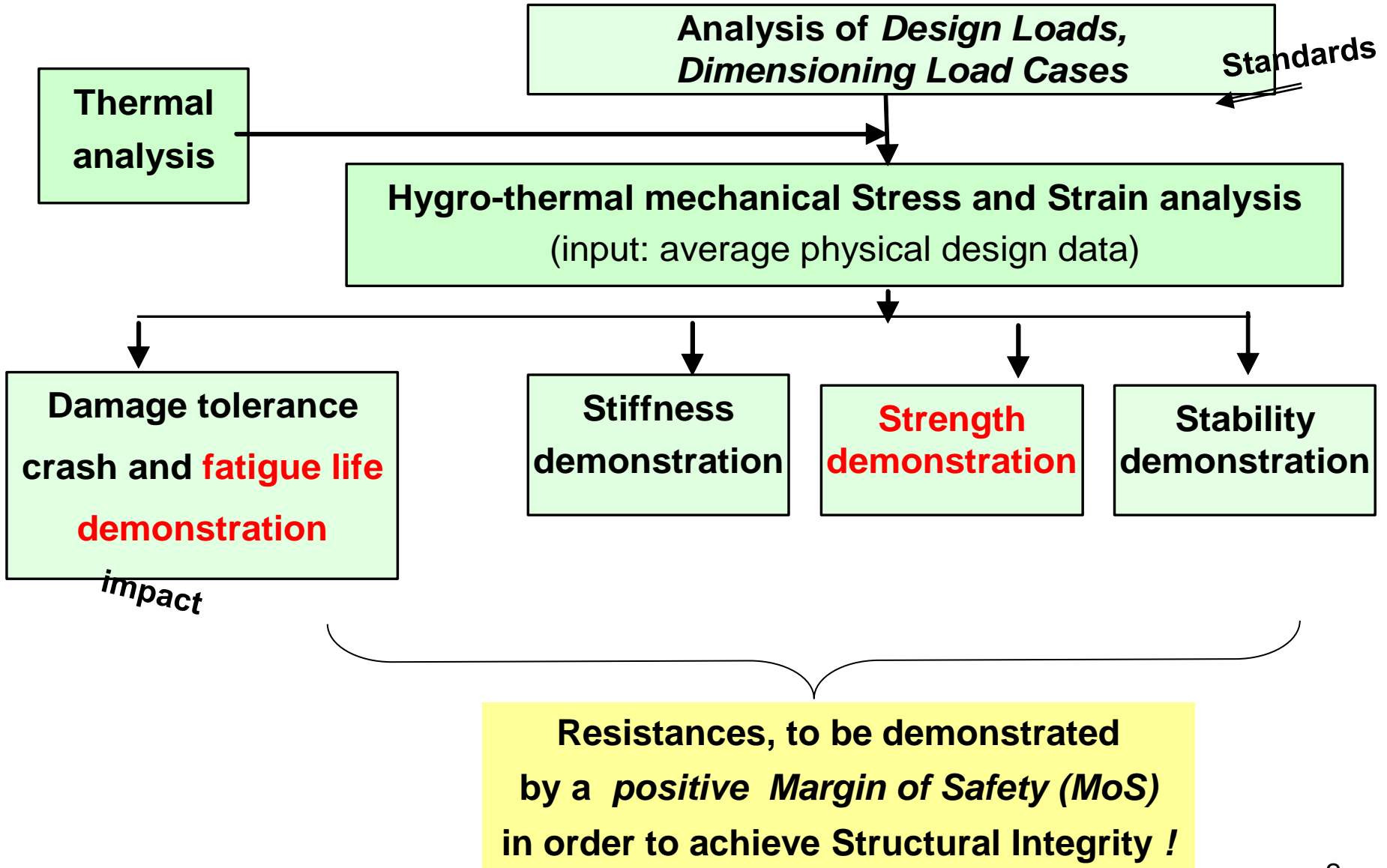
- 1 Introduction to Static and Fatigue Design
  - 2 Cuntze's Failure-Mode-Concept-based Strength Criteria
  - 3 Cuntze's Fatigue Life Estimation Method
  - 4 Generation and Interpretation of UD Haigh Diagrams
  - 5 Steps of the Fatigue Life Prediction Method Proposed
- Conclusions
- Example: Transversely-isotropic UD-CFRP

Results of a time-consuming, never funded „hobby“

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# Flow Diagram: Structural Design and Design Verification



# Verification Levels of the Structural Part

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- **Stress**, locally at a critical material 'point': **continuum mechanics, strength criteria**  
verification by a basic strength or a multi-axial failure stress state  
*Applied stresses are local stresses*
- **Stress concentration** at a notch (stress peak at a joint): **notch mechanics**  
verification by a *notch strength (usually Neuber-like, Nuismer, etc..)*  
*'Far'-field stresses are acting and are not directly used in the notch strength analysis*
- **Stress intensity** (delamination = crack): **fracture mechanics**  
verification by a *fracture toughness (energy-related)*  
*Applied stresses are 'far'-field stresses.(far from the crack-tip)*

*gilt statisch wie zyklisch*

# Design Verification by theoretical prediction

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## STATIC :

- Reserve Factor is load-defined :  $RF = \frac{\text{Predicted Failure Load}}{j \cdot \text{Design Limit Load}} > 1.$

*Material Stressing Effort* :  $Eff = 100\%$  if  $RF = 1$  (Anstrengung)

*Material Reserve Factor* :  $f_{Res} = \frac{\text{Strength Design Allowable}}{\text{Stress at } j \cdot \text{Design Limit Load}} > 1.$

if linear situation, then :  $f_{Res} = RF = 1 / Eff$

***Demonstration of  $MoS > 0$  or  $RF = MoS + 1 > 1$***

## CYCLIC :

$$RF_{life} \approx \frac{\text{Predicted Lifetime}}{j_{life} \cdot \text{Design Limit Lifetime}} > 1.$$

- Determination of Inspection time
- Determination of Replacement time

## Some Definitions *What is ??*

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**Material** : homogenized (smeared) model of the envisaged complex material which might be a material combination

**Failure** : structural part does not fulfil its functional requirements such as FF = fiber failure, IFF = inter-fiber-failure (matrix failure), leakage, deformation limit, delamination size limit, ...)

**Fatigue** : process, that degrades material properties

**Damaging** (not also damage, as used in English literature) : process wherein the results, the damaging portions, finally accumulate to a damage size such as a macro-scopic delamination.

The accumulation tool usually is *Miner's Damaging Accumulation Rule* (model)

**Damage** : sum of the accumulated damaging or an impact failure, that is judged to be critical. Then, *Damage Tolerance Analysis* is used to predict damage growth under further cyclic loading.

# State of the Art: Static Strength Analysis of UD laminas represent the results of the *World-Wide-Failure-Exercises*

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**Organizer :** *QinetiQ, UK* (Hinton, Kaddour, Soden, Smith, Shuguang Li)

**Aim:** *'Testing Predictive Failure Theories for  
Fiber-Reinforced Polymer Composites to the full !'*

*(was for the transversely-isotropic UD materials , only)*

**Method of the World-Wide-Failure-Exercises** (since 1991):

Part A of a WWFE: ***Blind Predictions on basic strength data***

Part B of a WWFE: ***Comparison Theory-Test*** with (reliable )  
*Uni-axial 'Failure Stress Test Data'* (= *basic strength*) and  
*Multi-axial 'Failure Stress Test Data'*

(plain test specimens, no notch)

# State of the Art : Cyclic Strength Analysis of UD Laminas (plies)

- **No Lifetime Prediction Method** available, applicable to any Laminate
- **Procedures base** – as with metals – on stress amplitudes and mean stress correction
- **Procedures base** on specific laminates and therefore cannot be generally applied
- **Presently: Engineering Approach:**
  - Static Design Limit Strain* of  $\epsilon < 0.3\%$  , negligible matrix-microcracking.  
Design experience proved: **No fatigue danger given**
- **Future : Design Limit Strain** shall be increased (EU-project: MAAXIMUS)  
*Beyond  $\epsilon \approx 0.5\%$  first filament breaks , diffuse matrix-microcracking changes to a discrete localized one .*

**Usually, *fiber-dominated laminates*  
are used in high-stress applications**

⇒

Fig. 1: Isolated (generates hardening curve) and embedded (softening curve part) of the UD material

**Isolated‘ lamina test specimens**

= weakest link results (series failure system)



**unconstrained lamina**

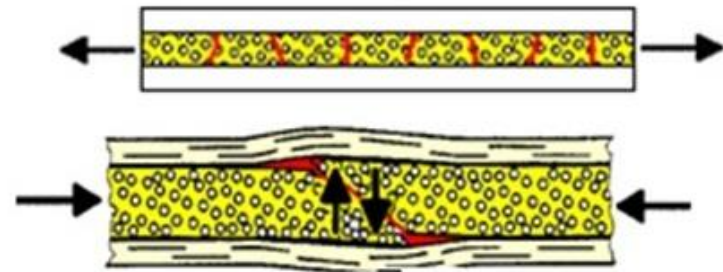
delivers strength property, stress-strain curve

(belongs to hardening)

delivers **basic strength**  
as analysis input !

**‘Embedded‘ laminas experience in-situ effects**

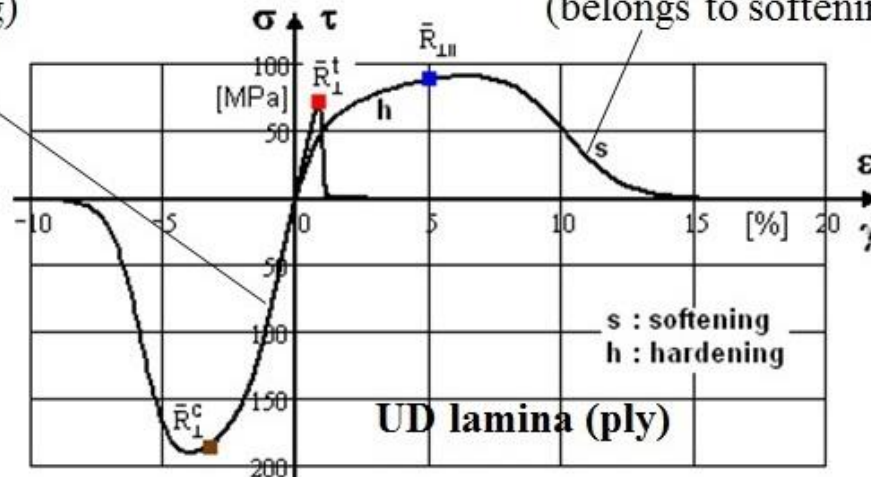
= redundancy result (parallel failure system)



**mutually constrained laminas, in laminates**

in non-linear laminate analysis

(belongs to softening)



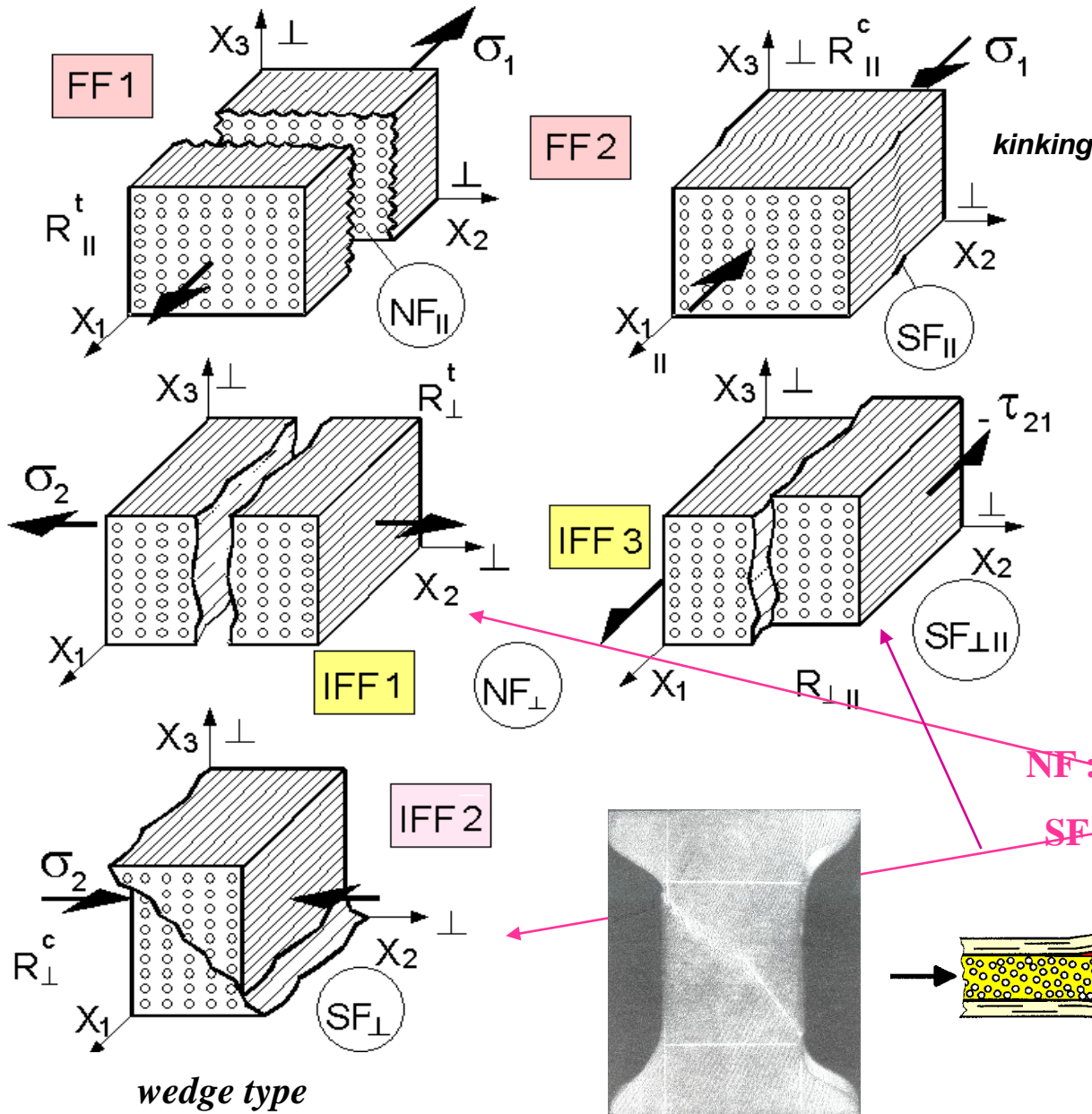


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# Observed Fracture 'Planes': Transversely-isotropic UD Material

[Cun04]

t = tension  
c = compression



- ▶ 5 Fracture modes exist
- = 2 FF (Fiber Failure)
- + 3 IFF (Inter Fiber Failure)

critical in a loaded laminate:  
FF1, FF2 + possibly IFF2 !

from these observations follow ..

NF := Normal Fracture

SF := Shear Fracture

## Basic Features of the author's Failure-Mode-Concept (FMC)

and a Confirmation that transversely-isotropic UD Materials exhibit a 5-fold material symmetry characteristic = 5 Strengths, 5 Failure Modes

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- Each failure mode represents 1 independent failure mechanism and thereby 1 piece of the complete *failure surface*
- Each failure mechanism is governed by 1 basic strength (is observed !)
- Each failure *mode* can be represented by 1 failure *condition*.

Therefore, *equivalent stresses* can be computed for each *mode* !!



Consequently, this separation requires :

***An interaction of all 5 Modal Failure Modes !***

# Cuntze's Set of Modal 3D UD Strength Failure Conditions (criteria)

**Cuntze = Mises under the UD criteria**

Invariants replaced by their stress formulations

FF1	$Eff^{\parallel\sigma} = \check{\sigma}_1 / \bar{R}_{\parallel}^t = \sigma_{eq}^{\parallel\sigma} / \bar{R}_{\parallel}^t,$	$\check{\sigma}_1 \cong \varepsilon_1^t \cdot E_{\parallel}^*$	strains from FEA	[Cun04, Cun11]
FF2	$Eff^{\parallel\tau} = -\check{\sigma}_1 / \bar{R}_{\parallel}^c = +\sigma_{eq}^{\parallel\tau} / \bar{R}_{\parallel}^c,$	$\check{\sigma}_1 \cong \varepsilon_1^c \cdot E_{\parallel}$		2 filament modes
IFF1	$Eff^{\perp\sigma} = [(\sigma_2 + \sigma_3) + \sqrt{(\sigma_2 - \sigma_3)^2 + 4\tau_{23}^2}] / 2\bar{R}_{\perp}^t = \sigma_{eq}^{\perp\sigma} / \bar{R}_{\perp}^t$			3 matrix modes
IFF2	$Eff^{\perp\tau} = [(\frac{\mu_{\perp\perp}}{1-\mu_{\perp\perp}}) \cdot (\sigma_2 + \sigma_3) + \frac{1}{1-\mu_{\perp\perp}} \sqrt{(\sigma_2 - \sigma_3)^2 + 4\tau_{23}^2}] / \bar{R}_{\perp}^c = +\sigma_{eq}^{\perp\tau} / \bar{R}_{\perp}^c$			3 matrix modes
IFF3	$Eff^{\perp\parallel} = \{[\mu_{\perp\parallel} \cdot I_{23-5} + (\sqrt{\mu_{\perp\parallel}^2 \cdot I_{23-5}^2 + 4 \cdot \bar{R}_{\perp\parallel}^2 \cdot (\tau_{31}^2 + \tau_{21}^2)}) / (2 \cdot \bar{R}_{\perp\parallel}^3)]\}^{0.5} = \sigma_{eq}^{\perp\parallel} / \bar{R}_{\perp\parallel}$			
	with $I_{23-5} = 2\sigma_2 \cdot \tau_{21}^2 + 2\sigma_3 \cdot \tau_{31}^2 + 4\tau_{23}\tau_{31}\tau_{21}$			

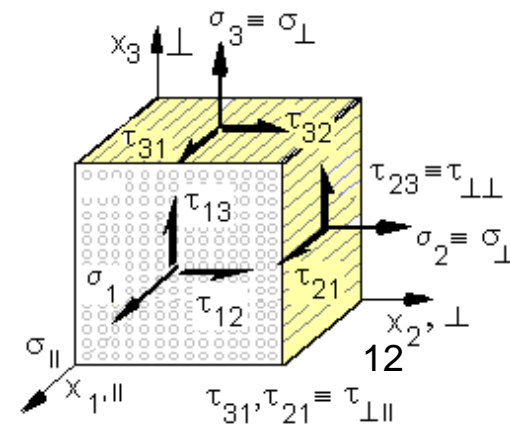
Modes-Interaction :

$$Eff^m = (Eff^{\parallel\tau})^m + (Eff^{\parallel\sigma})^m + (Eff^{\perp\sigma})^m + (Eff^{\perp\tau})^m + (Eff^{\perp\parallel})^m = 1$$

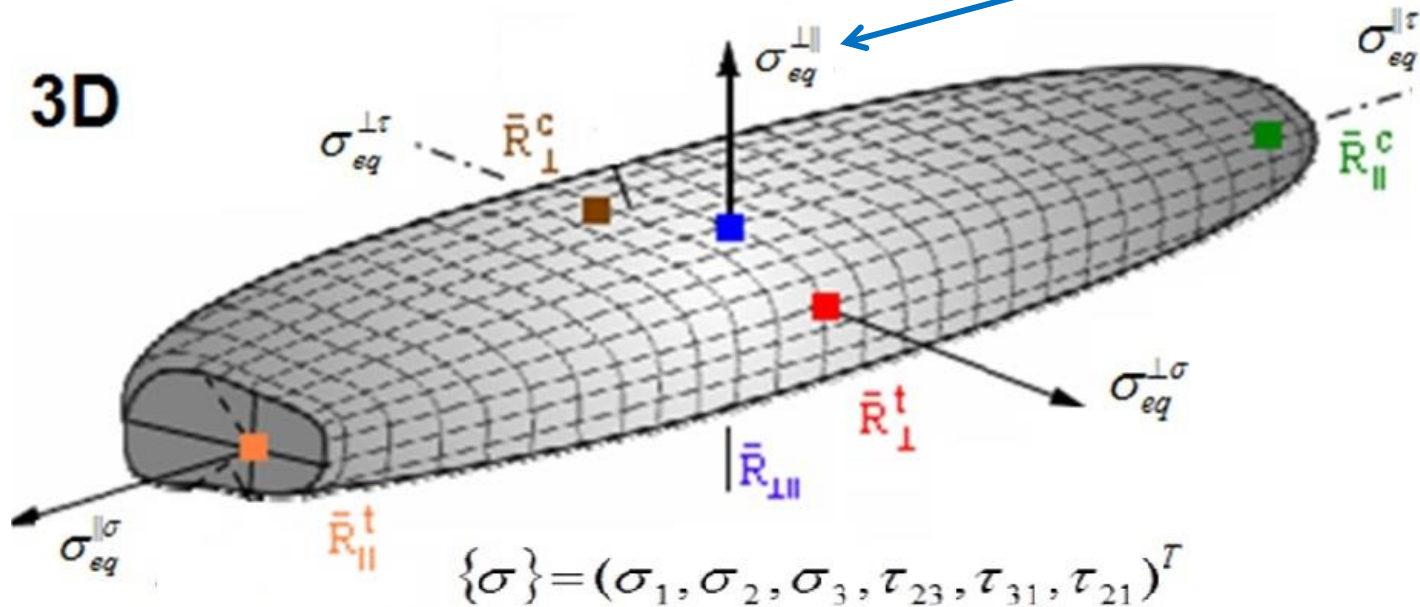
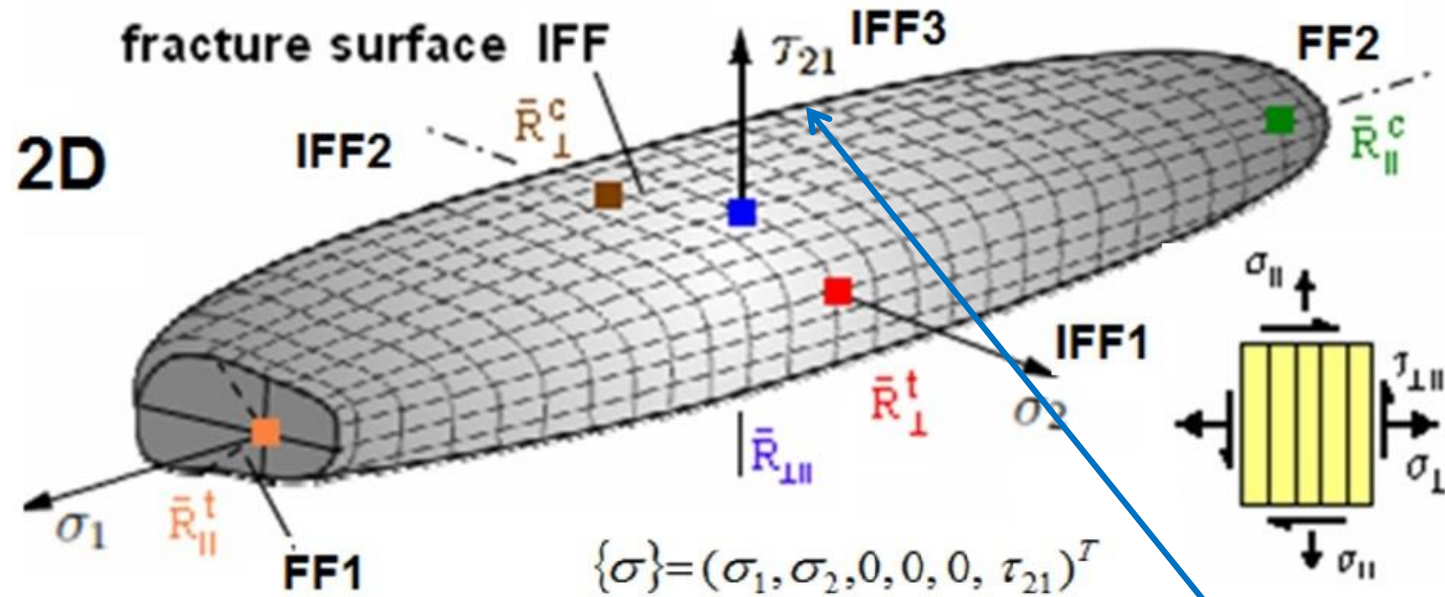
with mode-interaction exponent  $2.5 < m < 3$  from mapping tests data

Typical friction value data range:  $0.05 < \mu_{\perp\parallel} < 0.3, 0.05 < \mu_{\perp\perp} < 0.2$

Poisson effect \* : bi-axial compression strains the filament without any  $\sigma_1$   
 t:= tensile, c: = compression, || := parallel to fibre, ⊥ := transversal to fibre



# 2D = 3D Fracture surface by replacing the stress by the equiv. stress

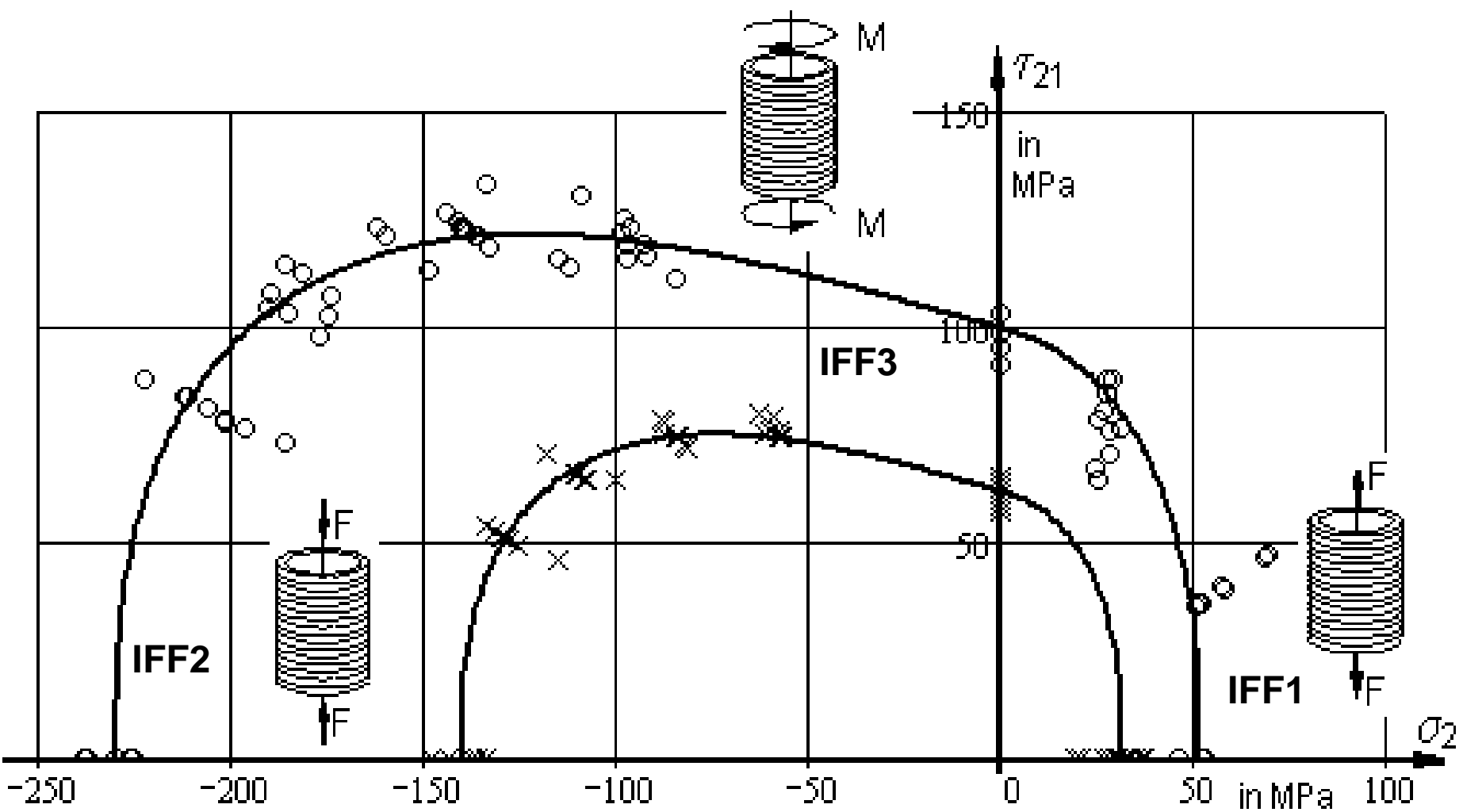


# Self-explaining, symbolic Notations for Strength Properties

		Fracture Strength Properties									
loading		tension			compression			shear			
direction or plane		1	2	3	1	2	3	12	23	13	
9	general orthotropic	$R_1^t$	$R_2^t$	$R_3^t$	$R_1^c$	$R_2^c$	$R_3^c$	$R_{12}$	$R_{23}$	$R_{13}$	friction
5	UD	$R_{//}^t$ NF	$R_{\perp}^t$ NF	$R_{\perp}^t$ NF	$R_{//}^c$ SF	$R_{\perp}^c$ SF	$R_{\perp}^c$ SF	$R_{//\perp}$ SF	$R_{\perp\perp}$ NF	$R_{//\perp}$ SF	$\mu_{\perp\perp}, \mu_{\perp//}$
6	fabrics	$R_W^t$	$R_F^t$	$R_3^t$	$R_W^c$	$R_F^c$	$R_3^c$	$R_{WF}$	$R_{F3}$	$R_{W3}$	Warp = Fill
9	fabrics general	$R_W^t$	$R_F^t$	$R_3^t$	$R_W^c$	$R_F^c$	$R_3^c$	$R_{WF}$	$R_{F3}$	$R_{W3}$	$\mu_{W3}, \mu_{F3}, \mu_{WF}$
5	mat	$R_{1M}^t$	$R_{1M}^t$	$R_{3M}^t$	$R_M^c$	$R_{1M}^c$	$R_{3M}^c$	$R_M^\tau$	$R_M^\tau$	$R_M^\tau$	(UD, turned direction)
2	isotropic matrix	$R_m$ SF	$R_m$ SF	$R_m$ SF	deformation-limited			$R_M^\tau$	$R_M^\tau$	$R_M^\tau$	$\mu$
		$R_m$ NF	$R_m$ NF	$R_m$ NF	$R_m^c$ SF	$R_m^c$ SF	$R_m^c$ SF	$R_m^\sigma$ NF	$R_m^\sigma$ NF	$R_m^\sigma$ NF	$\mu$

NOTE: \*As a consequence to isotropic materials (European standardisation) the letter R has to be used for strength. US notations for UD material with letters X (direction 1) and Y (direction 2) confuse with the structure axes' descriptions X and Y. \*Effect of curing-based residual stresses and environment dependent on hygro-thermal stresses. \*Effect of the difference of stress-strain curves of e.g. the usually isolated UD test specimen and the embedded (redundancy) UD laminae.  $R_m$  := 'resistance maximale' (French) = tensile fracture strength (superscript t here usually skipped),  $R$ := basic strength. Composites are most often brittle and dense, not porous! SF = shear fracture

# IFF Cross-section of the Fracture Failure Body (surface)



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***„Fatigue is the black art,  
to produce financial black holes“***

[J. Draper]

# Schädigungstreiber bei sich duktil und spröd verhaltenden Werkstoffen

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- **Duktiles Werkstoffverhalten** (Beispiel: isotrope Metalle)

**1 Mechanismus = “Schubspannungsgleiten“**

passiert unter allen zyklischen Beanspruchungen:

*Zugspannungen, Druckspannungen, Schub- und Torsionsspannungen !*

*Deswegen kann dieser einzige Mechanismus ‘Schubspannungsbasiertes Gleiten‘ mit einer einzigen Fließbedingung beschrieben werden!*

- **Sprödes Werkstoffverhalten bei isotropen Werkstoffen**

**2** Schädigung erzeugende **Mechanismen** wirken

*(ingenieurmäßige Berücksichtigung durch sog. Mittelspannungskorrektur)*

- **Sprödes Werkstoffverhalten bei UD- Werkstoffen**

**5** Schädigung erzeugende **Mechanismen** wirken

*(Ansätze mit und ohne Mittelspannungskorrektur)*

# Was sind die benötigten zyklischen Größen?

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- Wöhlerkurven  $R = const = \sigma_{unter} / \sigma_{ober}$
- Schädigungsakkumulationshypothese
- Quantifizierte Schädigungs‘portionen‘ (-inkremente)

Dazu Anwendbarkeit der statischen Festigkeitshypothesen, wenn die

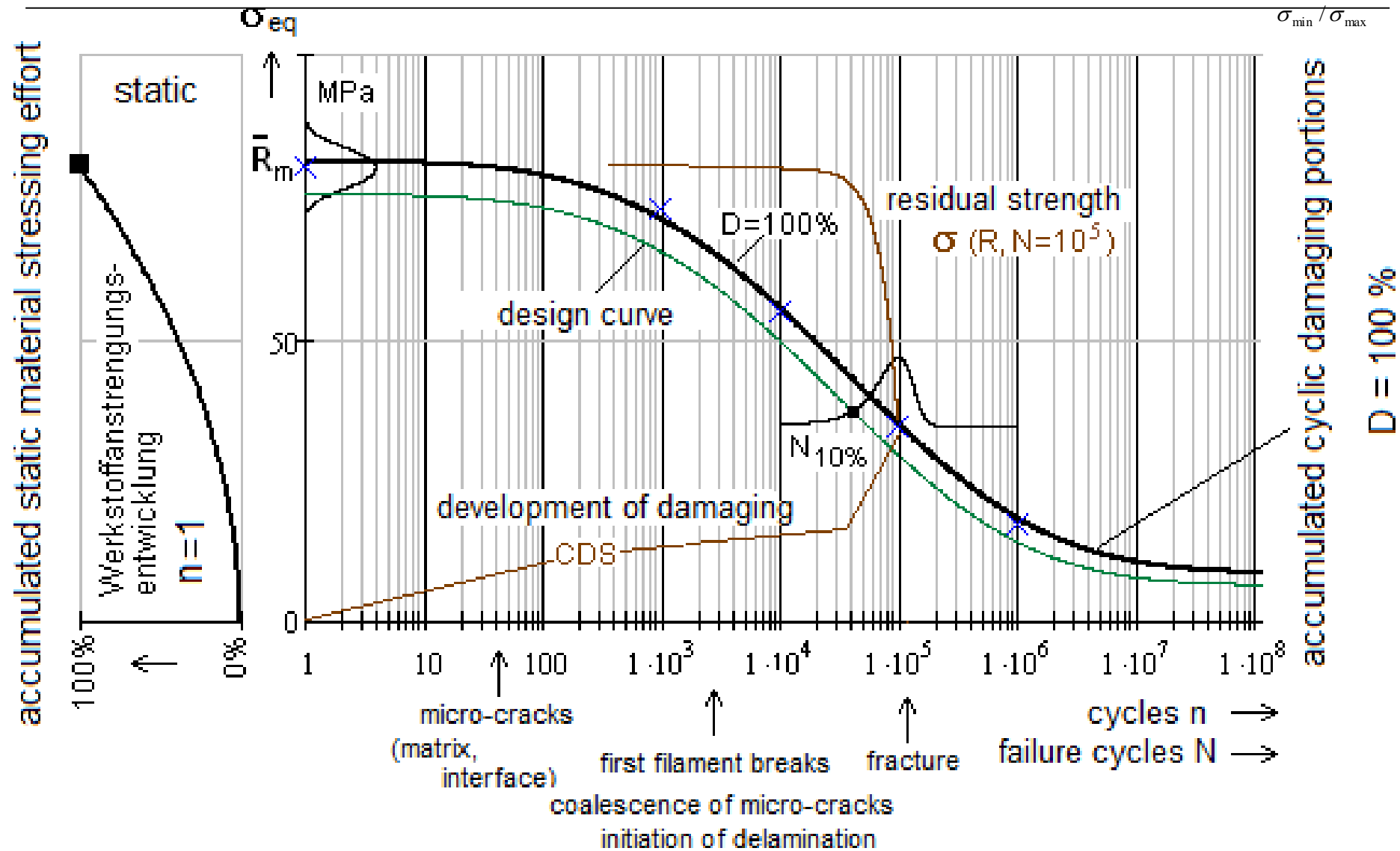
**Statischen Festigkeitswerte** durch  
**Restfestigkeitswerte für eine bestimmte Lebensdauer**  
ersetzt werden.

**Statische Anstrengungssumme  $Eff$  (material stressing effort)**

wird durch

**Zyklische Schädigungssumme  $D$**

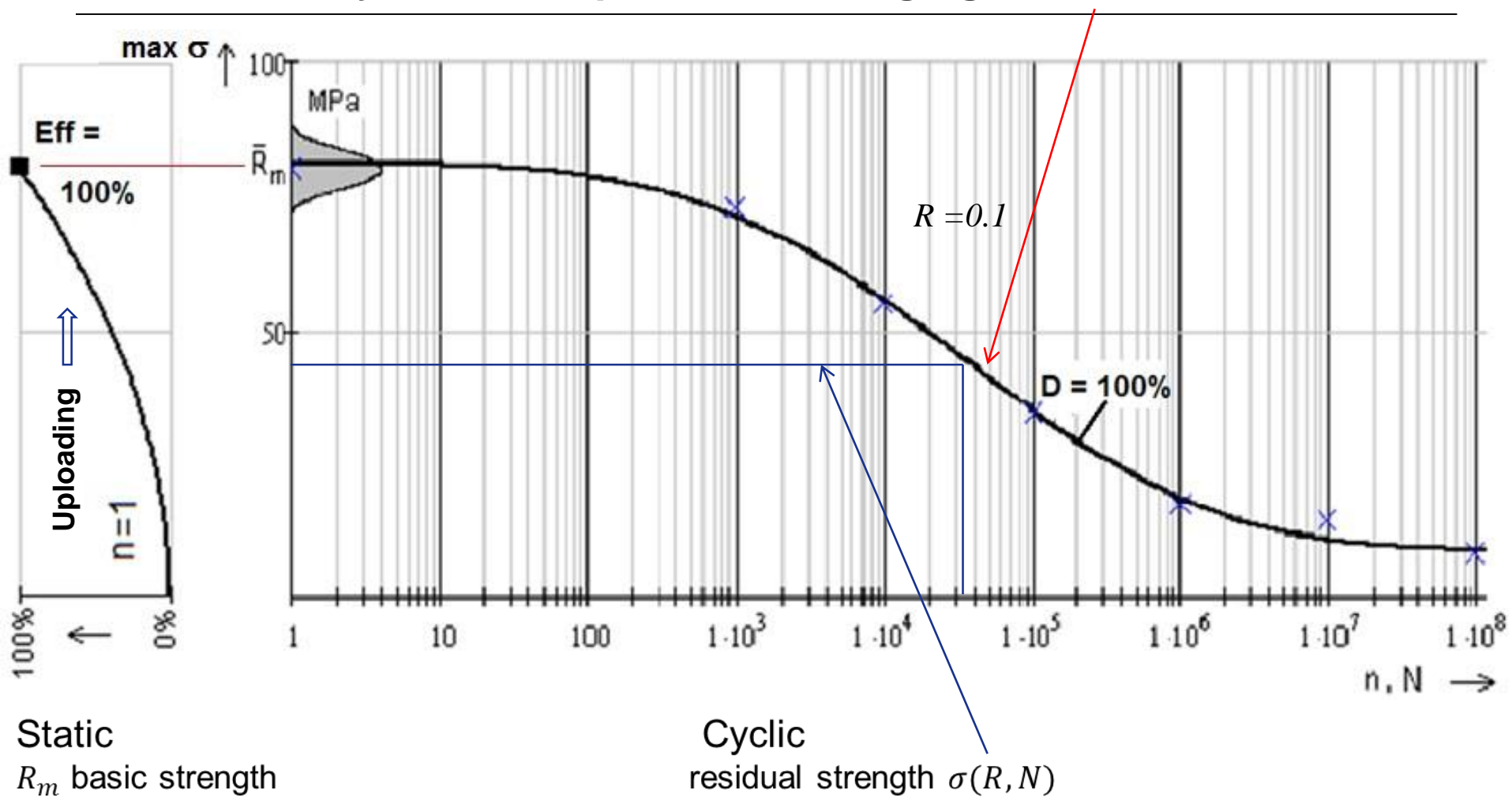
ersetzt !



FF:= fibre failure. IFF:= Inter Fibre Failure

CDS:= characteristic damage state at the end of diffuse damaging

# Static and cyclic development of damaging, S-N-curve



Analogous limits of the material capacities :

- Static : material stressing effort Eff = 100 %
- Cyclic : material damaging sum D = 100 %

# Parameter-Weibull-Ansatz – *allgemeine Form*

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$$\sigma_{\max}(n) = c_1 + (c_2 - c_1) / \exp\left(\frac{\log(N)}{c_3}\right)^{c_4}.$$

$$\log(N) = c_3 \cdot \left[ \ln\left(\frac{c_2 - c_1}{\sigma_a - c_1}\right) \right]^{1/c_4}$$

Deutsche akademische Forschergruppe ist seit fast 6 Jahren aktiv auf dem vereinbarten aussichtsreichsten gemeinsamen Nenner *physikalisch-basiert (Versagensmodi), schicht-orientiert.*

Ziel: Versagensmodus-basierte Lebensdauer-Vorhersagemethode

# Schritte bei der Lebensdauerabschätzung

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## 1 Input

**Betriebsbelastungen: Last-Zeit-Kurven** (Modellierung mit rain flow, ..)

**Sicherheitskonzept: Design to Life**  $j_{\text{Life}} = 3 - 4$

## 2 Übertragung der Betriebsbelastungen in Beanspruchungen (Spannungen) mittels Strukturanalyse)

## 3 Bereiche der Ermüdungsanalyse

**LCF: high stressing,**

**HCF: intermediate stressing**

**VHCF: low stressing and strains (SPP1466)**

## 4 Erfassung der Betriebsbelastung

**Zeitbereich:** Zyklus-für-Zyklus oder Kollektiv-für-Kollektiv (weniger Rechenaufwand)

**Frequenzbereich:** Lastspektren (Verlust der Last-Reihenfolge) oder Blockbelastungen, etc.

# Schädigungstreiber bei spröden zyklisch beanspruchten Composites

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**Annahmen:** Falls Versagensmechanismen(-modi) gleich?

- Dann auch die schädigungstreibenden Versagensparameter gleich.
- Übertragbarkeit statisches Versagen auf Ermüdung möglich,

*Dabei schädigen ebene (2D) und räumliche (3D) Spannungszustände*

**Meßbare Schädigungsgrößen:**

*Mikrorißdichte, Restfestigkeit, Reststeifigkeit*



# FMC-based UD Strength Failure Conditions, **Damaging drivers**

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- **Ductile material behaviour (e.g. many metals):**

- \* *Slip band shear yielding - as damaging driver - occurs under cyclic tensile stress, compressive stress, and under shear stress !*

- \* *Therefore, this single mechanism*

- shear stress–caused yielding can be principally described by*

- one yield failure condition to determine the needed damaging portions !*

- (Formulation is in normal stresses, but the shear stress is the damaging driver).*

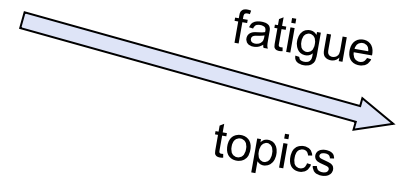
- \* *Increasing with brittleness, lifetime estimation is corrected by accounting for the ‘Mean stress effect’  $\sigma_{mean}$*

- (considers by Goodman Diagram that more mechanisms really act).*

## **Brittle material behaviour :**

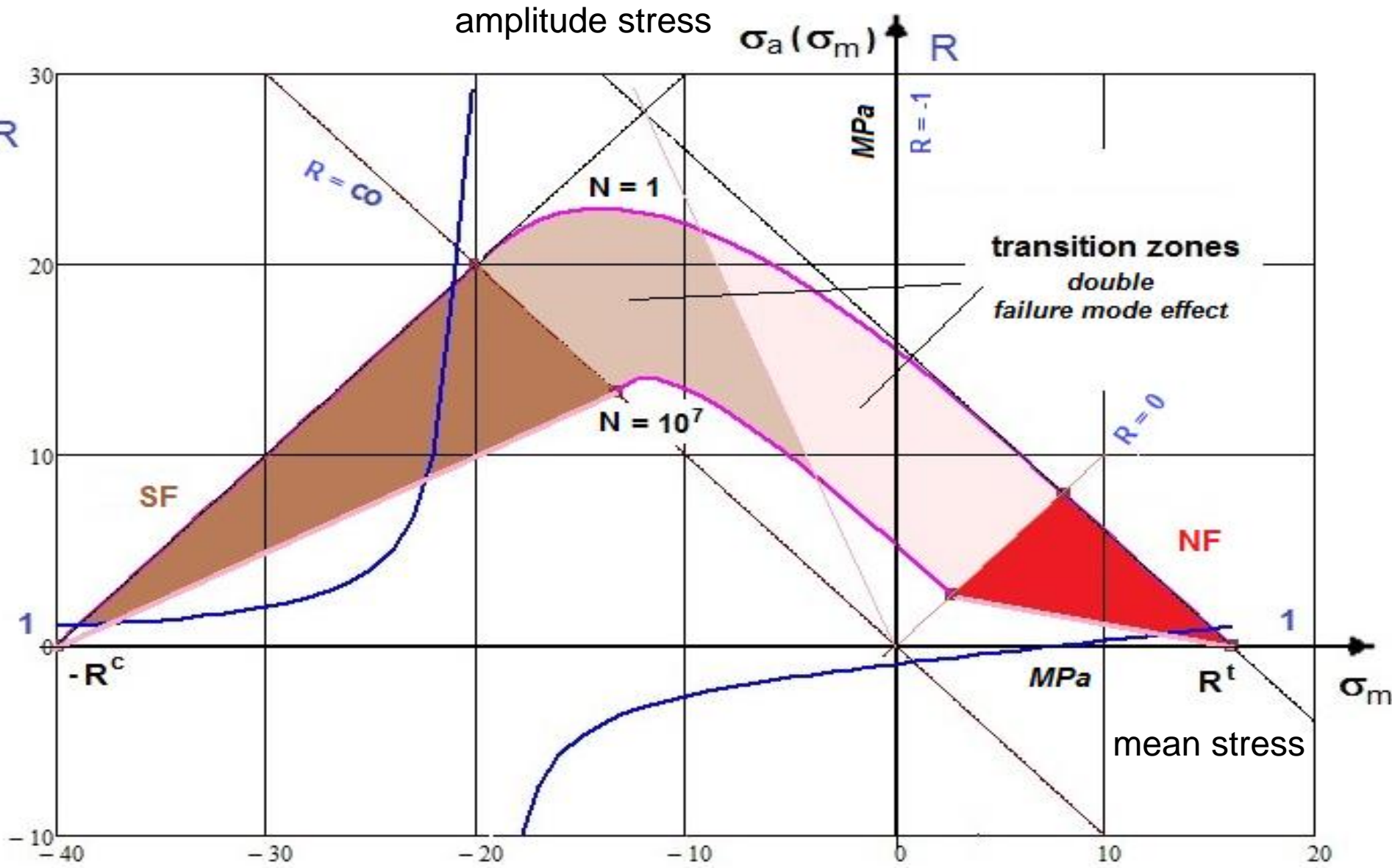
- \* *Many mechanisms, causing damaging, must be considered*

- Question is whether a correction by the ‘Mean stress effect’ makes sense.*



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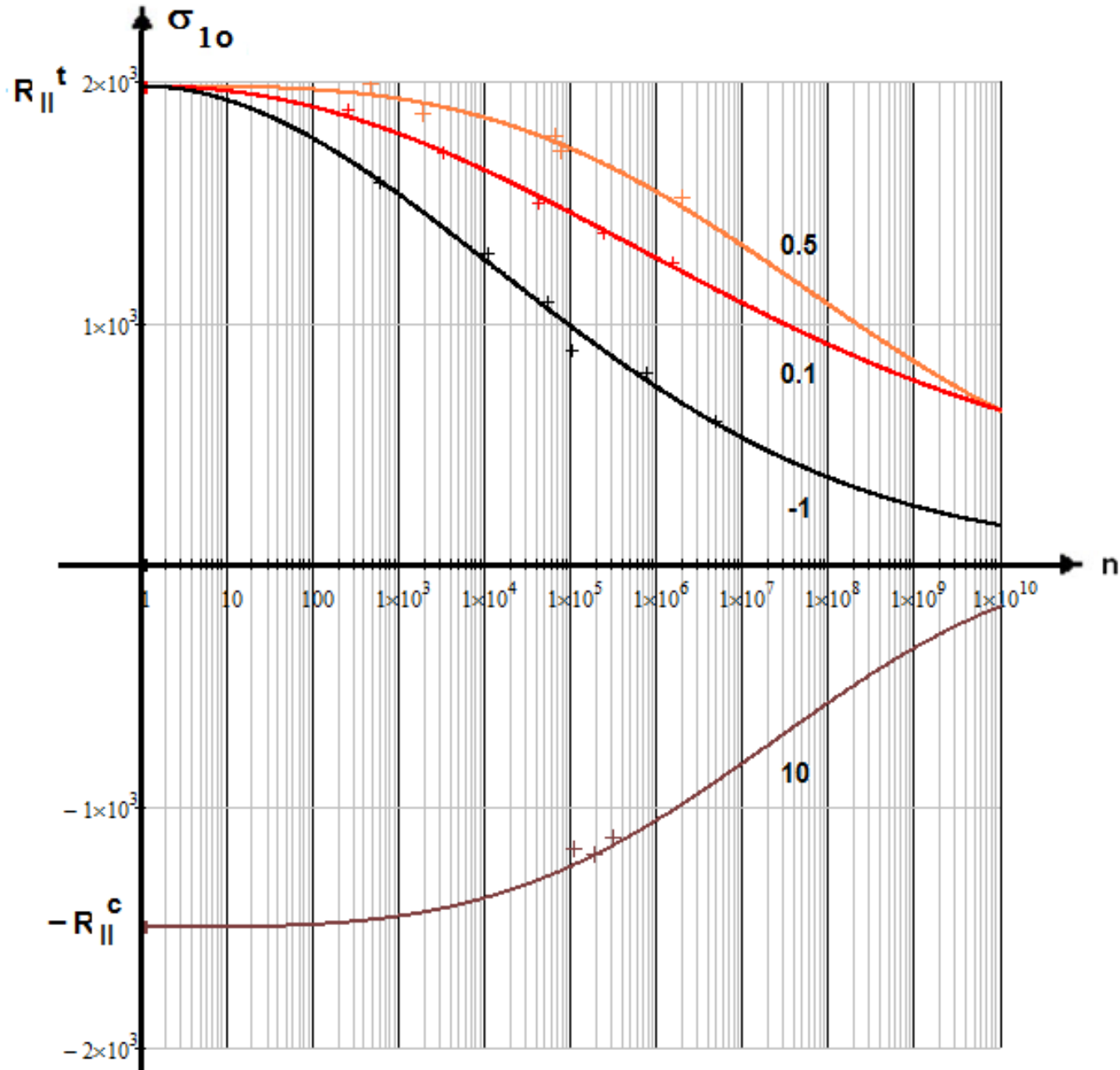
# Novel Haigh Diagram of a Brittle behaving Isotropic Material



$R :=$  stress ratio  $\sigma_{min}/\sigma_{max}$

NF = Normal Fracture, SF = Shear Fracture, N = fracture cycle number

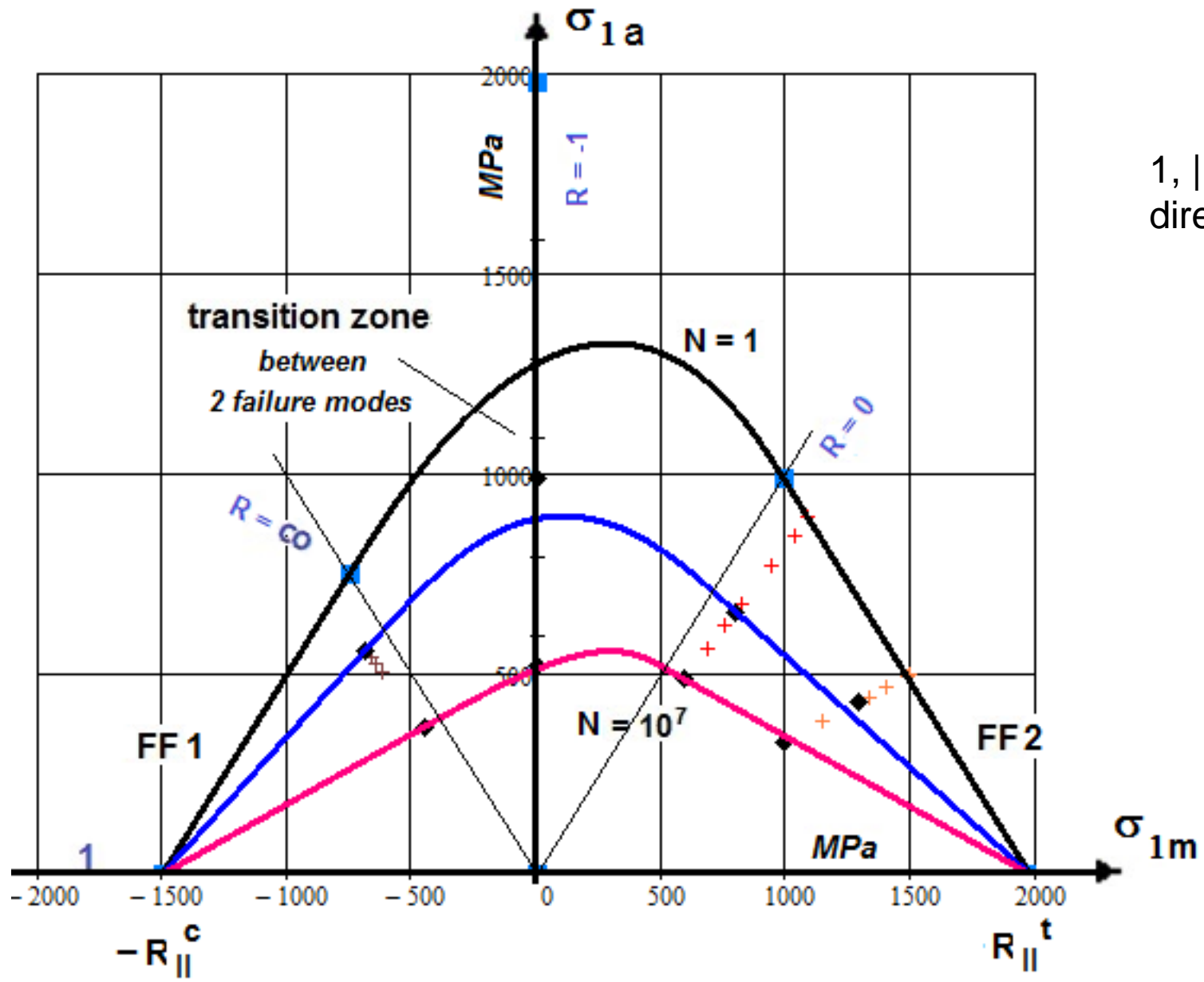
# Semi-log FF1-FF2-linked S-N curves [data, courtesy Kawai, Suda]



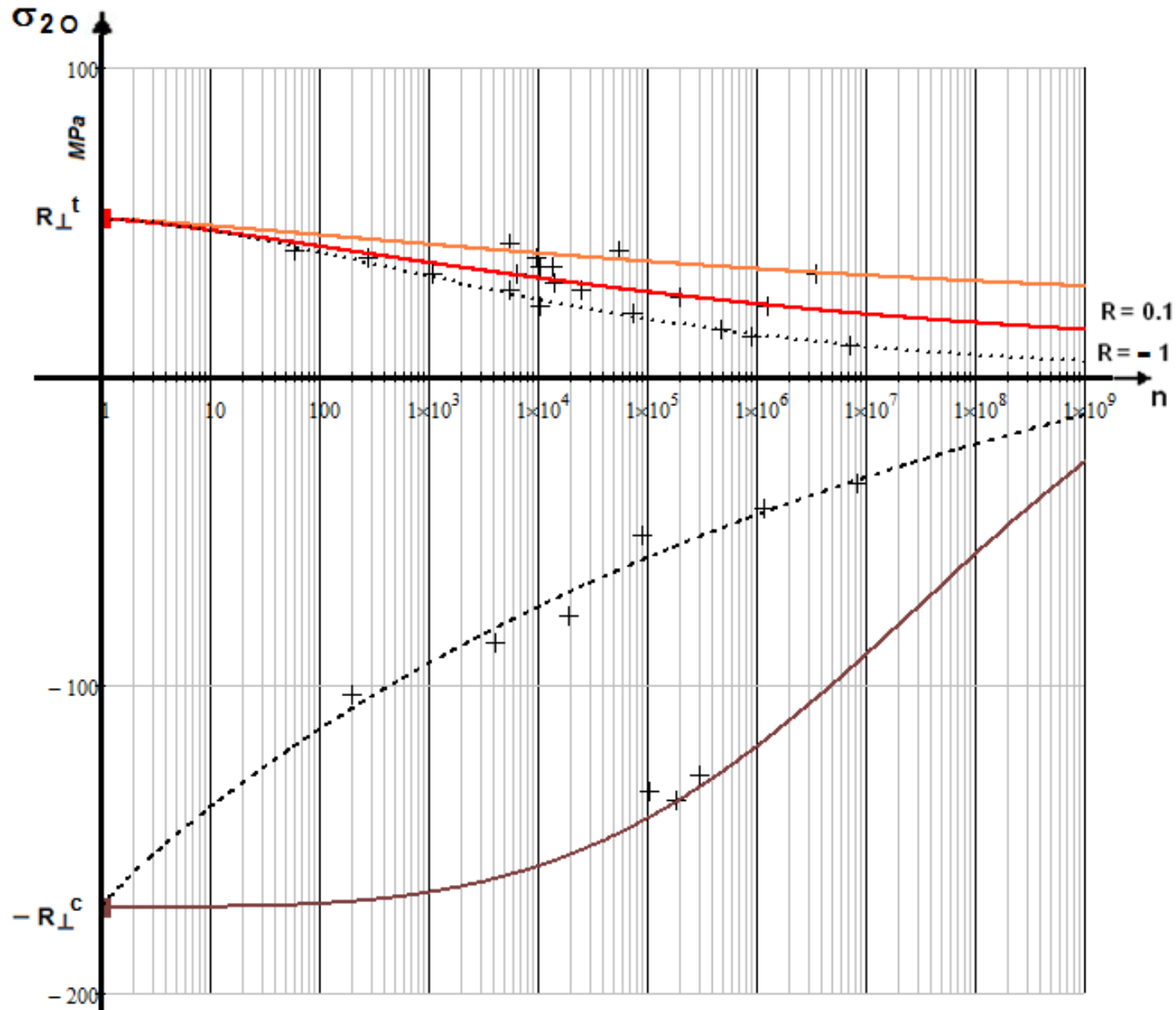
1, || = fiber direction

# FF1-FF2 Haigh diagram

displaying the failure mode domains, transition zone, test data [Hah14] and the **analytically** determined fix points for the predicted constant life curves



# Semi-log IFF1-IFF3-linked S-N curves [data, courtesy C. Hahne]



2,  $\perp$  =  
across fiber  
direction

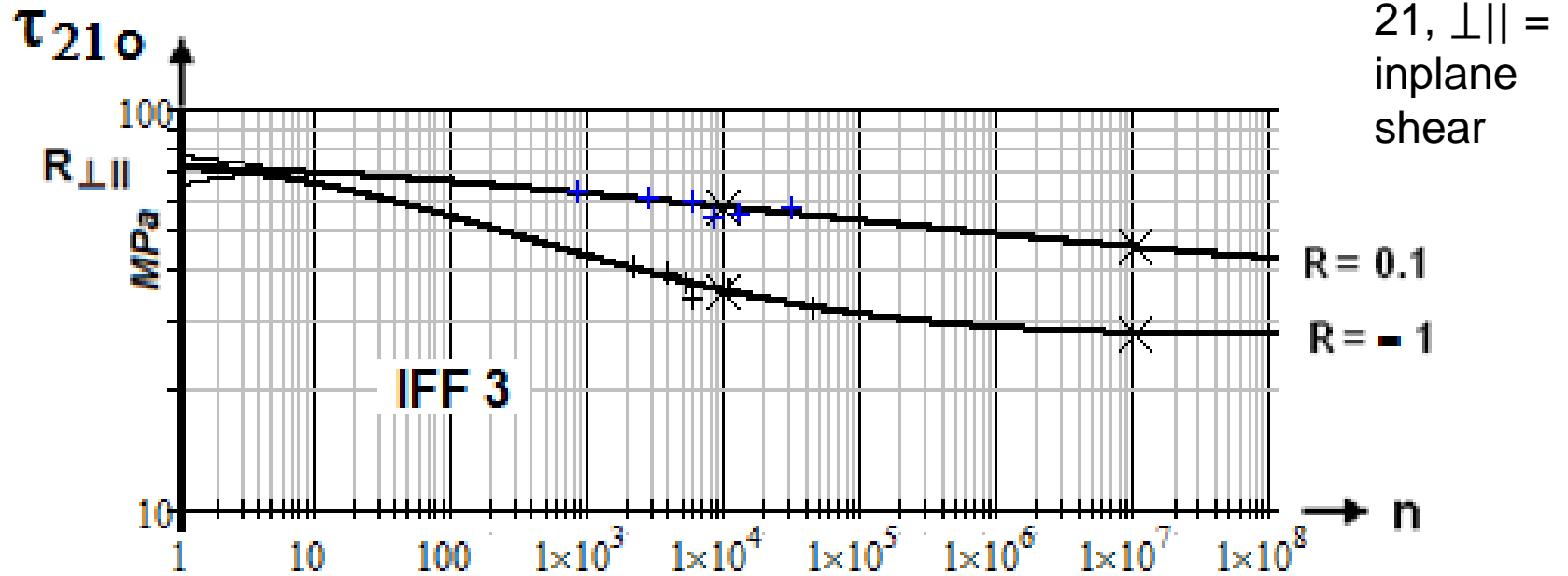
# IFF1-FF2 Haigh diagram

*displaying the failure mode domains, transition zone (test data [C. Hahne])*

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2,  $\perp$  =  
across fiber  
direction

# Log-log IFF3-linked S-N curves [data, courtesy C. Hahne]







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## Objectives of the Method

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An engineering, failure modes-linked lifetime prediction method for plain laminates which employs:

- 1.) Failure mode-linked determination of the cyclic loading
- 2.) Measurement of just a minimum number of the failure mode-linked representative *mode S-N curves* = master R-curve of each mode
- 3.) Prediction of other necessary stress-ratio '*mode S-N curves*' on basis of the measured mode master curve one (e.g. R=0.5 from R=0.1)
4. Determination of Damaging portions on basis of the static criteria
- 5.) Accumulation of Damaging Portions using Paömgren-Miner

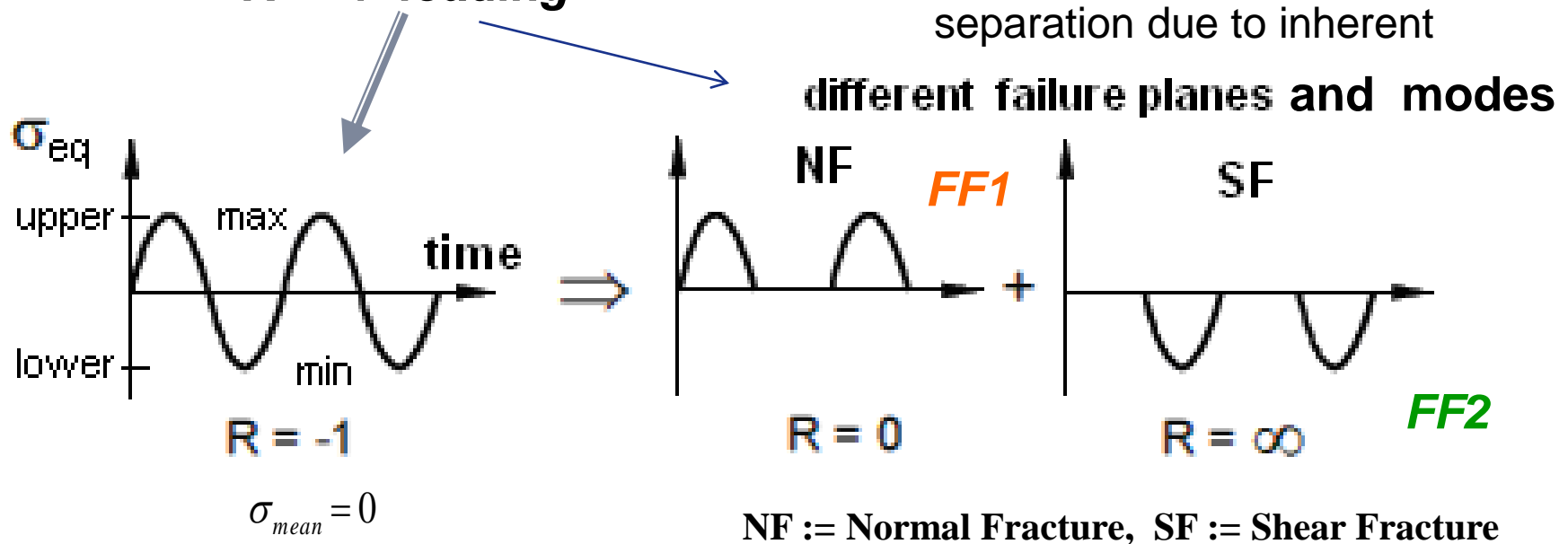
# Novel failure mode-wise modelling of Loading Cycles for

high-performance 'fiber-dominated designed', UD laminas-composed laminates

For simply displaying the **approach** *it is chosen* :

- the usually 'fiber-dominated' laminate and

-  $R = -1$  loading

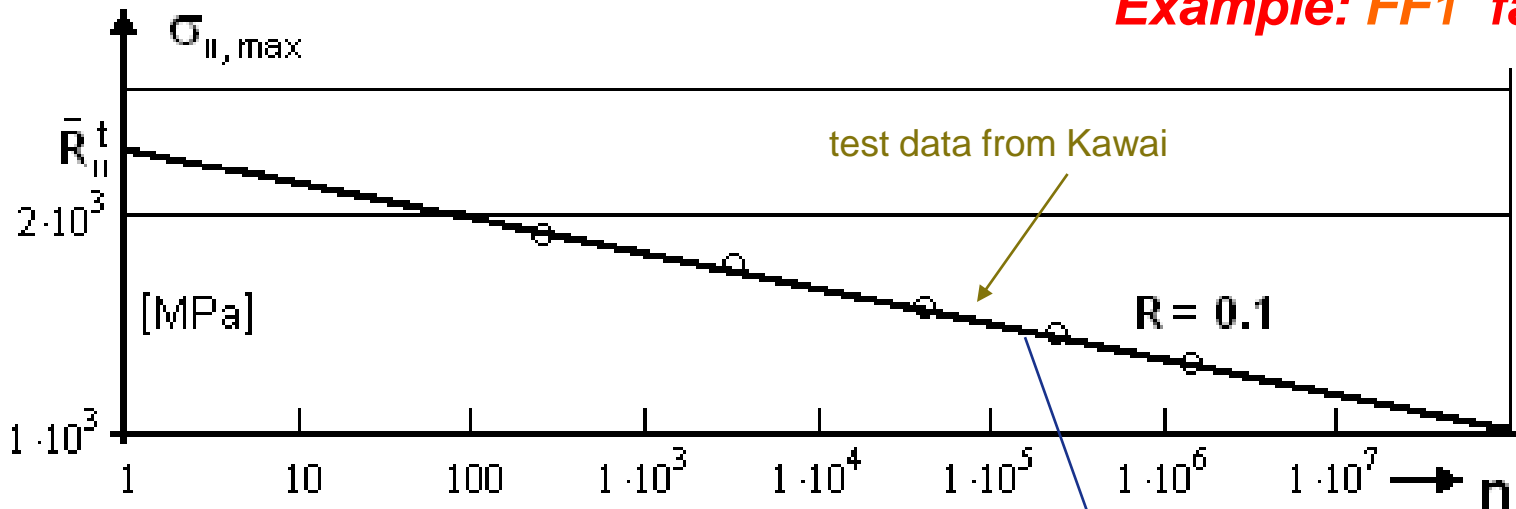


**Step 1 : Failure mode-linked apportionment of cyclic loading (novel)**

Specific **rain-fall** procedure to be applied,

# Mapping of S-N data and Mode-representative *Master S-N curve*

**Example: FF1 failure mode**



**Step 2 : S-N curve can be mapped by straight line in a *log-log* graph**

Measured curve used

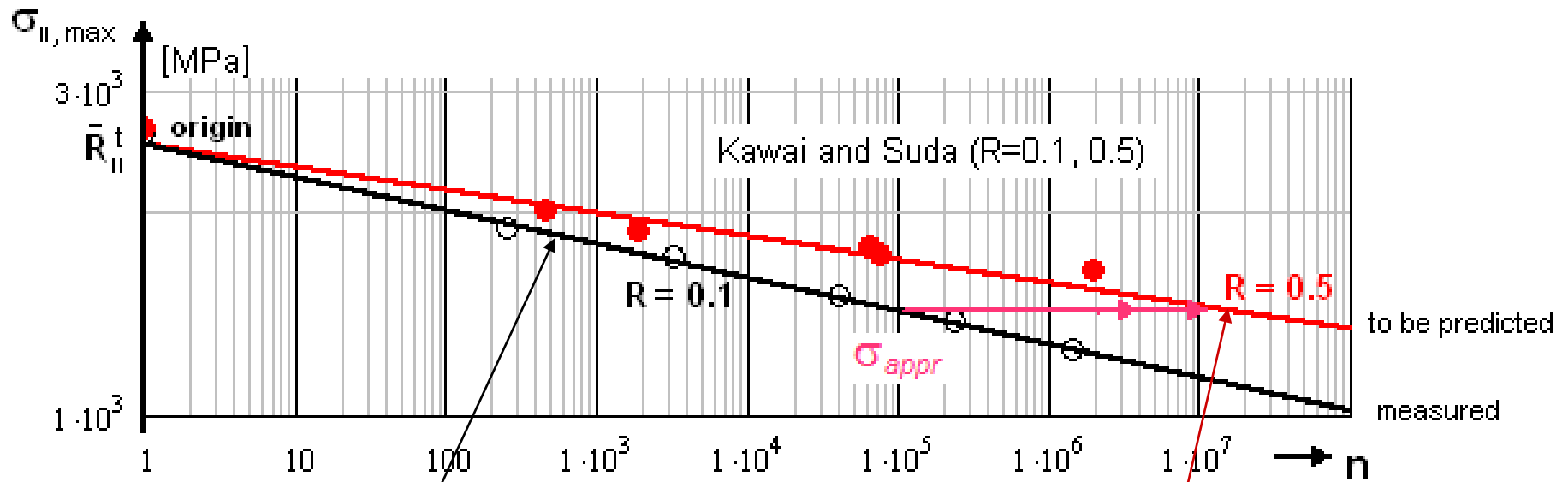
as mode-representative **Master S-N curve** for **FF1**

$$\sigma_{||, \max}^{Master}(n) \approx \bar{R}_{||}^t \cdot n^{C_{Master}}$$

FF1 strength

**In the general case of variable loading  $\implies$  Several S-N-curves are needed !**

# Prediction of needed other *FF1* S-N curves from Master *FF1* Curve



Given :  $\sigma_{||, \max}^{Master}(n) \approx \bar{R}_{||}^t \cdot n^{c_{Master}}$

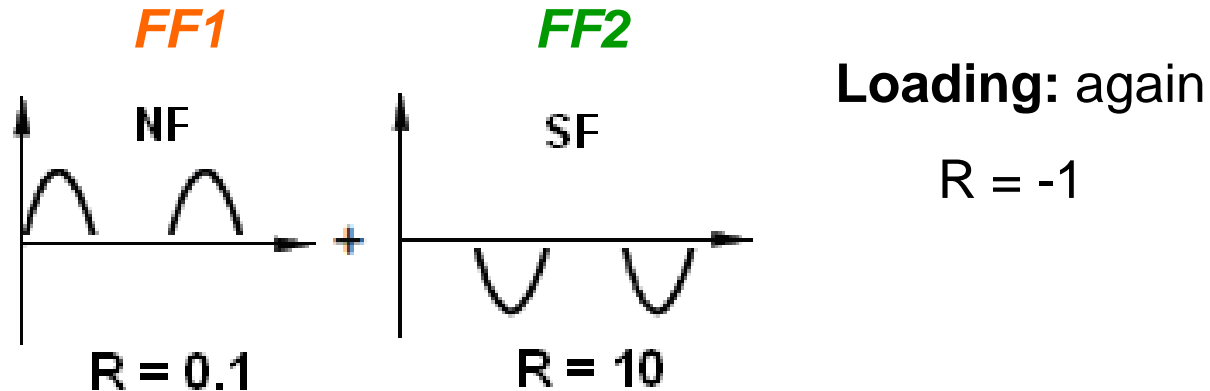
Searched :  $\sigma_{||, \max}^{pred} = \bar{R}_{||}^t \cdot n^{c_{pred}}$

Slope of **R = 0.5 ?**

**Step 3: Application of the Principle of constant strain energy equivalence**  
 A distinct strain energy level will be reached for  $R > 0.1$  at higher cycles.

$$\Delta W^{||\sigma} \cdot \bar{R}_{||}^t{}^2 = \sigma_{1, \max}^2 - \sigma_{1, \min}^2 = \sigma_{1, \max}^2 \cdot (1 - R^2)$$

# Application of Miner-‘Rule‘ *simple example*



$$D (FF1, FF2) = NF : (n_1 / N_1 + n_2 / N_2 + n_3 / N_3) + SF : (n_4 / N_4)$$

$$+ D (IFF1, IFF2, IFF3) = D \leq D_{feasible}$$

from test experience

**Step 4: Determination of Damaging Portions by Static Strength Criteria**

**Step 5: Mode-wise Accumulation of Damaging Portions (novel)**

# The presented Novel Lifetime Prediction Method for UD laminas *for the often fibre-dominated designed UD lamina-composed laminates* employs

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- 1) Failure mode-linked *load modelling* (novel idea)
- 2) Measurement of a minimum number of Master S-N curves
- 3) Prediction of other necessary *mode S-N curves* on basis of the master curve by the use of *strain energy equivalence*
- 4) Accumulation of damaging portions depends on cycles-linked shrinking of failure surface by FMC strength criteria.  
In-situ-effect, considered by deformation controlled testing.
- 5) Failure mode-linked *damaging accumulation* (novel idea)  
No mean stress correction to be performed.

To be done: Deeper investigation of the novel method and of the additional damaging caused by mode changes (FF1 to FF2 if  $R = -1$ ).

**Aim: Fatigue pre-dimensioning of ‘well-designed’, UD laminas-composed laminates just by lamina-dedicated mode-representative Master S-N curves, derived from *sub-laminate* test specimens, which capture the embedding (in-situ) effects, and S-N curves – *due to 3) possible* - from automatically constructed Haigh diagrams.**



Everything in the world  
is terminated  
by **chance** and **fatigue**.

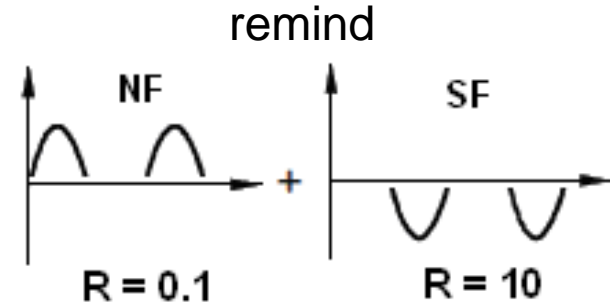
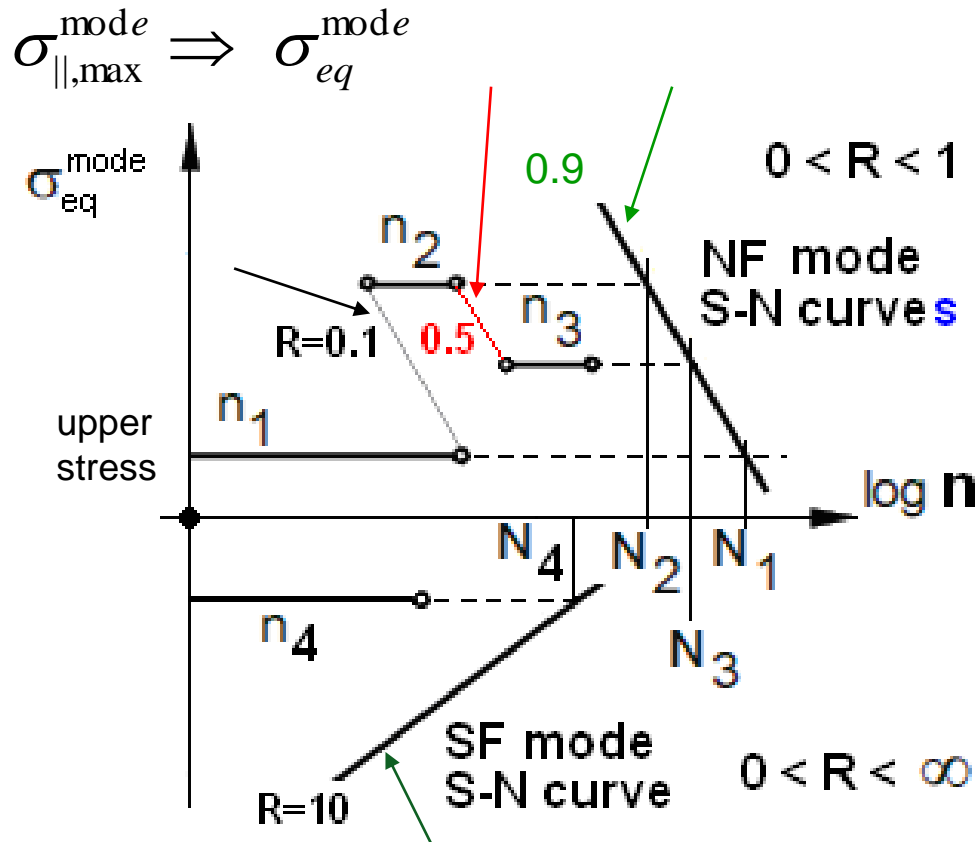
*Heinrich Heine*

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- 1 Introduction to Static and Fatigue Design
  - 2 Cuntze's Failure-Mode-Concept-based Strength Criteria
  - 3 Cuntze's Fatigue Life Estimation Concept
  - 4 Generation and Interpretation of UD Haigh Diagrams
  - 5 Steps of the Fatigue Life Prediction Method Proposed
  - 6 Example: Transversely-isotropic UD-CFRP**
- Conclusions

# Failure mode-based Lifetime Prediction Method

**Schematic Application** (principle: for simple isotropic case as example, 4 blocks)



here:

2 master curves

NF:  $R = 0.1$

SF:  $R = 10$

2 predicted curves

NF:  $R = 0.5, 0.9$

Miner application:

$$D = n_1 / N_1 + n_2 / N_2 + n_3 / N_3 + n_4 / N_4$$

# Ideas for Experimental Proof

## Choice of Test Specimens, Stress Combinations and Loading Types

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Demands on **test specimens**: Consideration of embedding of ply, ply-thickness effect, fibre volume fraction, stacking sequence, loadings

1 : Flat coupon material *test specimens* (relatively cheap compared to tubes)

2 : Tension/compression-torsion tube *test specimens*  $(\sigma_1, \sigma_2, \tau_{21})$

3 : Sub-laminate *test specimens* (with internal proof ply and outer supporting plies)

4 : Flat off-axis coupons (shortcomings 'free edge effect' + bi-axial stiffness loss not accurately considered)

**To be tested: Combinations of stresses** (3D or 2D state of stresses)

$$\{\sigma\} = (\sigma_1, \sigma_2, \sigma_3, \tau_{23}, \tau_{31}, \tau_{21})^T \Rightarrow \sigma_{\parallel}^t, \sigma_{\parallel}^c, \sigma_{\perp}^t, \sigma_{\perp}^c, \tau_{\perp\parallel} \text{ basic stresses}$$

**Model VALIDATION: Loading types** applied for the *operational lifetime estimation* are

- *Constant-amplitude loading* : delivers S-N curves (Wöhler curve)
- *Block-loading* : (if appropriate) for a more realistic Fatigue Life estimation
- *Random spectrum loading* : Fatigue Life (Gaßner) curve

# Assumptions for UD Modelling and Testing and Data Evaluation

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- **The UD-lamina is macroscopically homogeneous.**  
It can be treated as a homogenized ('smeared' material)
- **The UD-lamina is transversely-isotropic.**  
On planes, parallel with the fiber direction it behaves orthotropically and on planes transverse to fiber direction isotropically (quasi-isotropic plane)
- **Uniform stress state about the critical stress 'point'**
- **Test:**  
Pore-free material, specimen surfaces polished, well sealed (WWFE-II) , fiber volume is constant, tube specimens show no warping and do not bulge, perfect bonding, no layer waviness, edge effects do not exist, ...

*Load history, variable loading*

*Smeared (homogenized) composite material macro-scopically modelled.*

*It is composed of fiber, matrix and interphase*

*Fatigue model should be applicable for all laminates of the same material kind but different lay-up (stack) in order to further widen the use of composites*

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Stress (not strain) criteria are applied to determine the subsequent damaging portions:

- capture the combined effect of lamina stresses and
- consider residual stresses from manufacturing cooling down (essential for HCF)

- *Determination of damaging portions* (from diffuse and later discrete damaging)
- *Accumulation of damaging portions* (cycle-wise, block-wise, or otherwise ? )

# Failure mode-linked Master S-N-curves

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For lifetime estimation usually – even in a distinct failure mode – several S-N-curves are needed

*testing requires high effort!*

## Idea

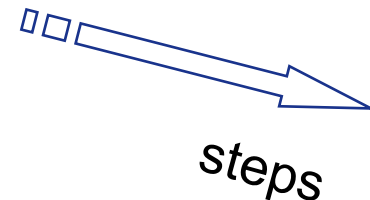
Measurement of just one failure mode linked Master S-N-curve

- for a fixed stress ratio  $R$
- prediction of additionally necessary S-N-curves on basis of the master curve and on the ‘principle of equivalent strain energy’!

Then, for the often used

all possible load orientations capturing fiber-dominatedly designed, multidirectional laminates, composed of UD plies,  
an engineering-like model is derivable.

Its characteristic steps are presented:





# Questions an engineer poses in the case of cyclic design

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1. **When does damaging start?**
2. **How can one consider the single (micro-)damaging portion?**
3. **How are the single damaging portions accumulated?**
4. **When do the accumulated damageing portions form a damage?**
5. **When becomes such a damage (delamination, impact) critical?**
6. **How is the damage growth in the 3rd or final phase of fatigue life (fixation of part replacement time, inspection intervals)?**



# Driver of damaging: Applicability of Static Strength Failure Conditions?

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## Proven Assumption:

If the damaging mechanisms (failure modes) are equal, then

- failure parameters that drive cyclic damaging are equal, too, and
- transferability from static failure to cyclic failure is permitted

However, static strength must be replaced by the  
fatigue strength = residual strength of the  
shrinking failure body.

Therefore,

as necessary static tool, *my*

FMC-based Static Failure Conditions (criteria) shall be briefly derived which  
were very successful in the World-Wide-Failure-Exercise (WWFE 1992-2014).

From all the contributors, my non-funded Failure Conditions  
well mapped the largest number of test data courses in WWFE-I and WWFE-II !

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***Step 4: Determination of Damaging Portions by Static Strength Criteria***